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# SOME MAGNETIC PROPERTIES OF THIN FILMS PREPARED FROM Fe-(M-Nb)-Si-B (M: Cu, Ag, Pt, Pd) ALLOYS\*

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Saturation magnetic induction,  $B_s$ , and coercive field,  $H_c$ , have been studied upon annealing in films flash evaporated from  $\text{Fe}_{73.5}\text{M}_1\text{Nb}_3\text{Si}_{13.5}\text{B}_8$  alloys. The changes in  $B_s$  significantly diverged at increasing annealing temperature,  $T_a$ , and did not indicate any formation of ultrafine grain phase in the films.  $H_c$  measured as an angular function in the plane of the films revealed oscillations due to weak in-plane anisotropy. The coercivity increased in the course of annealing at increasing  $T_a$ , while the in-plane anisotropy decreased at the same time.

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## 1. Introduction

Excellent soft magnetic properties of recently discovered nanocrystalline alloys have stimulated intense investigation of these materials both in the cognitive aspect and in the aspect of their application [1-4]. Production of such soft magnetic materials in the form of thin films would significantly enlarge the scope of their application.

An attempt was made to obtain thin films of nanocrystalline  $\text{Fe}_{73.5}\text{Cu}_1\text{Nb}_3\text{Si}_{13.5}\text{B}_8$  alloy by flash evaporation technique, but it proved only partly successful as, apart from large content of the ultrafine phase, the samples always contained a significant contribution of the phase of larger grains [5]. Certain magnetic properties of the obtained films were also measured [6-8].

In this paper we report new results of our study on the magnetic behaviour of the films obtained.

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## 2. Experimental

The films were obtained by flash evaporation of  $\text{Fe}_{73.5}\text{M}_1\text{Nb}_3\text{Si}_{13.5}\text{B}_9$  alloys, where  $\text{M} = \text{Cu}, \text{Ag}, \text{Pt}, \text{Pd}$ . The films were deposited onto cooled glass substrates. Electron diffraction study of the films indicates amorphous state of as-deposited samples which crystallize during heat treatment. After annealing for 10 min within the range from about 600 to 823 K, the nanocrystalline phase appeared in a few samples [5, 6].

The Hall effect has been used for determination of the saturation magnetic induction  $B_s = 4\pi M_s$  ( $M_s$  = saturation magnetization) while for the hysteresis loop tracing and determination of coercive field,  $H_c$ , we applied the magneto-optical Faraday effect.

## 3. Results and discussion

The dependence of the saturation magnetic induction,  $B_s$ , on the annealing temperature,  $T_a$ , for the alloys with  $\text{M} = \text{Cu}, \text{Ag}, \text{Pd}$  and  $\text{Pt}$  shows that, except for  $\text{M} = \text{Pt}$ , the magnetic induction  $B_s$  increases with increasing  $T_a$  (Table). For  $\text{M} = \text{Ag}$ ,  $B_s$  oscillates with subsequent annealings and it comes back to the nearly initial value after annealing at 923 K (Table). For the alloy with  $\text{M} = \text{Pt}$ ,  $B_s$  drastically decreases upon annealing. Thus, although it is clear from our TEM

TABLE  
 $B_s$  values upon annealing of the samples  
for different M metals in the alloys.

Sample, M =	Cu	Ag	Pd	Pt
as-depos.	1.62	1.85	1.72	1.80
$T_a = 473$	–	1.69	1.74	1.75
623	1.58	1.80	1.77	1.43
723	1.63	1.70	1.89	1.31
823	1.68	1.79	2.10	–
923	1.69	1.83	–	–

observation [6] that the nanocrystalline phase is formed during the annealing process, this fact cannot be deduced from the behaviour of  $B_s(T_a)$  only. Soft magnetic properties are connected with disappearance of magnetostriction and magnetocrystalline anisotropy. The films studied, when as-deposited, reveal very weak in-plane magnetic anisotropy. This fact makes it impossible to measure the anisotropy directly from hysteresis loops, but the in-plane anisotropy gives periodic oscillations in coercivity, i.e. in  $H_c$  measured versus the angle of the applied measuring field. The amplitude of these oscillations,  $\Delta H_c$ , can be taken as an approximate measure of the latter anisotropy, if exists. Figure 1 shows the angular dependence of  $H_c$  for the as-deposited and the annealed, at  $T_a = 823$  K, sample of the alloy with  $\text{M} = \text{Cu}$ . The oscillations of  $H_c$  are well apparent for the as-deposited film. After

