Thin Layers XRD Study Technique on an Example of Cobalt Tetrafluoro Phthalocyanine

A.S. Sukhikh\textsuperscript{a,b,*}, T.V. Basova\textsuperscript{a,b} and S.A. Gromilov\textsuperscript{a,b}

\textsuperscript{a}Nikolaev Institute of Inorganic Chemistry SB RAS, Lavrentiev Pr. 3, Novosibirsk 630090, Russia
\textsuperscript{b}Novosibirsk State University, Pirogova Str. 2, Novosibirsk, Russia

Thin layers X-ray diffraction study technique utilizing single-crystal X-ray diffractometer equipped with micro-focus X-ray tube is described. It is shown that the layers of the tetra-fluorinated cobalt phthalocyanine (CoPcF\textsubscript{4}), deposited by thermal evaporation in vacuum on a polished surface of the substrate (glass, quartz), have a highly oriented polycrystalline structure. All the crystallites have the (00l) plane oriented along the surface of the substrate. CoPcF\textsubscript{4} X-ray diffraction pattern indexing was conducted and unit cell parameters were determined. It is shown that crystal phase of both polycrystalline powder and thin layers of CoPcF\textsubscript{4} are isostructural to that of \(\alpha\)-CoPc.

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1. Introduction

Currently, metal phthalocyanines cause an increased scientific interest, because they are considered as promising materials for use in various electronic components such as thin-film diodes and organic field effect transistors [1–3], solar cells [4, 5] and gas sensors [6–8]. All these electronic devices contain metal phthalocyanines in form of a thin layers (a few tens of nm thick) deposited by thermal evaporation under vacuum or by molecular beam epitaxy on a pre-polished and prepared substrate. X-ray diffraction (XRD) analysis of such thin samples is possible in grazing incidence X-ray diffraction geometry (GIXD) [9] using the synchrotron radiation source or a high-power X-ray tube with a rotating anode [10]. In Ref. [11] a study of 100 nm cobalt phthalocyanine layer by the means of powder diffractometer equipped with a special attachment (0.4° long Soller slits) is described.

Cobalt phthalocyanine (CoPc) and hexadeca-fluorinated cobalt phthalocyanine (CoPcF\textsubscript{16}) are both sufficiently well described in the literature [12–14], while CoPcF\textsubscript{4} is not mentioned at all. In the present paper we identify unit cell parameters and describe XRD technique for studying thin layers of CoPcF\textsubscript{4} on serial single-crystal diffractometer with a microfocus tube and 2D CCD-detector.

2. Experimental setup

CoPcF\textsubscript{4} was synthesized by heating a 4:1 mixture of 4-fluorophthalonitrile (Aldrich) and anhydrous cobalt(II) chloride in a glass tube at 220°C during 6 h. After cooling to room temperature the mixture was washed with ethanol and acetone to remove organic impurities. The precipitate was purified by gradient sublimation in vacuum (10\textsuperscript{-5} Torr) at 450°C. CoPcF\textsubscript{4} was prepared as a statistical mixture of four regioisomers due to the various possible positions of fluorine substituents. No attempt was made to separate the CoPcF\textsubscript{4} isomers.

Thin CoPcF\textsubscript{4} layers were obtained by organic molecular beam deposition (OMBD) technique (VUP-5M vacuum station, 10\textsuperscript{-5} Torr, Knudsen cell source at 460°C). 10 \times 10 mm\textsuperscript{3} glass plates (0.1 mm thick) and polished ends of fused quartz rods (1 mm in diameter) were used as substrates. During the thin layer growth the substrates were held at room temperature, at a distance of 50 mm from the source with the surfaces perpendicular to the source. Deposition time was in range of 30–60 min. The samples obtained were solid transparent blue layers with estimated thickness of 60–80 nm.

XRD measurements were carried out using conventional Bragg–Brentano geometry (Shimadzu XRD-7000 powder X-ray diffractometer, Cu K\textsubscript{α}, Bragg–Brentano geometry with \(\theta-\theta\) goniometer, scintillation counter) and 2D GIXD geometry [15–18] using Bruker DUO single-crystal diffractometer (\(\lambda = 1.5418 \text{ Å}, 0.6 \text{ mm output collimator,} \) 1024 \times 1024 CCD detector, 60 \times 60 \text{ μm}\textsuperscript{2} physical pixel size), equipped with an Incoatec IP\textsubscript{Cu} microfocus X-ray tube (Cu K\textsubscript{α}, 45 kV, 0.64 mA).

*corresponding author; e-mail: a.sukhikh@niic.nsc.ru

Fig. 1. Experimental setup for studying thin layers in GIXD geometry on the Bruker DUO single-crystal diffractometer.
The general scheme of the measurements is shown in Fig. 1. 4-circle goniometer allows for: turning the sample around its own axis (\(\varphi\)) and around primary beam (\(\chi\)), changing primary beam incidence angle (\(\theta_{\text{sample}}\)), changing detector position (\(\theta_{\text{det}}\)) and the distance from the center of the sample to the detector (\(L, OA\) segment on picture). The sample on the polished end of the fused quartz rod was directly mounted into the goniometer head. Alignment was performed using a standard video camera. The angle between the primary beam and detector (\(\theta\)) was 10°. The distance from the center of the sample to the detector (\(L\)) was varied in range of 40 ± 160 mm, and \(\theta_{\text{sample}}\) angle — in range of 0 ± 3° (0.5° step). Measurements carried out at constant values of the angle \(\varphi\), in the range 0 ± 180° (30° step), showed that the general view of the observed diffraction patterns remains constant.

