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# Influence of Boron Addition on Magnetic Properties of $\text{Sm}_2\text{Fe}_{17}$ Alloy

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With the boron addition, the Curie temperature of  $\text{Sm}_2\text{Fe}_{17}$  permanent magnet was observed to increase from 130 °C and reach its highest value of 362 °C for the sample at  $x = 4$ . In the X-ray diffraction measurements, the peaks diffracted from the planes of hard magnetic  $\text{Sm}_2\text{Fe}_{17}$  phase, observed in the sample without boron, disappeared with the boron addition and magnetic  $\text{SmFe}_3$  and  $\text{Fe}_2\text{B}$  phases were detected. For all samples, the major peaks belong to the  $\alpha$ -Fe phase. According to the magnetization measurements, the  $M_s$  value of the sample without boron is 155.52 emu/g, it decreased to its lowest value of 97.93 emu/g for the sample at  $x = 4$ . Boron addition resulted in an evolution of the phase constitution that caused changes in magnetic properties of the alloys.

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## 1. Introduction

$\text{Sm}_2\text{Fe}_{17}$  compound which has a rhombohedral  $\text{Th}_2\text{Zn}_{17}$ -type structure has an easy plane magnetocrystalline anisotropy at room temperature and relatively low Curie temperature. To improve intrinsic magnetic properties, small interstitial atoms such as B, N, and C have been introduced into  $\text{Sm}_2\text{Fe}_{17}$  structure. The  $\text{Sm}_2\text{Fe}_{17}$  phase can absorb appreciable amount of C or N atoms without losing or transformation of its basic structure [1–3]. The interstitial modifications of  $\text{Sm}_2\text{Fe}_{17}$  phase have been initiated by using C atoms. The introduced interstitial C and N atoms expand the lattice and Fe–Fe distance in the structure resulted in the enhancement of the exchange interaction between magnetic moments leading to a high Curie temperature. Among the interstitial modifications,  $\text{Sm}_2\text{Fe}_{17}\text{N}_x$  nitride has been found to have the highest solubility in comparison with the  $\text{Sm}_2\text{Fe}_{17}\text{C}_x$  carbide and  $\text{Sm}_2\text{Fe}_{17}\text{B}_x$  boride [3, 4]. The purpose of this study is to investigate the effects of boron additions at high ratios on magnetic properties of stoichiometric  $\text{Sm}_2\text{Fe}_{17}$  alloy.

## 2. Materials and equipments

The alloy ingots with nominal compositions of  $\text{Sm}_2\text{Fe}_{17}\text{B}_x$  ( $x = 0, 1, 2, 3, 4$ ) were prepared by arc-melting method under Ar atmosphere followed by heat treatment at 1000 °C for 2 h under vacuum. The samples were remelted three times for the homogeneity of the ingots. The changes in the structures of the samples were measured by X-ray diffraction (XRD) with Cu  $K_\alpha$  radiation at room temperature. The microstructures of the materials were characterized by scanning electron microscope (SEM). The Curie temperatures of the samples

were determined by AC magnetic susceptibility measurements by heating the samples up to 550 °C. Vibrating sample magnetometer (VSM) was used to measure the magnetic properties of the samples with a maximum magnetizing field of 10 kOe at room temperature.

