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Comparative Study of Sb_2O_3 (Sb_2O_5) and Ta_2O_5 Doping Effects with TeO₂ on Electrical Properties of δ -Bi₂O₃

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In this study, Sb₂O₃ (Sb₂O₅) and Ta₂O₅ are used as co-dopants with TeO₂ to stabilize the delta phase of bismuth oxide (δ -Bi₂O₃). Some compositions with formula (1-x) BiO_{1.5}-(x/4) Sb₂Te₂O₉ and (1-x) BiO_{1.5}-(x/4) Ta₂Te₂O₉ (x = 0.1, 0.2, 0.3, 0.6, and 0.9) have been synthesized by solid state reaction at 850 °C and characterized by powder X-ray diffraction. The Bi_{0.9}Sb_{0.05}Te_{0.05}O_{1.575}, Bi_{0.9}Ta_{0.05}O_{1.575} and Bi_{0.8}Ta_{0.1}Te_{0.1}O_{1.65} retain a cubic fluorite structure of δ -Bi₂O₃ phase. The electric properties were studied by impedance spectroscopy. All samples were evaluated by calculating conductivities and activation energies. Various impedance model including constant phase element and the Warburg impedances have been used to interpret the Nyquist representations of electrical analyses.

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1. Introduction

Fluorite based materials with the type fcc structure $(Fm\bar{3}m)$ have been considered as potential materials for solid oxide fuel cell electrolyte. The delta phase of bismuth oxide, δ -Bi₂O₃ has high conductivity due to high mobile oxygen vacancies in random distribution. The high oxygen mobility is attributed to the weak Bi-O bond and the high polarizability of Bi^{3+} with its lone pair $6s^2$ electrons [1]. It has a conductivity of 1 S cm⁻¹ at 750 °C [2]. Therefore, this oxide has been explored for various potential applications in gas sensors [3], photovoltaic cells [4], supercapacitors [5] and photocatalysts [6]. Furthermore, it is an extremely promising solid electrolyte material and can be used as solid oxide fuel cells and oxygen sensors. However, it is stable only between 730 °C and its melting point of 825 °C. Therefore, it cannot be preserved at room temperature.

By using doping method with some rare earth [7, 8] and transition metal (V, Nb, Ta and W) oxides [9, 10], the fluorite type can be preserved to ambient temperature. It is also possible to use combination of metal oxide, so-called double doping strategy. Examples include the ternary systems $Bi_2O_3-Er_2O_3-WO_3$ [11], $Bi_2O_3 Er_2O_3-PbO$ [12], $Bi_2O_3-Er_2O_3-Wb_2O_5$ [13]. Mercurio et al. [14] investigated $Bi_2O_3-Ln_2O_3-TeO_2$ (Ln = La, Sm, Gd, and Er) ternary system and announced that $(Bi_2O_3)_{0.90}(La_2O_3)_{0.06}(TeO_2)_{0.04}$ had a higher ionic conductivity, of 5×10^{-3} S cm⁻¹ at 350 °C. We have recently reported the study of electrical properties of Bi_2O_3 based on the fluorite type phase in the Bi_2O_3 -TeO₂-Nb₂O₅ system.

The dopants have been chosen in accordance with several factors. The oxides TeO_2 and Sb_2O_3 (Sb_2O_5) were selected because Te^{4+} and Sb^{3+} present a lone pair of electrons stereochemically active and promotes a very high polarizability like Bi^{3+} . Concerning Ta_2O_5 , it has been introduced (5–10 mol.% Ta_2O_5) into Bi_2O_3 to stabilize δ phase [15]. The composition synthesized with $7.5 \text{ mol.}\% \text{ Ta}_2 \text{O}_5$ exhibits satisfying electrical results with a conductivity of 0.31 (or $10^{-0.5}$) S cm⁻¹ at 650 °C. Also, the Sb_2O_3 and Ta_2O_5 enlarged the stability domain of the δ -Bi₂O₃ phase [16]. This has encouraged us to use Ta_2O_5 and $Sb_2O_3(Sb_2O_5)$ with TeO_2 in the present study as dopants, with the aim to isolate $Bi_2O_3\delta$ -phase with best value of conductivity. The present research investigates the influence of selected ions $(Ta^{5+}, Sb^{3+}/Sb^{5+})$ and Te^{4+}), introduced as dopant in α -Bi₂O₃ and its effect on the structure and electrical conductivity.

