

# Bioactivity Properties and Characterization of Commercial Synthetic Hydroxyapatite – 5 wt.% Niobium (V) Oxide – 5 wt.% Magnesium Oxide Composite

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Hydroxyapatite is widely used for bone grafts due to its chemical and structural similarities to the mineral phase of hard tissues and due to its bioactivity and biocompatibility. However, hydroxyapatite has poor mechanical properties due to its brittleness. To improve mechanical properties of hydroxyapatite-ceramics, ceramic oxides, whiskers or fibers have been suggested. In this study, commercial synthetic hydroxyapatite composite reinforced with 5 wt.%  $\text{Nb}_2\text{O}_5$  and 5 wt.%  $\text{MgO}$  was characterized. Microstructural properties of all samples sintered at different temperatures were characterized using scanning electron microscopy technique. Phase analysis was carried out using X-ray diffraction technique. Mechanical properties were measured by compression and hardness tests. The bioactivity property was determined by in vitro bioactivity test. The best obtained values of density, compression strength and Vickers Microhardness were  $3.01 \text{ g/cm}^3$ , 96 MPa and 393 HV, for composite sintered at  $1300^\circ\text{C}$  for 4 hours. Bioactivity results for composite, sintered at  $1300^\circ\text{C}$ , show that apatite formation has started after two weeks in a simulated body fluid. At the end of the fourth week, the dense apatite layer and clusters were observed on the surface of the composite.

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## 1. Introduction

Hydroxyapatite ( $\text{HA}$ ,  $\text{Ca}_{10}(\text{PO}_4)_6(\text{OH})_2$ ) is the most popular bioceramic material used in orthopedical and dental applications, due to its excellent bioactivity and biocompatibility [1–3].  $\text{HA}$  can be obtained by both natural and synthetic methods. The synthetic  $\text{HA}$  is the most commonly used type of  $\text{HA}$ , however its fabrication is complex and expensive [2]. On the contrary, natural hydroxyapatite is inexpensive and recyclable [4]. The mechanical properties of  $\text{HA}$  should match those of the natural bone as closely as possible, to allow for proper bone remodeling at the implant site.  $\text{HA}$  is typically sintered to enhance the mechanical properties of  $\text{HA}$  bodies, but the mechanical properties of sintered  $\text{HA}$  implants are still inferior to those of natural bone [5, 6]. For this reason, there is a need for improving the mechanical properties of the  $\text{HA}$ -based bioceramics. Up to now the design of composite materials is one of the primary approaches to control the material properties [7].

In the literature there are very few studies about addition of niobium (V) oxide to  $\text{HA}$  ceramics. Tamai et al. and Fathi et al. have reported that niobium oxides were promoting the calcification process of human osteoblasts [8, 9]. Demirkol et al. have investigated the mechanical and microstructural properties of sheep hydroxyapatite – niobium oxide composites [10].

Magnesium was reported to stabilize  $\beta$  phase of TCP and to reduce risks of cardiovascular diseases. It is a

very important element in human body and controls biological functions. It is biocompatible, osteoconductive, osseointegrable and lightweight material with mechanical properties quite similar to those of bone tissue [11–13].

In this study, characterization and bioactivity properties of  $\text{HA}$  composite with addition of 5 wt.% of niobium (V) oxide and 5 wt.% of magnesia, sintered at different temperatures, were investigated.

## 2. Materials and methods

Commercial synthetic hydroxyapatite (CSHA-Acros Org., BE) was used as the major material. Niobium (V) oxide (Alfa Aesar Comp.) and magnesia ( $\text{MgO}$ ) were selected as additives. The CSHA – 5 wt.%  $\text{Nb}_2\text{O}_5$  – 5 wt.%  $\text{MgO}$  powders were mixed together for 4 h using a ball mill. This mixture (CSHA-5N5M) was pressed at 350 MPa and the pressed samples were sintered at 1000, 1100, 1200,  $1300^\circ\text{C}$  for 4 h ( $+5^\circ\text{C min}^{-1}$ ). X-ray diffraction (XRD) analysis was carried out using a Brucker D8-Advanced X-ray diffractometer with  $\text{Cu K}_\alpha$  radiation. Scanning electron microscopy (SEM) (Jeol JSM-5910 LV – Low Vacuum Scanning) and energy dispersive spectroscopy (EDX) (Oxford Inca Energy 200) were used to characterize the microstructure of the composites. Density, Vickers microhardness, and compression strength tests were performed. The hardness measurements of the samples were done using Vickers microhardness testing system (HMV Shimadzu, JP), 200 g load for 15 s. The compression tests were performed using Universal Testing machine (Shimadzu, JP) at a crosshead speed of 3 mm/min. In vitro bioactivity test of composite was done with Kokubo SBF solution during soaking time of 1–4 weeks at pH of 7.4.

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### 3. Results and discussion

Table I summarizes density and mechanical properties of CSHA-5N5M composite sintered at different temperatures. The best properties were obtained in the composite sample sintered at 1300 °C.

TABLE I

Density, compression strength and Vickers microhardness of CSHA-5N5M composite sintered at different temperatures.

