ARSENIC ANTISITE DEFECTS CORRELATIONS
IN LOW TEMPERATURE MBE GaAs*

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GaAs layers grown by molecular beam epitaxy at low temperatures (LT GaAs) have several interesting properties. For example, concentrations up to $2 \times 10^{20}$ cm$^{-3}$ of neutral and up to $5 \times 10^{18}$ cm$^{-3}$ of positively charged As$_{Ga}$ defects have been determined in as grown layers. It has been observed that electrical transport in LT GaAs is dominated by hopping conductivity. In the as grown and annealed up to 400°C layers, experiments show a high mobility of photo-excited electrons up to 20000 cm$^2$/Vs at about 130 K. Taking into account the very high concentration of ionized defects in LT GaAs, the best possible explanation of the measured value of mobility is an assumption that As$_{Ga}^+$ defects interact with ionized acceptors A$^-$. This leads to creation of As$_{Ga}^+\cdots$A$^-$ dipoles, which do not scatter electrons as efficiently as single ions. It has been shown that the donor–acceptor correlation could be destroyed by illumination which ionizes As$_{Ga}$ defects. Also analysis of hopping conductivity suggests the existence of the donor–acceptor correlations in non-illuminated LT GaAs. Numerical calculation of As$_{Ga}$–acceptors interaction has been done. It shows that at least below 200 K dipole formation in LT GaAs is possible and could lead to observed increase in mobility.

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1. Introduction

Defects correlations in semiconductors are rather rarely studied probably because of two contradictory conditions which should be fulfilled to obtain observable defects correlation: material should have good crystal structure and it should have high concentration of point defects. In poor quality crystals the concentration of extended defects is too high to observe effects originating from the correlation of point defects. In high quality materials, on the other hand, the concentration of point defects is usually too low to induce ordering. It seems that these two conditions are fulfilled by low temperature gallium arsenide (LT GaAs) layers. One of the most important features of LT GaAs is high nonstoichiometry (1.5 at.% of excess As) [1] which causes a high concentration of point defects and an expansive lattice strain of approximately 0.1% [2]. Nevertheless samples grown at 200°C or

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higher exhibit high crystalline perfection [2]. The concentration of point defects is remarkable. Neutral arsenic antisite $\text{As}_{\text{Ga}}$ defects dominate having concentration about $10^{20} \text{ cm}^{-3}$. Since the Fermi level is pinned to $\text{As}_{\text{Ga}}^{0/\text{+}}$ level, the number of ionized $\text{As}_{\text{Ga}}^{\text{+}}$ centers should be equal to the number of acceptors (mainly $\text{V}_{\text{Ga}}$). Its concentration ($N_{\text{As}_{\text{Ga}}^{\text{+}}}$) is $3-5 \times 10^{18} \text{ cm}^{-3}$ [3, 4] as measured by electron paramagnetic resonance (EPR). Concentration of $\text{V}_{\text{Ga}}$ defects about $10^{18}-10^{19} \text{ cm}^{-3}$ has been estimated by positron annihilation experiment [5]. This amount of defects is sufficiently high to create a deep defect band. It has been shown that in as grown LT GaAs layers hopping conductivity between arsenic antisite defects exists [3, 6].

2. Experimental

2.1. Samples

LT MBE GaAs layers have been grown at temperature 190°C under As rich conditions in MIT Lincoln Laboratory by Drs. F.W. Smith and R. Calawa. Layers of 2.0–2.5 μm thickness have been grown on semi-insulating substrate. The samples were annealed in the MBE system at temperatures from 300 to 600°C. As calculated from the near-infrared optical absorption spectra using Martin's calibration curve [7], the concentration of arsenic antisite defect in as grown samples was $1 \times 10^2 \text{ cm}^{-3}$, while in samples annealed at 400°C its concentration was about five times lower. Annealing at 600°C caused a decrease in $\text{As}_{\text{Ga}}$ defect concentration to such a level that optical detection was very difficult. Using results of hopping conductivity measurements $N_{\text{As}_{\text{Ga}}}$ was estimated to be $9 \times 10^{18} \text{ cm}^{-3}$.

2.2. Hopping conductivity

Concentration of arsenic antisite defects ($N_{\text{As}_{\text{Ga}}} \approx 10^{20} \text{ cm}^{-3}$) in LT GaAs is high enough to create a deep defect band. It has been shown that in as grown LT GaAs layers the hopping conductivity between arsenic antisite defects exists [3, 6] (see Fig. 1). In as grown samples hopping dominates even in room temperature. After annealing of the layer, hopping decreases but still could be observed in temperature below 250 K. In higher temperature electrons are thermally excited from $\text{As}_{\text{Ga}}$ to the conduction band and the free electron transport dominates.