2. Experimental

The method of solid-state chemical reaction was applied to synthesize the compositions of the type δ -Bi₂O₃ using the appropriate quantities of high purity oxides powder α -Bi₂O₃ (99.99%), α -TeO₂ (99.99%), Sb₂O₃ (99.99%) and Ta₂O₅ (99.99%). The starting compounds were mixed in an agate mortar and then heated at successively higher temperatures (700, 800, and 850 °C for 24 h). After each heating treatment, the sample was cooled down to room temperature and re-ground to improve homogeneity.

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The electrical study was performed using an electrical impedance spectrometer SOLARTRON SI 1260 coupled to an electrical cell operating under air and in the temperature range from 200 °C to 700 °C. The samples were cylindrical pellets (diameter 12.8 ± 0.1 mm, thickness 3 ± 0.05 mm) initially compacted at 5 kbar under ambient conditions. The pellets were placed between two cylindrical platinum electrodes in a specific cell. A constant pressure was applied to the electrodes via rings. The cell was placed in a furnace operating at up to 700 °C. The electrical analyses were carried out in the frequency range $(\omega = 2\pi\nu)10^{-3}$ to 10^6 , with an alternating current associated with a maximum voltage of 0.1 V. Samples were stabilized for 15 min at a fixed temperature. The recording time for the frequency range was 15 min. To ensure thermal stabilization of pellets, all compositions were subjected to two successive measuring cycles (with one temperature rise and drop for two cycles). The final impedance data were measured during heating mode of the second cycle.

The impedances Z = Z' + jZ'' (Z' and Z'' being respectively the real and imaginary components) were represented using Nyquist plots (X = Z', Y = -Z''). The software Zview [17] was used to fit the impedance of specific electrical circuit to the Nyquist experimental data (the Nyquist representations). The equivalent circuits associated with the all samples were generally based of parallel RC circuit.

3. Results and discussion

3.1. Structural characterization

The results of XRD analyses show that all of ours samples are formed with a fluorite fcc structure. The Xray diffraction spectra of the compositions are shown in Fig. 1.



The pattern refinement of the synthesized phases was carried out by means of the Rietveld method using the FULLPROF program [18]. The factors of reliability are $R_f = 0.91\%$, 2.34% and 2.54%, $R_{\text{Bragg}} =$

1.25%, 2.88% and 2.15% for $Bi_{0.9}Sb_{0.05}Te_{0.05}O_{1.575}$, $Bi_{0.9}Ta_{0.05}Te_{0.05}O_{1.575}$ and $Bi_{0.8}Ta_{0.1}Te_{0.1}O_{1.65}$, respectively. All the experimental details and crystallographic data are summarized in Table I.

The Rietveld refinements of X-ray powder diffraction data indicate that the cations Bi, (Ta or Sb), and Te are localized in (4a). The oxygen occupied the sites 8c and 32f, respectively [19].

TABLE I Experimental details and crystallographic

Space group		$Fm\bar{3}m$
	${\rm Bi}_{0.9}{\rm Sb}_{0.05}{\rm Te}_{0.05}{\rm O}_{1.575}$	5.523(4)
a [Å]	${\rm Bi}_{0.9}{\rm Ta}_{0.05}{\rm Te}_{0.05}{\rm O}_{1.575}$	5.485(4)
	${\rm Bi}_{0.8}{\rm Ta}_{0.1}{\rm Te}_{0.1}{\rm O}_{1.65}$	5.523(6)
Wavelength [Å]		1.5406 - 1.54426
Angular range $[^{\circ}]$		$20 < 2\theta < 90$
Step of measurement $[^{\circ}/\min]$		0.01
Integration time [s]		100

data of synthesized phases.

3.2. Electrical impedance spectroscopy

The evolution of the conductivity was followed at different temperatures. The shapes of the curves are variable according to the composition of each material; it depends in particular on the grain size according on the compound.

We present as an example the Nyquist plots for $Bi_{0.9}Ta_{0.05}Te_{0.05}O_{1.575}$ and $Bi_{0.9}Sb_{0.05}Te_{0.05}O_{1.575}$ at $T = 300 \,^{\circ}C$ (Fig. 2).



Fig. 2. Nyquist plot obtained at 300 °C for $Bi_{0.9}Ta_{0.05}Te_{0.05}O_{1.575}$ and $Bi_{0.9}Sb_{0.05}Te_{0.05}O_{1.575}$.

