

# ELECTRON SPIN RESONANCE STUDY OF THERMAL DEFECTS IN SrLaAlO<sub>4</sub> AND SrLaGaO<sub>4</sub> High-*T<sub>c</sub>* SUBSTRATES

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The effect of annealing SrLaAlO<sub>4</sub> (SLA) and SrLaGaO<sub>4</sub> (SLG) crystals in oxidizing and reducing atmospheres in the temperature range of 950°C–1300°C was investigated. Three kinds of anisotropic defects  $D_1$ ,  $D_2$ , and  $E$  at the temperature range of 4–300 K were found. By diagonalization of orthorhombic spin-Hamiltonian parameters:  $|D| = 0.0541(10) \text{ cm}^{-1}$ ,  $|E| = 0.0108(10) \text{ cm}^{-1}$ ,  $g_{\parallel} = 0.883(5)$ , and  $g_{\perp} = 1.922(5)$  for  $D_1$  defects for SLG and SLA were calculated and they had the same values within the margin of error.

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## 1. Introduction

Single crystals of SrLaAlO<sub>4</sub> (SLA) and SrLaGaO<sub>4</sub> (SLG) are considered as the promising substrates for HTSC films. Both, SLA and SLG, compounds crystallize in a tetragonal structure similar to that of K<sub>2</sub>NiF<sub>4</sub>. This structure has  $I4/mmm$  space group symmetry [1]. There is little information in literature about thermal defects detected by electron spin resonance spectroscopy (ESR). It was reported [2] that as SLA crystals change colour from green-yellow to colourless the intensity of ESR signal is lowered, respectively. It was also reported [3] that crystals of SLG grow with different colours, green amber and yellow. Thermal defects denoted  $D$  and  $E$  were shown in Ref. [4]. These investigations are continued in this work.

## 2. Experiment

Single crystals of SLA and SLG were grown by the Czochralski technique with the use of iridium crucible. Both types of crystals contained iron impurity lower than  $10^{-3}$  at.%. The samples typically  $3.5 \times 3.5 \times 2 \text{ mm}^3$  were measured in a BRUKER ESP-300 ESR spectrometer (X-band). The spectrometer was equipped with a helium flow cryostat type ESR-900 Oxford Instruments. The ESR lines were observed in the temperature range from 4 K to 300 K.

## 3. Results and discussion

The angular dependence of ESR spectrum at 20 K is depicted in Fig. 1. The variation of ESR spectrum from  $H \parallel c$  to  $H \perp c$  in (110) plane are shown in Fig. 2. Figure 2a depicts the spectrum obtained from the as-grown SLG sample and Fig. 2b shows a series of spectra of samples annealed in a dynamic vacuum of  $10^{-5}$  torr at  $1300^\circ\text{C}$  for 1 h. The intensity of  $D_1$  and  $D_2$  spectra decreases and new "E" spectra appears (see Fig. 2b). In both figures we see two lines of  $D_1$  and one line  $D_2$  that have a large line width for  $H \parallel c$  ( $\Delta H = 100$  mT) and for  $H \perp c$  lines of  $D_1$  and  $D_2$  are placed in the lowest magnetic field and have a minimal line width ( $\Delta H = 20$  mT). In all spectra on the left we see the typical ESR spectra of iron doped samples. In the investigated crystals iron existed only as a residual impurity. After heat treatment in the dynamic vacuum the sample was annealed at  $1200^\circ\text{C}$  for 6 h under a pressure of 10 kbar in 20%  $\text{O}_2$  and 80% Ar atmosphere. The disappearance of "E" spectra and an increase in  $D_1$ ,  $D_2$  spectra were observed. The as-grown SLA sample was treated under the same conditions and changes of ESR spectra were not observed.