According to theory of hopping conductivity, in samples with high impurities concentration it is possible that electrons can move from one center to another by tunneling between them. This conduction mechanism is important when temperature is too low for thermal excitation of electrons to the conduction band. Just below this temperature the following relation is expected [8]:

$$\sigma = \sigma_0 \exp \left( \frac{\varepsilon_3}{kT} \right),$$  \hspace{1cm} (1)

where $\varepsilon_3$ is an average difference of potential energies between initial and final state of tunneling electron. This difference is usually caused by potential fluctuations in crystal. Theoretical line plotted in Fig. 1 shows that relation (1) fits to the experimental data down to 100 K. In lower temperature the conductivity changes more slowly which is in agreement with the theory developed by Mott and Shklovskii [8, 9].
By fitting Eq. (1) to the experimental data, energies $\varepsilon_3$ for all samples were determined (see Table). The energy $\varepsilon_3$ can reach 0.2 eV (in as grown material). Since an average distance between centers is about $20 \, \text{Å} \left(d = (N_{\text{AsGa}})^{-1/3}\right)$ it would suggest existence of an electric field $E = 10^6 \, \text{V/cm}$. Another possible explanation is that As$_a$ center interacts with ionized acceptor $\text{A}^-$. In order to explain the high value of $\varepsilon_3$ the two centers have to lie very closely. It is possible in situation when As$_a^+\text{Ga}^-\text{A}^-$ correlation exists.

### 2.3. Transport in the conduction band

In the case of hopping conductivity the Hall mobility is infinitesimal, which has been confirmed by the measurements of LT GaAs. However, in higher temperature where transport in conduction band dominates an effective mobility of the order of $10^3 \, \text{cm}^2/(\text{V s})$ was observed (continuous line in Fig. 2). It was due to transport by electrons thermally excited to the conduction band.

In the case of two transport mechanisms effective mobility and concentration
are as follows (since $\mu \leq 1 \text{m}^2/(\text{V s})$ and $B = 0.05 \text{T}$, components containing $\mu^2 B^2$ were neglected):

$$\mu_{\text{eff}} = \mu \frac{n^* \mu \rho_3 e}{1 + n^* \mu \rho_3 e}, \quad n_{\text{eff}} = n^* \frac{n^* \mu^2 \rho_3 e}{(1 + n^* \mu \rho_3 e)^2},$$

(2, 3)

where $n^*, \mu$ — concentration and mobility of excited electrons, $\rho_3$ — hopping resistivity. Value of $\rho_3$ can be approximated from dependence (1) fitted to the low temperature data. Basing on Eqs. (2) and (3), the Hall mobility $\mu$ and concentration $n^*$ as functions of temperature were calculated (see Fig. 2). Unfortunately, below 310 K the electron concentration decreased so strongly that it was impossible to calculate mobility. But above this temperature, it seems that the method gives reliable results. For example, concentration of excited electrons is in agreement with assumption that they came from As$_{Ga}$ deep level (activation energy $E_a = 0.67 \text{eV}$ is lower than expected value 0.75 eV probably due to potential fluctuations in sample). However, the obtained value of mobility $\mu \approx 5000 \text{cm}^2/(\text{V s})$ is much higher than expected. In the sample with ionized impurity concentration of the order of $10^{19} \text{cm}^{-3}$ the mobility should be lower than $10^3 \text{cm}^2/(\text{V s})$. One of the possible explanation is that positively and negatively charged impurities are correlated which leads to decrease in electron scattering.

**2.4. Electrical transport under illumination**

Photoconductivity measurement of LT GaAs shows (see Fig. 3) that PC spectrum was dominated by photo-ionization band at $h\nu = 0.8-1.5 \text{eV}$ which was identified as photoionization spectrum of EL2 defect (As$_{Ga}$) [10]. In low tempera-
ture (12 K), this band could be quenched by illumination with white or monochromatic $h\nu = 1.3$ eV light. The band recovers in temperature higher than 130 K. It means that metastable behavior of this band is the same as behavior of EL2 ($\text{As}_{\text{Ga}}$) photoionization band in standard semi-insulating (SI) GaAs.

As was shown in Fig. 2, below room temperature it was impossible to measure the Hall effect on LT GaAs. However, after illumination of layers, a dramatic increase in mobility was observed. It was due to the transport by electrons excited to the conduction band. In the experiment, monochromatic light with energy $h\nu = 1.9$ eV and $h\nu = 1.3$ eV was used. Light of $h\nu = 1.9$ eV creates electron–hole pairs whereas 1.3 eV light excites electrons from $\text{As}_{\text{Ga}}$ defects to the conduction band.