For $Bi_{0.9}Sb_{0.05}Te_{0.05}O_{1.575}$, the Nyquist plot is constituted by two Nyquists circles; the first one is associated with grain core conduction while the second one is associated with grain boundary. The Nyquist diagram for $Bi_{0.9}Ta_{0.05}Te_{0.05}O_{1.575}$ presents a broad and flattened semicircle which is associated with grain-core conduction. At high temperature (from $640 \,^{\circ}$ C), we noticed the same behavior for both compositions. The Nyquist plot is formed by a vertical line which is observed within the range of high frequencies corresponding to negative values of Z'' and a semicircle which is very flattened in the zone of lower frequencies. We have the presence of Warburg signal corresponding to ionic diffusion at the electrode interfaces (Fig. 3).



Fig. 3. Nyquist plot obtained at high temperature.

The modeling of the experimental results was performed using the Zview software [17]. For each composition and each temperature, several types of equivalent circuits have been proposed and tested. We have kept the models that agree more with the experimental results for each sample. Figure 4 allows comparing the experimental results with simulation results obtained from the equivalent circuits used at different temperatures and for both studied compositions ($Bi_{0.9}Ta_{0.05}Te_{0.05}O_{1.575}$ and $Bi_{0.9}Sb_{0.05}Te_{0.05}O_{1.575}$).



Fig. 4. Comparison between experimental results and obtained models for (a) $Bi_{0.9}Sb_{0.05}Te_{0.05}O_{1.575}$ at 340 °C and $Bi_{0.9}Ta_{0.05}Te_{0.05}O_{1.575}$ (b) at 700 °C.

The electrical conductivity σ (S cm⁻¹) can be expressed by the formula:

$$\sigma = \frac{1}{R} \frac{t}{S},$$

where R (Ω) is the electrical resistance, t/S is the geometrical factor of the pellet including the thickness t (cm) and the area S (cm²).

The electrical conductivity was determined to be $2.26 \times 10^{-4} \text{ S cm}^{-1}$ and $3.06 \times 10^{-4} \text{ S cm}^{-1}$ at 700 °C for $Bi_{0.9}Sb_{0.05}Te_{0.05}O_{1.575}$ and $Bi_{0.9}Ta_{0.05}Te_{0.05}O_{1.575}$, respectively.

The conductivity follows the Arrhenius law [20, 21]:

$$\sigma = \sigma_0 \exp(-E_a/RT)$$

where σ_0 , R and E_a denote the pre-exponential factor, the Boltzmann constant R = 8.314 J K⁻¹ mol⁻¹ and the activation energy, respectively.

The activation energy values (E_a) of the oxygen ion mobility were determined experimentally according to the equation and were calculated from the Arrhenius plots. It is 0.51 eV and 0.48 eV for Bi_{0.9}Sb_{0.05}Te_{0.05}O_{1.575} and Bi_{0.9}Ta_{0.05}Te_{0.05}O_{1.575}, respectively (Fig. 5).

This weak value that the samples exhibit in conductivity at the same concentration of dopants (5%) is surprising. This phenomenon could be explained by the nature of dopants. The cations Sb^{3+} , Te^{4+} and Bi^{3+} occupied the same site and all of them possess a lone pair of electrons forming probably a sphere as a screen that prevents the mobility of oxygens and influences negatively on the electrical conductivity.



Fig. 5. Arrhenius plots for (a) $Bi_{0.9}Sb_{0.05}Te_{0.05}O_{1.575}$ and (b) $Bi_{0.9}Ta_{0.05}Te_{0.05}O_{1.575}$.

4. Conclusion

The investigation of Bi₂O₃-Ta₂O₅-TeO₂ and Bi₂O₃-Sb₂O₃(Sb₂O₅)-TeO₂ ternary systems enabled us to stabilize the delta phase of Bi₂O₃ at room temperature by introducing Sb₂O₃(Sb₂O₅) and TeO₂ then Ta₂O₅ and TeO₂ into α -Bi₂O₃ with formulae (1 - x)BiO_{1.5}-(x/4)Sb₂Te₂O₉ (x = 0.1) and (1 - x)BiO_{1.5}-(x/4)Ta₂Te₂O₉ (x = 0.1, 0.2), respectively. As clarified previously, the property of lone pair contained in selected cations as well as their localization in the same site (4a) are probably the major factors that lead to obtain the unsatisfying values of conductivity.

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