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Electrical Transport Properties of $\text{Yb}_{8-x}\text{Y}_x\text{V}_2\text{O}_{17}$ ($x = 0, 2, 8$)

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The UV-vis-NIR and electrical properties of $\text{Yb}_{8-x}\text{Y}_x\text{V}_2\text{O}_{17}$ for $x = 0, 2$, and 8 were investigated. The band energy gap of 2.6 eV determined for $\text{Yb}_6\text{Y}_2\text{V}_2\text{O}_{17}$ ($x = 2$) and comparable for the remaining compounds with $x = 0$ and 8 is characteristic for insulators. Low electrical conductivity with a characteristic minimum shifting to higher temperatures from 322 , via 360 to 370 K in the sequence $x = 0, 2$ and 8 , which decreases with increasing content of ytterbium was observed. Temperature dependence of thermoelectric power showed n - p transition at 410 and 467 K for $x = 0$ and 2 , respectively, and n -type conductivity for $x = 8$, indicating mainly n -type electrical conductivity. A breakdown voltage of 26 V/mm is mainly observed for the I - V characteristics at 400 K and showing a varistor-like behavior.

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1. Introduction

Complex oxide systems based on rare earth metal oxides are of great interest for creation of new materials for production e.g. of plasma displays, electroluminescent diodes, or fluorescence lamps [1–5].

Preliminary studies of the ternary oxides V_2O_5 - Yb_2O_3 - Y_2O_3 system have shown that, in one of its sections, i.e. in the system $\text{Yb}_8\text{V}_2\text{O}_{17}$ - $\text{Y}_8\text{V}_2\text{O}_{17}$, a previously unknown phase of the formula $\text{Yb}_6\text{Y}_2\text{V}_2\text{O}_{17}$ is formed [6]. The new phase $\text{Yb}_6\text{Y}_2\text{V}_2\text{O}_{17}$ was obtained by heating a mixture of the three oxides, i.e. Yb_2O_3 , V_2O_5 and Y_2O_3 in a molar ratio (3:1:1) in air, between temperatures 873 and 1823 K [6]. The X-ray powder diffractogram showed that $\text{Yb}_6\text{Y}_2\text{V}_2\text{O}_{17}$ is isostructural with $\text{Yb}_8\text{V}_2\text{O}_{17}$ which crystallizes in the triclinic systems [7]. This result is contrary to the literature data [8]. According to the results of our research the elementary cell parameters of new phase $\text{Yb}_6\text{Y}_2\text{V}_2\text{O}_{17}$ which were refined with the use of the program REFINEMENT are as follows: $a = 0.8972$ nm, $b = 0.9292$ nm, $c = 0.9824$ nm, $\alpha = 77.668^\circ$, $\beta = 106.301^\circ$, $\gamma = 116.291^\circ$ [7].

In the present study the electrical and optical properties of $\text{Yb}_{8-x}\text{Y}_x\text{V}_2\text{O}_{17}$ for $x = 0$ and 8 [7–9] as well as for a new phase ($x = 2$) were carried out.

2. Experimental details

The powder diffraction patterns of the samples obtained were recorded with the aid of the diffractometer EMPYREAN II (PANalytical, Netherlands) using $\text{Cu } K_\alpha$ with graphite monochromator. Ultraviolet-visible and near-infrared (UV-vis-NIR) diffuse reflectance spectra were recorded at room temperature and in the wavelength range of 200 – 900 nm using a JASCO-V670

spectrophotometer equipped with an integrating sphere. In order to determine the energy gap, E_g , these spectra were transformed using the Kubelka-Munk function: $F(R) = (1 - R)^2/2R$, where R is reflectancy [%] [10, 11]. The sample morphology was observed on SEM images. Scanning electron microscopy study was performed on JSM-1600, JEOL, Japan with an X-ray energy dispersive analysis — EDX (ISIS-300, Oxford).

The electrical conductivity $\sigma(T)$ and the I - V characteristics have been measured with the aid of the DC method in the temperature range 300 – 400 K using a KEITHLEY 6517B Electrometer/High Resistance Meter. The thermoelectric power $S(T)$ was measured in the temperature range 300 – 600 K with the aid of a Seebeck Effect Measurement System (MMR Technologies, Inc., USA). For the electrical measurements, the powder samples were compacted in a disc form (10 mm in diameter and 2 mm thick) using a pressure of 1.5 GPa and then they were sintered during 2 h at 1073 K. The electrical and thermal contacts were made by a silver lacquer mixture (Degussa Leitsilber 200).

3. Results and discussion

Figure 1 shows a fragments of diffraction pattern (XRD) of $\text{Yb}_8\text{V}_2\text{O}_{17}$ (Fig. 1a), $\text{Yb}_6\text{Y}_2\text{V}_2\text{O}_{17}$ (Fig. 1b) and $\text{Y}_8\text{V}_2\text{O}_{17}$ (Fig. 1c) obtained in the context of this paper, which are almost identical to that shown in [7].