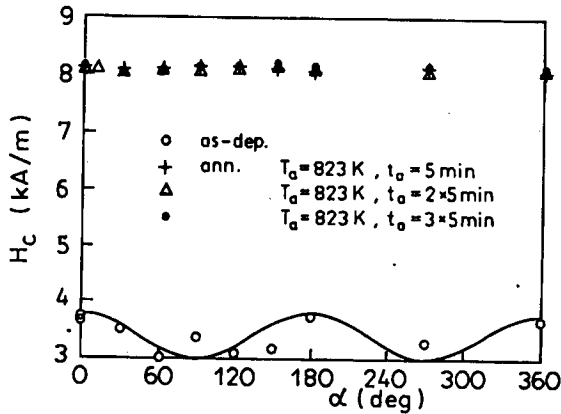


Fig. 1. Angular dependence of coercive field, as measured in the plane of the film: circles — as-deposited sample, stars, triangles and crosses — samples annealed for 5, 10, and 15 min, respectively. Continuous line represents the  $H_c$  oscillations corresponding to a reasonable value of uniaxial in-plane anisotropy.

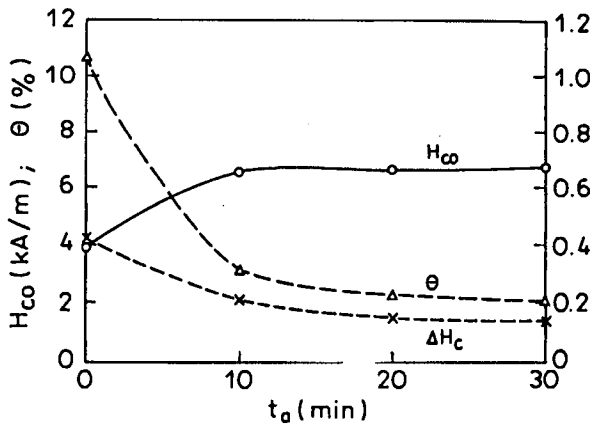


Fig. 2. Easy axis coercive field,  $H_{co}$ , amplitude of angular  $H_c(\alpha)$  oscillations,  $\Delta H_c$ , and  $\theta = \Delta H_c / H_{co}$  the supposed measure of the in-plane magnetic anisotropy, as a function of annealing time,  $t_a$ .

annealing of this film for only  $t_a = 5$  min,  $H_c$  increases and already achieves a high value. At the same time, the amplitude of the coercivity,  $\Delta H_c$ , significantly decreases demonstrating vanishing of the in-plane anisotropy.

Figure 2 illustrates the annealing time dependence, at  $T_a = 823$  K, of the (1) easy axis coercivity,  $H_{co}$ , (the temperature  $T_a = 300$  K has been attributed to the as-deposited samples); (2) oscillation amplitude,  $\Delta H_c$ ; (3) the relative value  $\theta = \Delta H_c / H_{co}$ , expressed in percent is the supposed measure of the uniaxial in-plane magnetic anisotropy. This value is almost independent of the applied field direction. This means that upon heating the uniaxial in-plane anisotropy signif-

icantly decreased, although at the same time the  $H_c$  value increased. The latter value remains constant in spite of subsequent annealings at the same temperature. On the other hand, for many evaporated samples annealed for 10 min in a low temperature range,  $H_c$  decreases to a minimum and then strongly increases when the annealing temperature exceeds 600 K. Thus, the increase in the  $H_c$  value upon heating is in contradiction to the vanishing of the anisotropy. Such behaviour is, of course, at variance to that found for nanocrystalline ribbons.

#### 4. Conclusion

It is well known that the nanocrystalline phase reveals a low coercivity. Thus, the above discussed results allow us to conclude that apart from a large content of the ultra-fine grain phase the samples contained significant, and increasing on annealing, amount of larger, single-domain grains, which was responsible for the rise in the coercivity of the sample after its annealing.

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