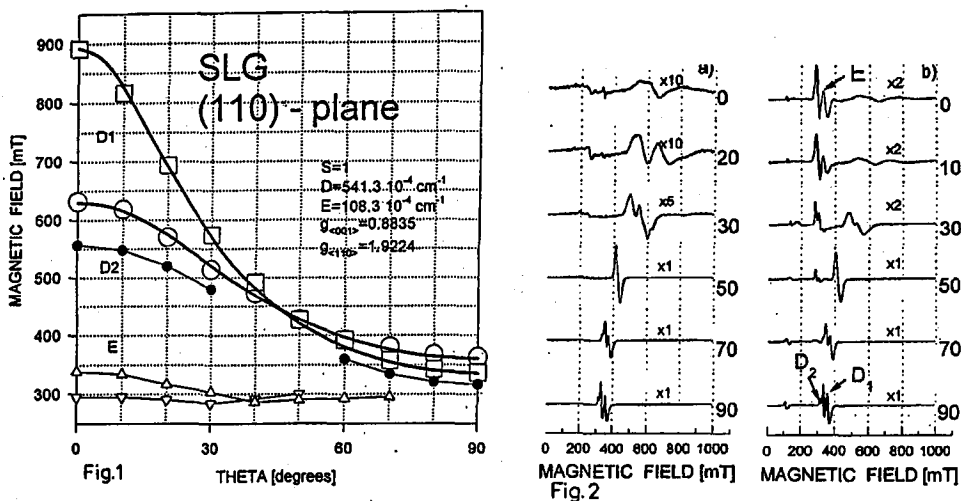


Fig. 1. Angular dependence of X-band spectra. Open squares are related to  $D_1$ -spectrum, the solid lines are calculated from spin-Hamiltonian with the effective spin  $S = 1$ , solid circles and open triangles are related to  $D_2$  and E defect, respectively. Fig. 2. The variation of  $D_1$ ,  $D_2$ , and E shape lines for SLG crystal obtained for X-band at  $T = 20$  K. The series of spectra are shown from  $H \parallel c$  ( $0^\circ$ ) to  $H \perp c$  ( $90^\circ$ ) in (110) plane, (a) as-grown sample, (b) annealed in a dynamic vacuum of  $10^{-5}$  torr at  $1300^\circ\text{C}$  for 1 h.

The spin Hamiltonian

$$\mathcal{H} = g\beta\mathbf{H} \cdot \mathbf{S} + D \left[ S_z^2 - \frac{1}{3}S(S+1) \right] + E(S_x^2 - S_y^2)$$

with  $S = 1$  was used to fit the experimental data with the theoretical calculations. The best fitting was obtained with the following parameters:  $g_{\parallel} = 0.883(5)$ , and  $g_{\perp} = 1.922(5)$ ,  $|D| = 0.0541(10) \text{ cm}^{-1}$ ,  $|E| = 0.0108(10) \text{ cm}^{-1}$  for  $D_1$  defects for SLG and SLA and they had the same values within the margin of error.

#### 4. Conclusions

Considering the crystal structure of both compounds it can be concluded that the position of oxygen in the lattice, O1 laying on the  $ab$  plane, is more stable than the oxygen O2 laying along the axis  $c$ . The results of ESR investigation have been interpreted as related to defects in which the oxygen O2 is missing (the spectrum marked  $D_1$ ,  $D_2$ ). The intensity of these lines depends on the colour of the crystals. The change of colour from colourless through yellow, green to dark-green correlates with the increasing of the number of defects represented in the ESR spectrum by lines  $D_1$ ,  $D_2$  (see Fig. 3). The observed splitting of the ESR line (see Fig. 4) can be interpreted in the case of better crystallographic perfection of the crystal and in consequence a lower width of the lines. In such case the ESR lines related to the particular defects  $D_1$ ,  $D_2$  can be separated from each other.

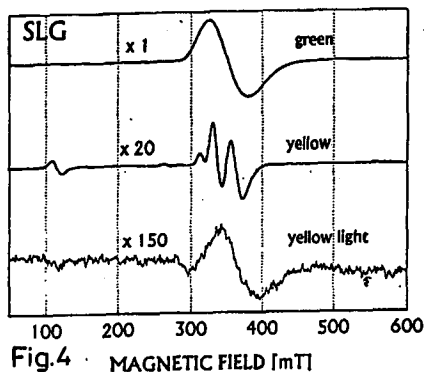
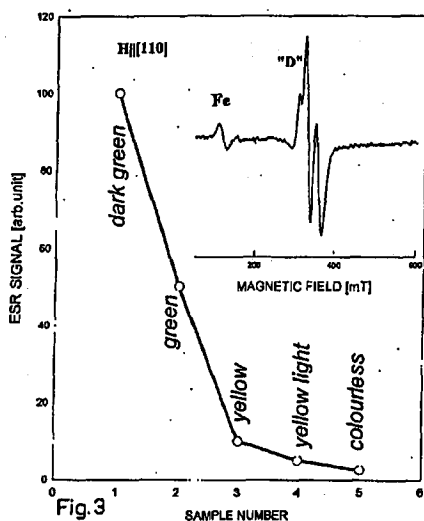


Fig. 3. ESR intensity signals for variously coloured crystals related to the intensity of  $D_1$  line shown in the spectrum of the inset. Spectra are given at X-band,  $T = 20 \text{ K}$ , and  $P = 2 \text{ mW}$ .

Fig. 4. The line shape of the ESR spectrum for different concentrations of defects.

## References

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