The SEM image of new polycrystalline phase is presented in Fig. 2.

The morphology of $\text{Yb}_{8-x}\text{Y}_x\text{V}_2\text{O}_{17}$ for $x = 2$ ($\text{Yb}_6\text{Y}_2\text{V}_2\text{O}_{17}$) crystallites is very similar to those of the matrix i.e. $\text{Yb}_8\text{V}_2\text{O}_{17}$ [7]. They look like polygons of irregular shapes and different sizes, varying from 0.5 to 9 μm (Fig. 2).

The UV-vis-NIR measurements and the Kubelka-Munk transformation [10,11] (Fig. 3) have shown that all phases of $\text{Yb}_{8-x}\text{Y}_x\text{V}_2\text{O}_{17}$ are insulators for the energy gap of 2.6 eV for $\text{Yb}_6\text{Y}_2\text{V}_2\text{O}_{17}$ ($x = 2$), whose value is comparable to that of other samples ($x = 0$ and 8).

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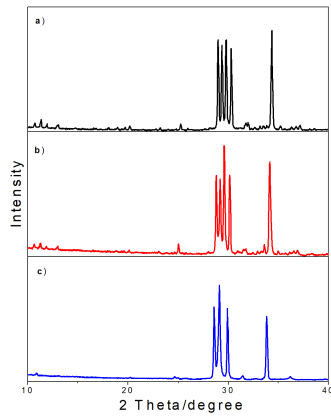


Fig. 1. Fragments of XRD powder patterns of the: a) $\text{Yb}_8\text{V}_2\text{O}_{17}$, b) $\text{Yb}_6\text{Y}_2\text{V}_2\text{O}_{17}$, c) $\text{Y}_8\text{V}_2\text{O}_{17}$.

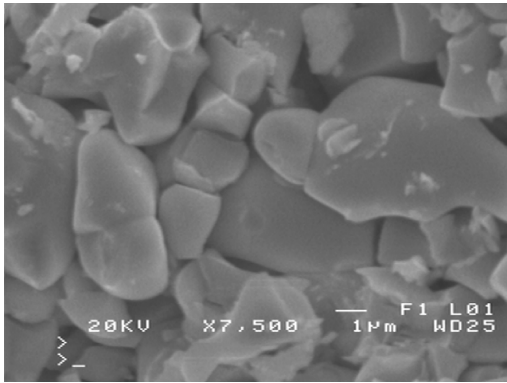


Fig. 2. SEM image of $\text{Yb}_6\text{Y}_2\text{V}_2\text{O}_{17}$.

The compounds under study showed low electrical conductivity, $\sigma(T)$, with a characteristic minimum shifting to higher temperatures from 322, via 360 to 370 K in the sequence $x = 0, 2$ and 8 (Fig.4). It is noteworthy that the electrical conductivity decreases with increasing content of ytterbium. The temperature dependence of thermo-

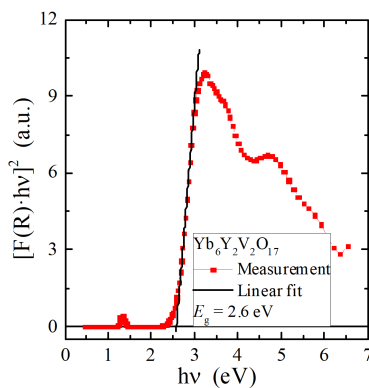


Fig. 3. Plots of $[F(R)h\nu]^2$ vs. the energy of the incident photon $h\nu$ for $\text{Yb}_6\text{Y}_2\text{V}_2\text{O}_{17}$. E_g is the band gap energy.

power, $S(T)$, showed $n-p$ transition at 410 and 467 K for $x = 0$ and 2 , respectively, and n -type conductivity for $x = 8$, indicating mainly n -type electrical conductivity (Fig. 5) probably due to predominant contribution of oxygen vacancies. In this case, the thermoelectric power increases as the content of ytterbium increases.