3. Results and discussion

Figure 2 shows XRD patterns for CoPcF\(_4\) in comparison with the theoretical diffraction data for \(\alpha\)-CoPc and \(\beta\)-CoPc. The last ones were calculated according to the known structural data: \(a = 12.090(8), b = 3.754(2), c = 12.800(9)\) \(\AA\), \(\alpha = 88.96(6), \beta = 90.97(6), \gamma = 95.09(7)\), \(V = 578.48\) \(\AA^3\), \(P\bar{1}\) for \(\alpha\)-CoPc [19] and \(a = 14.495(5), b = 4.742(4), c = 19.107(5)\) \(\AA\), \(\alpha = 90, \beta = 120.76(2), \gamma = 90\), \(V = 1128.56\) \(\AA^3\), \(P\bar{2}1/c\) for \(\beta\)-CoPc [20]. Powder Cell 2.5 [21] software was used for calculations. Comparison of diffraction patterns for \(\alpha\)-CoPc and CoPcF\(_4\) (Fig. 2a,c) has allowed us to conclude that these phases are isostructural. Positions of the 19 most intensive CoPcF\(_4\) single diffraction peaks were determined using Origin9.1. By analogy with the corresponding \(\alpha\)-CoPc diffraction peaks, \(hkl\) indexes were assigned to the 11 of 19 CoPcF\(_4\) diffraction peaks and then preliminary unit cell parameters were calculated by the least squares method. Based on the preliminary results, indexing and refinement of all 19 diffraction peaks were carried out, yielding the final parameters: \(a = 12.41(5), b = 3.67(1), c = 13.35(6)\) \(\AA\), \(\alpha = 88.4(1), \beta = 89.00(8), \gamma = 96.0(1)\), \(V = 605(1)\) \(\AA^3\), \(P\bar{1}\).

The XRD pattern of CoPcF\(_4\) thin layers deposited on glass plates is shown in Fig. 2d. Only the single diffraction reflex is present, indicating that the crystallites are ideally oriented relative to the substrate plane. However because of that, it is impossible to assert unequivocally that the resulting layer of CoPcF\(_4\) has the same crystalline phase as polycrystalline CoPcF\(_4\). It is known [22, 23] that CoPc can be deposited on the substrate in two possible crystal phases — metastable \(\alpha\)-phase (Fig. 2a) and stable \(\beta\)-phase (Fig. 2b), and by heating the metastable \(\alpha\)-phase gradually turns into the \(\beta\)-phase. Thus, in our case a thin layer of CoPcF\(_4\), obtained by heating polycrystalline powder to 460°C may contain two crystalline phases in varying proportion. Attempts to find additional diffraction peaks for unambiguous identification by increasing the sampling time were not successful, so further research was carried out in the 2D GIXD geometry.

![Fig. 2. Comparison of theoretical CoPc diffraction patterns and experimental CoPcF\(_4\) diffraction patterns (Shimadzu XRD-7000, Cu K\(_{\alpha}\)).](image)

In all cases, local diffraction spots typical of single-crystal sample were observed (see Fig. 3). By turning the sample relative to the primary beam at a fixed angle \(\chi\) corresponding rotation of the diffraction pattern was observed. (001) reflex is the most intensive; the other pairs of reflexes have a relative intensity of 2% and about 1.5 times higher than the background level. Each crystal plane corresponds to a pair of symmetrically arranged reflexes: 2,3 — (101), (–101); 4,5 — (102); 6,7 — (–201). Positions of the diffraction peaks centers and angles between the corresponding planes and the sub-

![Fig. 3. The diffraction pattern of the CoPcF\(_4\) sample obtained by 2D GIXD (\(\chi = 60°, L = 40\) mm, \(\theta_{\text{det}} = 10°\)).](image)
strate surface were determined using Origin9.1 (Gauss2D function). After switching to the interplanar distances the following results were obtained: (001) — 13.11 Å, 0°; (101), (–101) — 9.18 Å, 44.5°; (102) — 5.87 Å, 27°; (–201) — 5.58 Å, 63.5°, which are fully consistent with unit cell parameters given earlier.

4. Conclusions

It is shown that CoPcF\textsubscript{4} layers deposited by OMBD on a polished surface of the substrate (glass, quartz) have highly oriented polycrystalline structure. When studying these layers in the Bragg–Brentano geometry, only the orders of reflections from one crystallographic plane are available, as demonstrated in Fig. 2d. This makes it difficult to radiographically identify the crystal phase. Using the 2D GIXD, we have shown that thin layer samples of CoPcF\textsubscript{4} are isostructural to \(\alpha\)-CoPc. In addition, 2D GIXD allows us to estimate the angles between the substrate surface and crystal planes for which the corresponding reflexes can be clearly observed on diffraction pattern. Because all of the crystallites have (001) plane parallel to the substrate plane, then the angle between the (001) plane and any other plane is equal to the angle between this plane and the substrate surface. Such information may be useful for diffraction pattern indexing when the unit cell parameters are unknown.

Acknowledgments

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References