Temperature [°C]	Density [g/cm <sup>3</sup> ]	Compression strength [MPa]	Vickers microhardness [HV]
1000	2.47 ± 0.07	47 ± 0.09	97 ± 1.3
1100	2.69 ± 0.03	61 ± 0.03	140 ± 0.9
1200	2.83 ± 0.05	78 ± 0.08	225 ± 1.1
1300	3.01 ± 0.07	96 ± 0.05	393 ± 0.9

Density and mechanical properties of composite increased with increasing sintering temperature. In former studies, Demirkol et al. studied density and mechanical properties of CSHA [11]. Density, compression strength and Vickers microhardness of CSHA [14] increase with the simultaneous addition of 5 wt.% Nb<sub>2</sub>O<sub>5</sub> and 5 wt.% MgO to the CSHA by 20%, 50% and 140%, respectively.

Simultaneous addition of niobium (V) oxide and magnesium is more useful than adding them separately, according to physical and mechanical properties [14].

Table II summarizes product phases obtained in XRD diagrams for CSHA-5N5M composite sintered at 1000–1300 °C. Composites sintered at both 1000 °C and 1300 °C include hydroxyapatite (HA), whitlockite (beta-tricalcium phosphate) (W) and calcium magnesium phosphate (CaMgPO<sub>4</sub>). CSHA-5N5M composite sintered at 1300 °C has shown niobium magnesium phosphate (NbMgPO<sub>4</sub>) phase in addition to these phases. This phase is very effective for improving the mechanical properties of the composite.

TABLE II

Phases obtained from XRD analysis of CSHA with addition of 5 wt.% Nb<sub>2</sub>O<sub>5</sub> and 5 wt.% MgO composite sintered at 1000 and 1300 °C.

	1000 °C	1300 °C
CSHA-5N5M	Ca <sub>10</sub> (PO <sub>4</sub> ) <sub>6</sub> (OH) <sub>2</sub> (HA) Nb <sub>2</sub> O <sub>5</sub> Ca <sub>3</sub> (PO <sub>4</sub> ) <sub>2</sub> (W) MgO CaMgPO <sub>4</sub>	Ca <sub>10</sub> (PO <sub>4</sub> ) <sub>6</sub> (OH) <sub>2</sub> Ca <sub>3</sub> (PO <sub>4</sub> ) <sub>2</sub> CaMgPO <sub>4</sub> NbMgPO <sub>4</sub>

Figure 1 presents the micrographs of composites sintered at 1000 and 1300 °C. When Fig. 1a is compared with Fig. 1b, it is seen that grain growth has occurred and porosity of composite has decreased with the increase of sintering temperature.

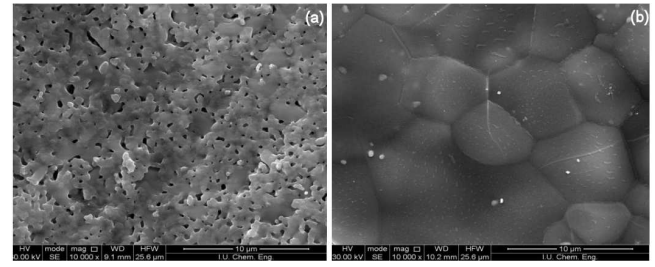


Fig. 1. Microstructures of composites sintered at different temperatures (×10000), (a) CSHA-5N5M at 1000 °C, (b) CSHA-5N5M at 1300 °C.

In vitro bioactivity test results of CSHA – 5 wt.% Nb<sub>2</sub>O<sub>5</sub> – 5 wt.% MgO composite show that simultaneous addition of Nb<sub>2</sub>O<sub>5</sub> and MgO to HA, as reinforcement materials, has a positive effect on formation of apatite. This composite shows a good apatite formation in simulated body fluid (SBF), which indicated bioactivity. Figure 2 shows the TF-FTIR spectrum of apatite coating on the CSHA-5N5M composite sintered at 1300 °C, after the bioactivity test. O–H strain mod band and vibration mod peak were positioned at 3354 cm<sup>−1</sup> and 1657 cm<sup>−1</sup>, respectively. Bands at 1041, 598 and 557 cm<sup>−1</sup> correspond to PO<sub>4</sub><sup>3−</sup>/HPO<sub>4</sub><sup>2−</sup> groups. Bands of carbonate groups are positioned at 1415, 1452 and 872 cm<sup>−1</sup>. This figure shows that the solid formed in the SBF is the hydroxycarbonapatite.

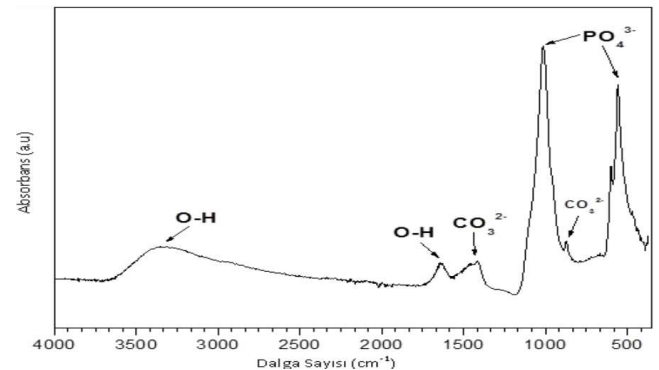


Fig. 2. TF-FTIR spectrum of apatite coating formed during the bioactivity test of CSHA-5N5M composite, sintered at 1300 °C.

### 4. Conclusions

The findings of this study are summarized as follows:

Mean density values and mechanical properties of CSHA-5N5M composite increase with increasing sintering temperature. Using Nb (V) oxide and magnesium oxide together, as the reinforcement materials for improvement of mechanical properties, is better than using them separately.

The highest density, compression strength and Vickers microhardness values were obtained for composite sintered at 1300 °C.

This material is a good candidate for orthopedical applications.

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