Based on measured dependence of $\mu_{\text{eff}}$ and $n_{\text{eff}}$ versus temperature, using Eqs. (2) and (3), the excited carriers concentration $n^*$ and their mobility $\mu$ as functions of temperature were calculated. It was taken into account that $\rho_3$ value was the same as in the dark. Figure 4A shows dependence of the Hall concentration of photo-excited electrons versus temperature. The activation energy of the photo-excited carriers concentration was determined as $E_a = 0.07 \pm 0.02$ eV. This value corresponds to the activation energy of electron capture cross-section of EL2 ($\text{As}_{\text{Ga}}$) $E_b = 0.066$ eV [11], which suggests that electron lifetime $\tau_n$ is governed by the capture by $\text{As}_{\text{Ga}}$ defects.

The mobility of the excited carriers versus temperature for as grown sample was shown in Fig. 4B. Using 1.9 eV light, the mobility of the photo-excited electron having maximum at temperature 130 K equal to $\mu = 20000$ cm$^2$/V s was observed. Absorption coefficient for the $h\nu = 1.9$ eV light is $\alpha = 3.7$ $\mu$m$^{-1}$, electron diffusion length in LT GaAs is about $L_D = 0.1$ $\mu$m, therefore nearly all photo-excited electrons should stay in the LT GaAs layer. While illuminating with the 1.3 eV light, analogous results for mobility were obtained (see Fig. 4B), although for this light the sample is more transparent and part of it (about 2%)

![Graph showing photocurrent spectra of LT GaAs](image)
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is absorbed in the substrate which complicates the effect. It can be noticed that below certain temperature the mobility quickly decreases. It is probably connected with a strange effect of lowering of mobility by intensive illumination which occurs in low temperatures. The effect is stronger for the 1.3 eV light than for the 1.9 eV light. Therefore, for the further analysis peak value of the mobility should be taken rather than the full curve. It is worth noticing that the mobility value $\mu = 20000 \text{ cm}^2/(\text{V s})$ is nearly two orders of magnitude higher than expected for material with $10^{20} \text{ cm}^{-3}$ defects and about $10^{19} \text{ cm}^{-3}$ ionized centers.

It is proposed that both these effects, high mobility value and the lowering of mobility by intensive illumination, could be connected with spatial correlation of ionized centers. Under weak illumination the correlated ionized impurities very weakly scatter electrons. Under intensive illumination the correlations are destroyed and scattering increases. This observation gives opportunity to gather more information about the postulated correlations. For this purpose an experiment with two beams of light was performed.

2.5. Two beam experiment

As it has been described above, in temperature range 110–160 K under illumination of 1.9 eV light high mobility of electrons in LT GaAs has been observed. It was suggested that electron scattering in the sample is weak because of correlation between charged centers leading to formation of $\text{As}^+_{\text{Ga}} - \text{A}^-$ pairs (see Fig. 6). Since number of acceptors is much lower than number of $\text{As}_{\text{Ga}}$ defects and electrons can migrate via hopping between $\text{As}_{\text{Ga}}$ defects, it is possible that the $\text{As}_{\text{Ga}}$ defect closest to the acceptor would be chosen and positively ionized. When the material with correlated donor–acceptor pairs is illuminated by light which can ionize de-

![Diagram of Hall effect measurements](image-url)
Arsenic Antisite Defects Correlations...  

In equilibrium Under Illumination

Fig. 5. Hall mobility measured under illumination with two beams of light at $T = 120$ K. The first beam, $h\nu = 1.9$ eV, generates electrons, the second can also ionize deep centers.

Fig. 6. Model of defect correlation: left, in equilibrium ionized impurities $\text{As}^+_\text{Ga}$ and acceptors $A^-$ form dipoles; right, under illumination, light destroys dipoles.

Defects (see Fig. 6 right), it is possible that part of uncorrelated $\text{As}_{\text{Ga}}$ defects will be ionized and that excited electrons will recombine on previously correlated centers producing unpaired acceptors. Therefore, instead of the dipoles a great number of unscreened ionized impurities is created which effectively scatters carriers.

Results of this experiment, plotted in Fig. 5 shows that indeed illumination with second beam changes electron mobility. Carriers were excited using low intensity 1.9 eV light. Second beam was swept from 0.5 to 2.0 eV. A wide band ranging from 0.8 to 1.5 eV has been observed. The change of mobility has been remarkable, nearly two orders of magnitude. Comparison with Fig. 3 shows that this "low mobility band" corresponds to photoionization band of $\text{As}_{\text{Ga}}$ defect, thus the effect
is obviously connected with AsGa ionization. The Hall concentration of electrons under this illumination is about $10^{11}$ cm$^{-3}$. The change of ionized defects concentration should be equal to this number. Therefore it is impossible that increase in scattering was caused by increase in ionized defects concentration. Simultaneously, changes in ordering could be very significant, because time of repairing of correlation is much longer than the electron lifetime. Concluding, this experiment directly confirms existence of ionized impurity correlations in LT GaAs and shows crucial role of AsGa in these correlations.