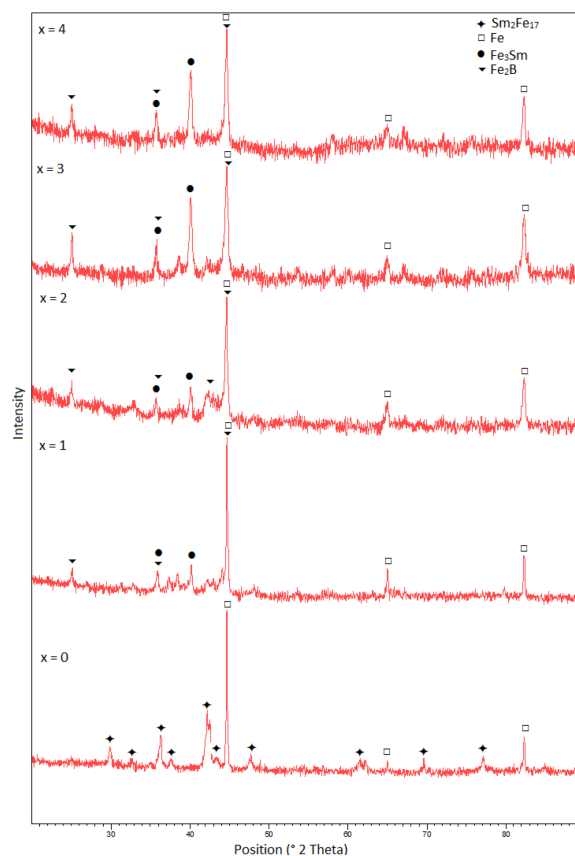


Fig. 1. Effect of boron addition on the crystal structure of  $\text{Sm}_2\text{Fe}_{17}\text{B}_x$  alloy.

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### 3. Results and discussion

Figure 1 shows the effect of the B addition on the  $\text{Sm}_2\text{Fe}_{17}$  composition, where the B concentration was increased from  $x = 0$  to  $x = 4$  in  $\text{Sm}_2\text{Fe}_{17}\text{B}_x$ . Diffraction peaks of the  $\text{Sm}_2\text{Fe}_{17}$  were determined to be soft magnetic cubic Fe and hard magnetic rhombohedral  $\text{Sm}_2\text{Fe}_{17}$  phases. With the addition of boron at  $x = 1$ , the diffraction peaks belonging to  $\text{Sm}_2\text{Fe}_{17}$  phase disappeared and the presence of the magnetic rhombohedral  $\text{SmFe}_3$  phase and tetragonal  $\text{Fe}_2\text{B}$  phase were observed. The peak intensity of the  $\text{SmFe}_3$  and  $\text{Fe}_2\text{B}$  phases increased significantly at  $x = 3$  and  $x = 4$ .

In SEM micrographs (Fig. 2), large grain size of about  $20\ \mu\text{m}$  was observed for the samples of  $\text{Sm}_2\text{Fe}_{17}$  and

$\text{Sm}_2\text{Fe}_{17}\text{B}$  showed in Fig. 2a and b. With the increase of boron ratio at  $x = 2$ , in some parts of the microstructure, there are lamellar formation exhibited in Fig. 2c. The amount of lamellar structure increased significantly in the microstructure with the further addition of boron at  $x = 3$  and  $x = 4$

In Fig. 3, the measured Curie temperature ( $T_c$ ) of the alloy at  $x = 0$  is about  $130^\circ\text{C}$ , corresponding to the Curie temperature of hard magnetic  $\text{Sm}_2\text{Fe}_{17}$  phase. With the addition of boron, the Curie temperature of the alloys increased to about  $361^\circ\text{C}$  at  $x = 1-4$  because of disappearance of the  $\text{Sm}_2\text{Fe}_{17}$  phase and formation of the ferromagnetic  $\text{Fe}_3\text{Sm}$  and  $\text{Fe}_2\text{B}$  phases detected in XRD measurement. The Curie temperatures of  $\text{Fe}_3\text{Sm}$  and  $\text{Fe}_2\text{B}$  phases are  $367^\circ\text{C}$  and  $742^\circ\text{C}$ , respectively [5, 6].

