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# Interaction of the Components in the $BaO-Tb_2O_{3+\delta}-CuO$ and Related Systems

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The isothermal section of the phase diagram of the BaO–Tb<sub>2</sub>O<sub>3+ $\delta$ </sub>–CuO system at 1173 K was constructed based on X-ray powder diffraction data. The existence of the compounds BaTbO<sub>3</sub> (SrZrO<sub>3</sub> structure type, *I4/mcm*, a = 0.6034(1), c = 0.8576(1) nm) and Ba<sub>44</sub>Cu<sub>45</sub>O<sub>90</sub> (own structure type, *Im*-3m, a = 1.8266(3) nm) was confirmed in the binary BaO–Tb<sub>2</sub>O<sub>3</sub> and BaO–CuO systems, respectively, but no ternary oxides were found. The investigation of samples with nominal composition A<sub>0.5</sub>Tb<sub>0.5</sub>TO<sub>3</sub> (A = Ca, Sr or Ba, T = Ni or Zn) showed the presence of phases from the corresponding binary systems, or unreacted reagents.

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### 1. Introduction

The A–R–Cu–O systems, where A is an alkaline-earth and R is a rare-earth metal, have attracted considerable interest for a long time. The compounds in these systems are usually characterized by layered structures, and are first of all interesting as high-temperature superconductors. The most prominent among them is undoubtedly YBa<sub>2</sub>Cu<sub>3</sub>O<sub>7</sub> (YBCO), the crystal structure and properties of which have been meticulously studied. The interaction of the components in the system Ba–Y–Cu–O and related systems with other alkaline-earth and rare-earth metals has also been extensively investigated [1]. In this work we focus on the system Ba–Tb–Cu–O and some related systems with Tb, for which data in the literature are scarce.

## 2. Experimental

Polycrystalline samples of the BaO–Tb<sub>2</sub>O<sub>3+ $\delta$ </sub>–CuO system were prepared by solid-state reaction of appropriate amounts of high-purity carbonates and oxides at 1173 K in air in two 24 h stages. Firstly, the reagents were mixed and heated in a corundum crucible to decompose the carbonates. Then, after cooling to room temperature, the mixtures were reground to achieve homogeneity, pressed into pellets and sintered. Samples of nominal composition A<sub>0.5</sub>Tb<sub>0.5</sub>TO<sub>3</sub>, where A = Ca, Sr or Ba, T = Ni or Zn, were prepared in a similar way. Phase and structure analyses were carried out on X-ray powder diffraction data (DRON 2.0M diffractometer, Fe  $K_{\alpha}$  radiation, DBWS program [2]).

# 3. Results and discussions

The isothermal section of the phase diagram of the BaO–Tb<sub>2</sub>O<sub>3+ $\delta$ </sub>–CuO system at 1173 K was constructed based on the investigation of 15 samples. Table I (at the end) contains the results of the phase analysis of some of



Fig. 1. Crystal structure of  $BaTbO_3$  (the coordination polyhedra of the Tb atoms are shown as shaded octahedra  $[O_6]$ ).

#### TABLE II

Number of quaternary compounds formed in Ba–R–Cu– O systems, where R is a rare-earth metal [1].

La	Ce	Pr	Nd	Pm	Sm	Eu	Gd
14	1	5	9	-	9	7	8
Y	Tb	Dy	Но	Er	Tm	Yb	Lu
13	1	6	8	6	5	8	3

the samples. The existence of the compounds BaTbO<sub>3</sub> (SrZrO<sub>3</sub> structure type, I4/mcm, a = 0.6034(1), c = 0.8576(1) nm,  $R_{\rm B} = 0.052$ ) and Ba<sub>44</sub>Cu<sub>45</sub>O<sub>90</sub> (own structure type, Im-3m, a = 1.8267(3) nm,  $R_{\rm B} = 0.183$ ) was confirmed in the binary BaO–Tb<sub>2</sub>O<sub>3</sub> and BaO–CuO boundary systems. The former adopts a tetragonal perovskite structure (Fig. 1), whereas the large cubic unit cell of the latter contains some 400 atoms. It was found that the interaction of the components in the ternary region of the BaO–Tb<sub>2</sub>O<sub>3+ $\delta$ </sub>–CuO system is characterized by the absence of compounds. The only compound reported in the literature, TbBa<sub>2</sub>Cu<sub>3</sub>O<sub>7</sub> with orthorhombic YBa<sub>2</sub>Cu<sub>3</sub>O<sub>7</sub> structure type [3], was consequently

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Fig. 2. Isothermal section of the phase diagram of the BaO–Tb<sub>2</sub>O<sub>3+ $\delta$ </sub>–CuO system at 1173 K (the samples used for the investigation are indicated as circles).

TABLE III

Results of the phase analysis of samples with nominal composition  $A_{0.5}Tb_{0.5}TO_3$ , where A = Ca, Sr or Ba, T = Ni or Zn.