TABLE I

The voltage at the point of breakthrough V [V] and breakdown voltage V_b [V/mm] of $\text{Yb}_{8-x}\text{Y}_x\text{V}_2\text{O}_{17}$ for the specimen thickness of 2 mm at 300 K and 400 K.

x	Compound	V	V_b	V	V_b
		300 K		400 K	
0	$\text{Yb}_8\text{V}_2\text{O}_{17}$	–	–	60	30
2	$\text{Yb}_6\text{Y}_2\text{V}_2\text{O}_{17}$	92	46	44	22
8	$\text{Y}_8\text{V}_2\text{O}_{17}$	–	–	54	27

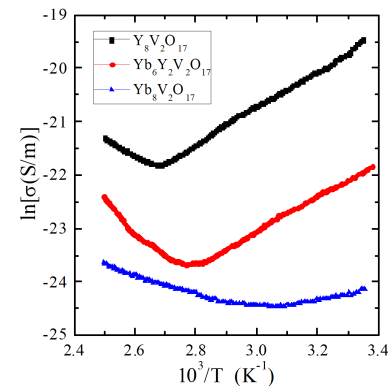


Fig. 4. The electrical conductivity ($\ln \sigma$) vs. reciprocal temperature T^{-1} for $\text{Yb}_{8-x}\text{Y}_x\text{V}_2\text{O}_{17}$ ($x = 0, 2$ and 8).

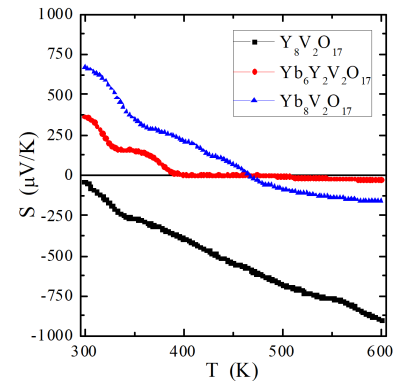


Fig. 5. The thermoelectric power S vs. temperature T for $\text{Yb}_{8-x}\text{Y}_x\text{V}_2\text{O}_{17}$ ($x = 0, 2$ and 8).

The most interesting observation concerns the non-linear $I - V$ characteristics (Figs. 6–8) for all the phases under study, similar to back-to-back Zener diodes but with much greater current and energy handling capabilities [12]. A breakdown voltage $V_b \sim 26$ V/mm (Table I) is mainly observed for the $I - V$ characteristics at 400 K and it is almost twice lower than the value of V_b at 280

and 300 K, showing a varistor-like behavior. From the physics research of varistors follows that the non-linearity $I-V$ characteristics in this case may be due to the phenomenon of grain-boundary where a barrier to majority charge carriers exists in the depletion layers of the adjacent grains [12].

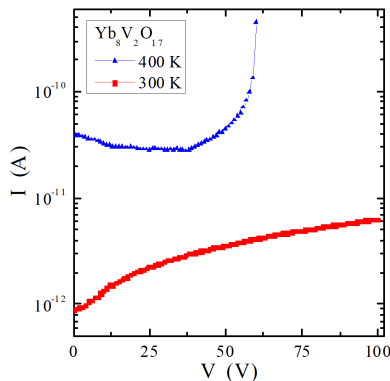


Fig. 6. The $I-V$ characteristics at 300 and 400 K for $\text{Yb}_8\text{V}_2\text{O}_{17}$.

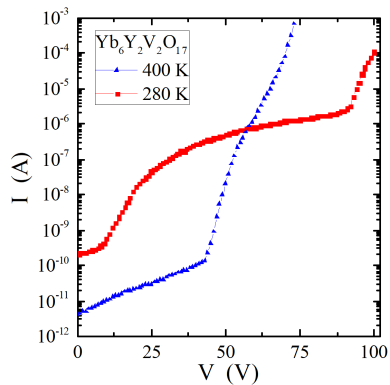


Fig. 7. The $I-V$ characteristics at 280 and 400 K for $\text{Yb}_6\text{Y}_2\text{V}_2\text{O}_{17}$.

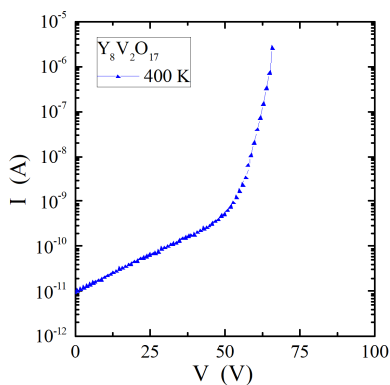


Fig. 8. The $I-V$ characteristics and 400 K for $\text{Y}_8\text{V}_2\text{O}_{17}$.

The studied materials can be used as the voltage stabilizers, the electronic power systems and the transient surge suppression in electronic circuits.

4. Conclusions

We have investigated the UV-vis-NIR spectroscopy and electrical properties of $\text{Yb}_{8-x}\text{Y}_x\text{V}_2\text{O}_{17}$ phases for $x = 0, 2$ and 8. All samples turned out to be insulators with the band energy gap of 2.6 eV, low electrical conductivity mainly of n -type with a characteristic minimum shifting to higher temperatures in the sequence $x = 0, 2$ and 8 which decreases with increasing content of ytterbium and the breakdown voltage of 26 V/mm which is mainly observed for the $I-V$ characteristics at 400 K and showing a varistor-like behavior. The solid solutions under study can be considered as varistors and for applications as the electronic power systems

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