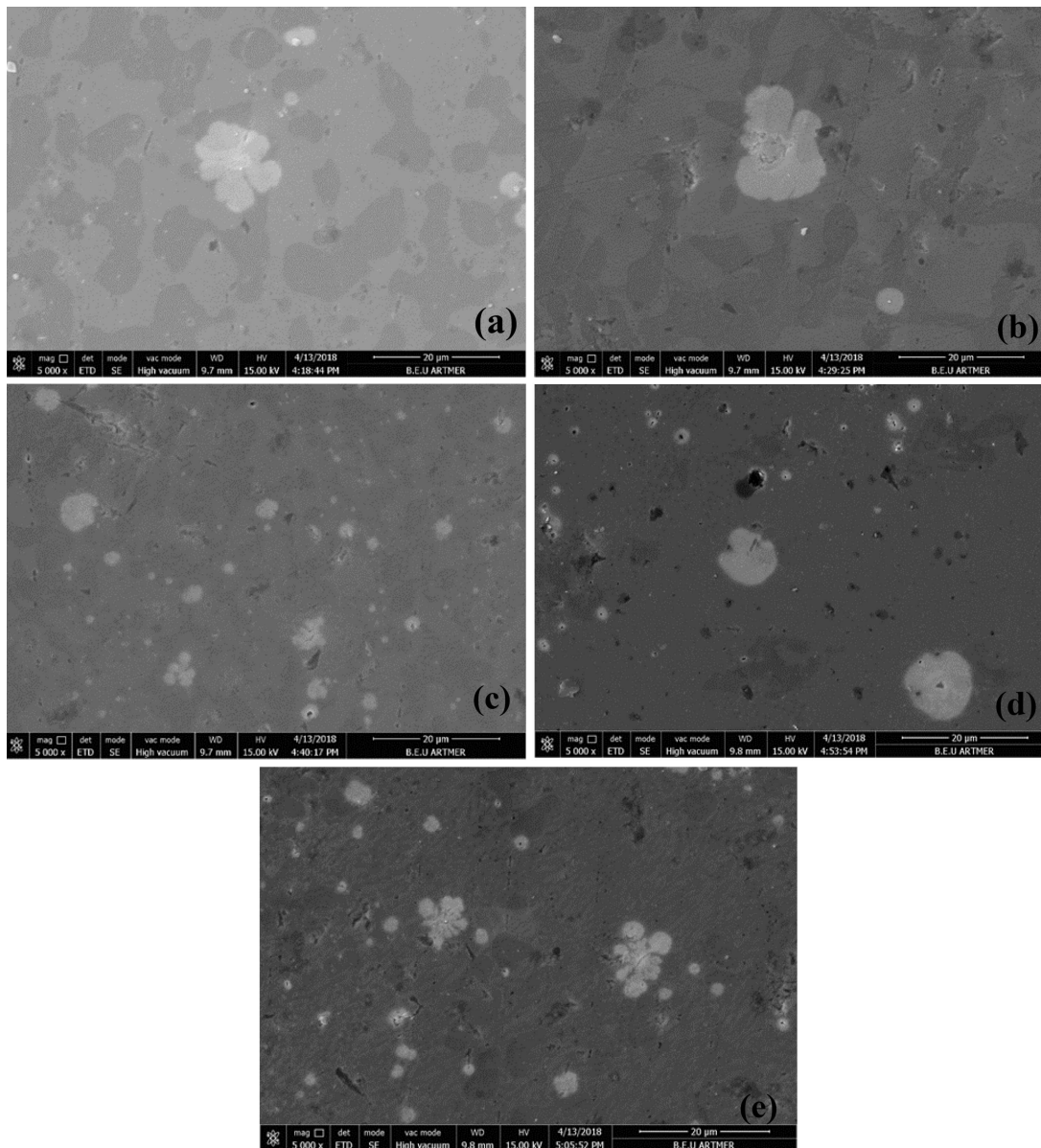


Fig. 2. SEM micrographs of the arc melted  $\text{Sm}_2\text{Fe}_{17}$  (a),  $\text{Sm}_2\text{Fe}_{17}\text{B}$  (b),  $\text{Sm}_2\text{Fe}_{17}\text{B}_2$  (c),  $\text{Sm}_2\text{Fe}_{17}\text{B}_3$  (d), and  $\text{Sm}_2\text{Fe}_{17}\text{B}_4$  (e) samples heat treated at  $1000^\circ\text{C}$  for 2 h.