Nominal composition	Phase	Structure type	Pearson symbol	Space group	Phase content [wt%]
Pasa The 7nOs	$BaTbO_3$	$SrZrO_3$	tI20	I4/mcm	59.9
Da0.5100.521103	ZnO	ZnO	hP4	$P6_3mc$	40.1
Sma The 7nOa	$\mathrm{SrTbO}_3$	$GdFeO_3$	oP20	Pnma	71.9
510.51 00.521103	ZnO	ZnO	hP4	$P6_3mc$	28.1
	ZnO	ZnO	hP4	$P6_3mc$	43.0
$\mathrm{Ca}_{0.5}\mathrm{Tb}_{0.5}\mathrm{ZnO}_3$	$\mathrm{Tb}_{7}\mathrm{O}_{12}$	$Pr_7O_{12}$	hR57	<i>R</i> -3	42.0
	CaO	NaCl	cF8	Pm- $3m$	15.0
	$\mathrm{Tb}_{7}\mathrm{O}_{12}$	$Pr_7O_{12}$	hR57	R-3	42.9
$\mathrm{Ca}_{0.5}\mathrm{Tb}_{0.5}\mathrm{NiO}_3$	NiO	NaCl	cF8	Pm- $3m$	41.1
	CaO	NaCl	cF8	Pm- $3m$	16.0

not observed under the conditions of our study. The BaTbO<sub>3</sub> phase is in equilibrium with the Ba<sub>44</sub>Cu<sub>45</sub>O<sub>90</sub> and CuO phases. Hence, there are three three-phase and two two-phase regions in the system (Fig. 2). It should be

noted that, during the synthesis, the initial cubic  $Tb_2O_3$  phase with  $(Mn_{0.5}Fe_{0.5})_2O_3$ -type structure was oxidized into the rhombohedral  $Tb_7O_{12}$  phase with  $Pr_7O_{12}$ -type structure.

Among the systems Ba–R–Cu–O, where R is a rareearth metal, the lowest numbers of compounds (Table II) have been reported in the systems with Ce (only  $CeBa_2Cu_3O_{7.4}$  [4]) and Tb (only  $TbBa_2Cu_3O_7$  [3]). Presumably this is connected with the possibility of these R elements to adopt oxidation states higher than 3+. Our investigation confirmed that the interaction of the components in the system Ba–Tb–Cu–O does not lead to the formation of a high number of compounds, as is the case for some related systems.

Among the selected samples from related systems with Tb (Table III), Ba<sub>0.5</sub>Tb<sub>0.5</sub>ZnO<sub>3</sub> and Sr<sub>0.5</sub>Tb<sub>0.5</sub>ZnO<sub>3</sub> were found to be two-phase samples, containing in equilibrium BaTbO<sub>3</sub> (SrZrO<sub>3</sub> structure type, I4/mcm, a = 0.6039(1), c = 0.8578(1) nm,  $R_{\rm B} = 0.099$ ), or SrTbO<sub>3</sub> (GdFeO<sub>3</sub> structure type, *Pnma*, a = 0.5945(1), b = 0.8347(2), c = 0.5879(1) nm,  $R_{\rm B} = 0.160$ ), and ZnO. The investigation of the samples with nominal compositions Ca<sub>0.5</sub>Tb<sub>0.5</sub>ZnO<sub>3</sub> and Ca<sub>0.5</sub>Tb<sub>0.5</sub>NiO<sub>3</sub> showed the presence of the initial compounds CaO, Tb<sub>7</sub>O<sub>12</sub> and ZnO (or NiO) in the expected ratios. Consequently, it can be concluded that the intensity of the interaction of the components decreases on going from the Ba- to the Ca-containing system.

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Results of the phase analysis of the quaternary samples used for the investigation of the BaO–Tb<sub>2</sub>O<sub>3+ $\delta$ </sub>–CuO system.

Nominal composition of the sample BaO:Tb <sub>2</sub> O <sub>3</sub> :CuO [mol.%]	Phase	Structure type	Pearson symbol	Space group	Phase content [wt%]
	BaTbO <sub>3</sub>	SrZrO <sub>3</sub>	<i>tI20</i>	I4/mcm	55.8
36.4:9.1:54.5	Ba44Cu45O90	Ba44Cu45O90	cI400	Im-3m	25.9
	CuO	CuO	mS8	C2/c	18.3
60.20.20	BaTbO <sub>3</sub>	SrZrO <sub>3</sub>	<i>tI</i> 20	I4/mcm	77.1
00:20:20	Ba44Cu45O90	Ba44Cu45O90	cI400	Im-3m	22.9
40-20-40	BaTbO <sub>3</sub>	SrZrO <sub>3</sub>	tI20	I4/mcm	77.9
40:20:40	CuO	CuO	mS8	C2/c	22.1
	BaTbO <sub>3</sub>	SrZrO <sub>3</sub>	tI20	I4/mcm	42.6
20:20:60	CuO	CuO	mS8	C2/c	36.1
	$\mathrm{Tb}_{7}\mathrm{O}_{12}$	Pr <sub>7</sub> O <sub>12</sub>	hR57	<i>R</i> -3	21.3
	BaTbO <sub>3</sub>	SrZrO <sub>3</sub>	<i>tI</i> 20	I4/mcm	63.2
40:40:20	Tb <sub>7</sub> O <sub>12</sub>	Pr <sub>7</sub> O <sub>12</sub>	hR57	<i>R</i> -3	28.1
	CuO	CuO	mS8	C2/c	8.7
	$Tb_7O_{12}$	Pr <sub>7</sub> O <sub>12</sub>	hR57	<i>R</i> -3	49.9
20:40:40	BaTbO <sub>3</sub>	SrZrO <sub>3</sub>	<i>tI</i> 20	I4/mcm	33.9
	CuO	CuO	mS8	C2/c	16.2
	Tb <sub>7</sub> O <sub>12</sub>	Pr <sub>7</sub> O <sub>12</sub>	hR57	<i>R</i> -3	58.0
20:60:20	BaTbO <sub>3</sub>	SrZrO <sub>3</sub>	<i>tI20</i>	I4/mcm	35.2
	CuO	CuO	mS8	C2/c	6.8