

Proceedings of the European Conference "Physics of Magnetism 96", Poznań 1996

MAGNETIC PROPERTIES OF THIN FILMS OF Fe-Cu-R-Si-B ALLOYS WITH R = Nd, Gd, Ho, Y

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Hysteresis loops of thin films of $\text{Fe}_{73.5}\text{Cu}_1\text{R}_3\text{Si}_{13.5}\text{B}_9$ (R = Nd, Gd, Ho, Y) alloys were studied by the magneto-optical Faraday method. The films were deposited by flash evaporation technique onto cooled glass substrates. Lanthanides were chosen as substitutes for niobium in the FINEMET type alloys. Coercivity, H_c , was determined from the easy- and hard-axis hysteresis loops. The as-deposited films containing Gd and Ho are characterized by rather low values of H_c while the films with Nd and Y show a high isotropic value of H_c . Attempts were made to determine the conditions of the film annealing which would ensure the optimum soft magnetic properties of the films studied.

PACS numbers: 75.70.-i, 68.55.-a

Magnetic properties of novel magnetic materials as well as possibility of their utilization are the subject of great interest of scientists and engineers. E.g. nanocrystalline alloys of the $\text{Fe}_{73.5}\text{Cu}_1\text{Nb}_3\text{Si}_{13.5}\text{B}_9$ type belong to such materials [1, 2]. In this paper we present results of a study of magnetic properties of thin films made on the basis of the classical nanocrystalline alloy $\text{Fe}_{73.5}\text{Cu}_1\text{R}_3\text{Si}_{13.5}\text{B}_9$, in which Nb has been replaced by rare-earth elements R. The lanthanides R = Nd, Gd, Ho and Y (non-lanthanide) were chosen as substitutes in the FINEMET type alloys to test whether they generate nuclei of various magnetic structures [3] and what differences in the magnetic properties they introduce.

The films were prepared by flash evaporation technique and deposition in an ultrahigh vacuum onto glass substrates cooled with liquid nitrogen [4]. The initial material for evaporation was a pulverized alloy with the nominal composition $\text{Fe}_{73.5}\text{Cu}_1\text{R}_3\text{Si}_{13.5}\text{B}_9$. The films obtained were of 20 to 50 nm in thickness.

In order to determine the temperatures at which the structural changes occur, the temperature dependences of electrical resistivity $\rho(T)$ were analysed. The temperature increased at a rate of about 10 K/min. The samples studied were annealed at temperatures at which changes in their structure were observed.

The resistivity $\rho(T)$ was analysed for the films covered with the protective SiO films and without that protection.

Hysteresis loops of the films obtained were studied by magneto-optical Faraday method and the coercive field, H_c , was determined.

Analysis of the $\rho(T)$ dependences (Fig. 1) obtained for films of all the alloys studied reveals two characteristic types of courses. The first one suggests structural changes of the relaxation type in the low-temperature range, down to about 623 K and changes related to crystallization above 700 K (Fig. 1a). The curves of the

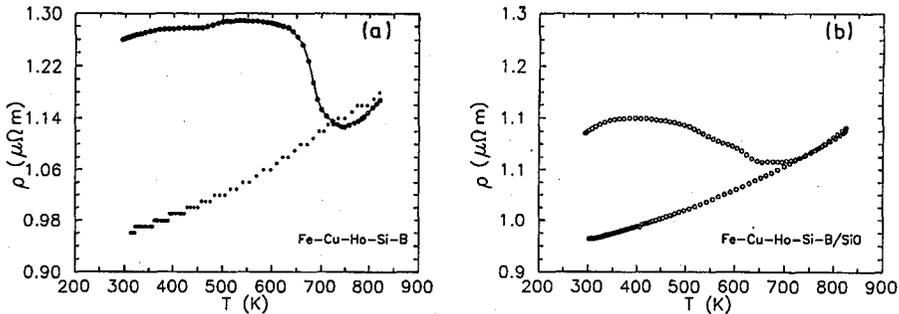


Fig. 1. Two exemplary $\rho(T)$ dependences in the films of Fe-Cu-Ho-Si-B for a sample (a) not covered with SiO layer; and (b) covered with the SiO protective layer.