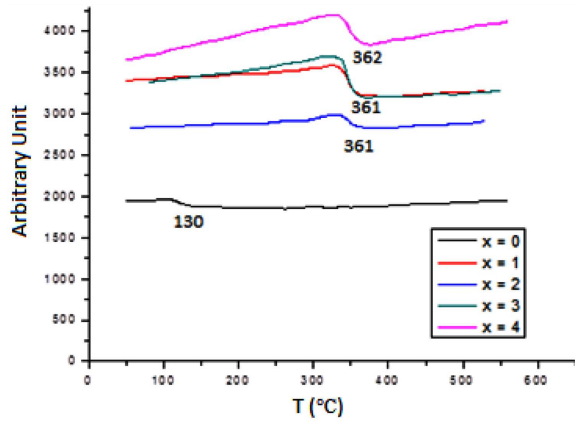


Fig. 3. The Curie temperature measurements of the samples with different boron concentrations.

Figure 4 indicates the hysteresis curves of the  $\text{Sm}_2\text{Fe}_{17}\text{B}_x$  ( $x = 0-4$ ). Although  $\text{Sm}_2\text{Fe}_{17}$  is hard magnetic, the shape of the curve for the sample at  $x = 0$  shows soft magnetic behaviour due to the considerable amount of iron (Fe) in the microstructure. For the boron added samples, the hysteresis curves are similar to that without boron but the saturation magnetizations decreased with the increase of boron ratio.

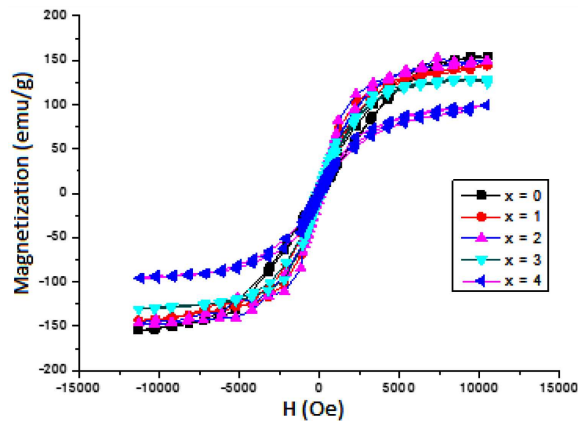


Fig. 4. Hysteresis curves of the samples.

In Table I the saturation magnetization  $M_s$  of the  $\text{Sm}_2\text{Fe}_{17}$  alloy has the highest value of 155.5 emu/g. With the addition of boron, the saturation magnetization decreased to its lowest value of 97.9 emu/g for the sample at  $x = 4$ . This may be due to the decrease in the amount of Fe in the microstructure and also due to the fact that  $\text{Sm}_2\text{Fe}_{17}$  has the higher saturation magnetization (117 emu/g) than that of  $\text{SmFe}_3$  phase (80.7 emu/g) [5].

TABLE I

Saturation magnetizations of the samples.

Alloy	$M_s$ [emu/g]
$\text{Sm}_2\text{Fe}_{17}$	155.5
$\text{Sm}_2\text{Fe}_{17}\text{B}$	145.4
$\text{Sm}_2\text{Fe}_{17}\text{B}_2$	150.9
$\text{Sm}_2\text{Fe}_{17}\text{B}_3$	127.8
$\text{Sm}_2\text{Fe}_{17}\text{B}_4$	97.9

#### 4. Conclusion

In XRD analysis,  $\text{Sm}_2\text{Fe}_{17}$  phase decompose to magnetic  $\text{SmFe}_3$  and  $\text{Fe}_2\text{B}$  phases, with the increase in boron ratio at  $x = 3$  and  $x = 4$  the intensity of the peaks diffracted from  $\text{SmFe}_3$  phase increased. The Curie temperature of the samples increased from 130 °C to 361 °C with the boron addition to the alloy, due to the existence of  $\text{SmFe}_3$  phase in the microstructure. The saturation magnetization has the highest value of 155.5 emu/g and decreases to its lowest value of 97.9 emu/g for the sample at  $x = 4$  because  $\text{Sm}_2\text{Fe}_{17}$  has the higher saturation magnetization (117 emu/g) than that of  $\text{SmFe}_3$  phase (80.7 emu/g) [5]. In the study, boron addition and heat treatment at 1000 °C resulted in an evolution of the phase constitution that caused changes in magnetic properties of the alloy instead of occupation of boron atoms in the interstitial sites of  $\text{Sm}_2\text{Fe}_{17}$  structure such as C and N.

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