3. Numerical calculations

From experiment, it is known that in LT GaAs correlations of ionized defects exist which causes an increase in electron mobility. It was also proved that AsGa defect plays crucial role in this ordering. J. Mycielski [12] has shown that in HgFeSe an ordering in space of ionized donors occurs which has a dramatic effect on scattering of free carriers. In this model charged centers should form a superlattice of equally spaced centers. Unfortunately, liquifying temperature of this system is rather low, $T_0 \approx 40$ K. The idea of correlation which is described in this paper is different. It assumes that negatively charged acceptors and positively charged AsGa defects form pairs.

Let us take into account that there is an acceptor $A^-$ surrounded by AsGa defects. One of these defects should be ionized. Taking into account the Coulomb interaction between charged centers, probability of existence of ionized AsGa defect at a distance $r_i$ from the acceptor is given by equation

$$P_{AsGa}^+(r_i) = A P_{AsGa}(r_i) \exp \left( -\frac{e^2}{4\pi\varepsilon_0\varepsilon_r r_i k_B T} \right),$$

where $P_{AsGa}(r_i)$ is the probability of existence of AsGa defect at a distance $r_i$ from the acceptor and $A$ is a normalization factor defined by the equation

$$\sum_{r_i \in V} P_{AsGa}^+(r_i) = 1,$$

where $V$ is the volume of crystal containing one acceptor.

If we take into account random distribution of AsGa defect the probability $P_{AsGa}(r_i)$ is

$$P_{AsGa}(r_i) = d^3 n_s(r_i) N_{AsGa} / 4,$$

where $d$ is lattice constant and $n_s(r_i)$ is number of gallium sites at the distance $r_i$ from the acceptor. Distribution (6) could be changed for example by As diffusion during growth.

Based on Eqs. (4), (5), (6), probability distributions $P_{AsGa}^+(r)$ for different temperatures were calculated. Analysis of the results (see Fig. 7) shows the significant increase in probability $P_{AsGa}^+(r)$ for small radius $r$ in low temperature. We can conclude that in temperature below 200–300 K the probability of placing ionized AsGa defect closely to the acceptor is very high so the postulated correlations should exist. In temperature 400 K or higher correlations are dissolved. Curve plotted for $T = \infty$ is proportional to $P_{AsGa}(r)$ distribution.
It is important to compare the experimentally measured mobility with the theoretical calculations based on the correlation model (see Fig. 8). The following scattering mechanisms were taken into account for calculation of electron mobility in LT GaAs: optical and acoustic phonon scattering, scattering on ionized impurities and scattering on dipoles. Since free electron concentration is too low for effective screening of charged centers, the Conwell–Weisskopf formula [13] has been applied for the scattering on ionized impurities. The effect of the dipole scattering was calculated using Dimitrov formula [14]. Based on the Erginsoy formula [15] it has been estimated that scattering on deep neutral impurities could be neglected. Calculations have been done for two acceptor concentrations $N_A = 1 \times 10^{18}$ cm$^{-3}$ and $N_A = 5 \times 10^{18}$ cm$^{-3}$. Three models have been checked. The first one assumes that charged centers in the sample are not correlated and $N_{As^+_{Ga}} = N_A$. The second model stands that ionized $As^+_{Ga}$ defects are distributed according to Eqs. (4), (5), (6). The third takes into account very weak As diffusion during growth, therefore the distribution (6) is replaced by the distribution calculated using numerical simulation of As diffusion. This diffusion allows arsenic ions to be moved over few angstroms distances. It means that in the crystal there are built-in correlations between arsenic antisites and acceptors (although they are very weak).

The results of numerical calculations are compared with experimental data in Fig. 8. As is seen, in the case of scattering on uncorrelated impurities the calculation results strongly differ from experimental data. The discrepancy is higher than order of magnitude. The models with correlations fit much better. The model
4. Conclusions

In spite of a great concentration of defects in LT GaAs, carriers excited to the conduction band have high mobility (nearly 20000 cm²/(V s) at 130 K). Such high mobility suggests that ionized impurities are not randomly distributed, but their positions are correlated. It was assumed that mainly the As₉₀Ga defect nearest to acceptors are ionized. Numerical calculations proved that formation of the postulated correlation is possible. The As⁺₉₀Ga–A⁻ pairs should be spontaneously created below room temperature due to the Coulomb interaction. It has been shown that this effect could explain the experimentally observed phenomena.

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References