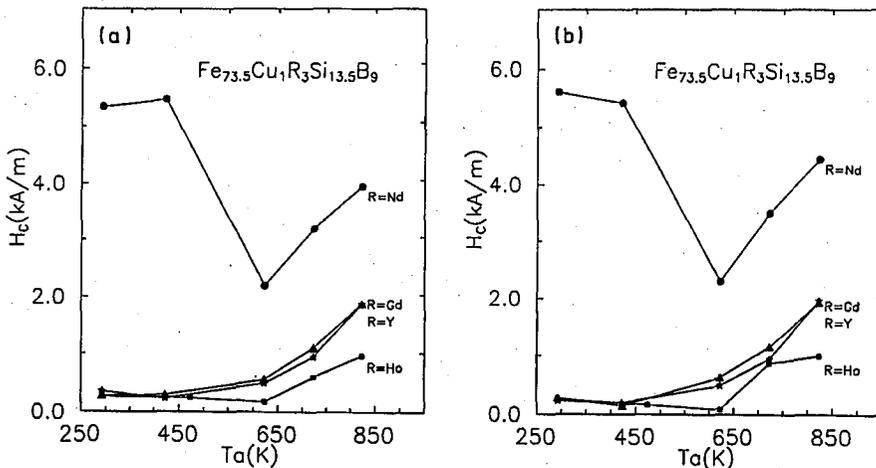


Fig. 2. H_c^{\parallel} (a) and H_c^{\perp} (b) vs. T_a for the films of Fe-Cu-R-Si-B (R = Nd, Gd, Ho, Y).

latter type indicate only the occurrence of the crystallization transition (Fig. 1b). Accordingly, it can be concluded that the films characterized by the former type of $\rho(T)$ dependence are amorphous as-deposited while those characterized by the latter type of $\rho(T)$ have already partly relaxed on evaporation of the protective SiO film. This effect does not depend on the kind of alloy but rather on their thermal treatment. For as-deposited amorphous films, the temperature of crystallization, T_x , can be relatively accurately estimated to be of about 673 K for all films studied. Such a low value of T_x makes highly unlikely the occurrence of the nanocrystalline structure.

Hysteresis loops and temperature dependences of H_c were measured for the films not covered with a film of SiO. Figure 2 presents the temperature dependences of coercive field for the as-deposited films annealed for 10 minutes at the structure relaxing temperatures $T_{a1} = 423$ K, $T_{a2} = 623$ K, and at temperatures higher than the crystallization temperature: $T_{a3} = 723$ K, $T_{a4} = 823$ K. A general character of the dependences of H_c^{\parallel} (Fig. 2a) and H_c^{\perp} (Fig. 2b) on the annealing temperature T_a is the same, at first H_c decreases and increases after crystallization. However, a detailed analysis of $H_c^{\parallel}(T_a)$ reveals that the value of H_c is largely affected by the magnetic structure of crystal nuclei which is sperimagnetic in Nd-Fe, ferrimagnetic in Gd-Fe and Ho-Fe and asperomagnetic in Y-Fe. The as-deposited Fe-Cu-Nd-Si-B films are characterized by high $H_c > 5$ kA/m and isotropic in-plane magnetization. The value of their H_c decreases with growing T_a and increases after crystallization point. For Fe-Cu-Ho-Si-B films the coercive field H_c^{\parallel} changes from 0.27 kA/m for as-deposited state to 0.18 kA/m after annealing at 623 K and subsequently increases to 0.97 kA/m after annealing at $T_{a4} = 823$ K (Fig. 2). The H_c values obtained for Fe-Cu-Gd-Si-B are similar to those of the alloy containing Ho. The films of the alloy containing Y (Fig. 3) have

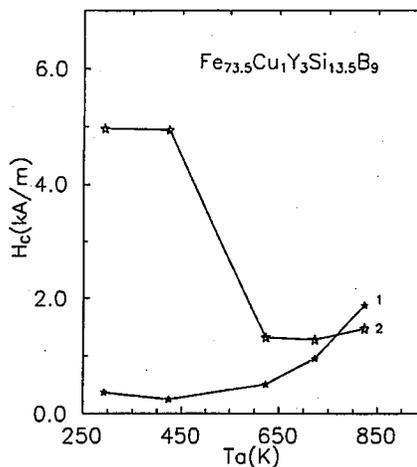


Fig. 3. H_c^{\parallel} vs. T_a for the films of Fe-Cu-Y-Si-B; (1) not covered with SiO, (2) covered with SiO protective layers.

their H_c^{\parallel} values very sensitive to the conditions of thermal treatment and changing from 0.36 to 5 kA/m.

Two types of temperature dependence of resistivity $\rho(T)$, depending on the amount of the amorphous phase, were observed for the films studied. The values of the coercive field of the films were found to depend on the type of magnetic structure of crystalline clusters. The highest values: $H_c > 5$ kA/m, were measured for Nd-Fe of sperimagnetic structure and Y-Fe of asperomagnetic structure. Films of the alloys containing Gd and Ho were characterized by low H_c values as expected of the ferrimagnetic Gd-Fe and Ho-Fe.

We shall continue the studies on a larger range of lanthanides in order to confirm our conclusions.

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