

Magnetic Properties and Phase Constitution of the Nanocrystalline $(\text{Nd}_{10}\text{Fe}_{67}\text{B}_{23})_{100-x}\text{Nb}_x$ (where $x = 1, 2, 3, 4$) Alloy Ribbons

M. KAŻMIERCZAK, K. PAWLIK, P. PAWLIK, P. GĘBARA, A. PRZYBYŁ, J.J. WYSŁOCKI
Institute of Physics, Częstochowa University of Technology, Armii Krajowej 19, 42-200 Częstochowa, Poland

(Received February 2, 2015)

In the present work the magnetic properties and phase constitution of $(\text{Nd}_{10}\text{Fe}_{67}\text{B}_{23})_{100-x}\text{Nb}_x$ (where $x = 1, 2, 3, 4$) alloys in a form of ribbons were investigated. The ribbon samples were obtained by controlled atmosphere melt-spinning technique. In order to generate the nanocrystalline microstructure, ribbons were annealed at various temperatures (from 923 K to 1023 K) for 5 min. Subsequent annealing resulted in an evolution of the phase constitution accompanied by a change of their magnetic properties. The X-ray diffraction studies show presence of hard magnetic $\text{Nd}_2\text{Fe}_{14}\text{B}$, paramagnetic $\text{Nd}_{1+\epsilon}\text{Fe}_4\text{B}_4$ and ferromagnetic metastable $\text{Nd}_2\text{Fe}_{23}\text{B}_3$ phases. The best hard magnetic parameters were measured for annealed ribbons of the $(\text{Nd}_{10}\text{Fe}_{67}\text{B}_{23})_{96}\text{Nb}_4$ alloy.

DOI: [10.12693/APhysPolA.128.91](https://doi.org/10.12693/APhysPolA.128.91)

PACS: 75.20.En, 91.60.Pn

1. Introduction

In modern technology permanent magnets are used as key components of electromechanical, electronic and medical devices [1, 2]. Nanocrystalline RE–Fe–B-based magnets have attracted much attention since their discovery in 1984 [3, 4] and have been extensively studied until now. Much of the work was focused on Nd–Fe–B and Pr–Fe–B alloys [5–9], due to their excellent hard magnetic properties. The microstructure and phase constitution have significant influence on coercivity JH_c , remanence B_r and maximum energy product $(BH)_{\max}$. In recent years, one of the most frequently used methods for production of magnets is the melt-spinning technique. In order to obtain optimal magnetic properties, appropriate alloy composition as well as suitable heat treatment, are crucial issues [5].

Currently, the bulk rapidly solidified nanocrystalline of Nd–Fe–B and Pr–Fe–B alloys doped with Nb, with a high boron content, produced in a form of amorphous rods, tubes or plates, are intensively studied [6]. Relatively high coercivities were measured for 1 mm diameter rod samples of Nd–Fe–B alloy doped with 4 at.% of Nb [7, 8]. Furthermore, it was shown in [9] that large content of B influences glass forming abilities (GFA), of this group of alloys, while Nb addition has a significant impact not only on the GFA, but also retards the growth of nanocrystalline grains formed during annealing [9, 10]. The heat treatment process is an important factor leading to optimization of magnetic properties of magnets. The aim of the present work was to determine the influence of Nb addition and annealing conditions on the phase constitution and magnetic properties of ribbons produced from the $(\text{Nd}_{10}\text{Fe}_{67}\text{B}_{23})_{100-x}\text{Nb}_x$ (where $x = 1, 2, 3, 4$) alloys.

2. Experimental material and methods

The ingot samples of nominal composition of $(\text{Nd}_{10}\text{Fe}_{67}\text{B}_{23})_{100-x}\text{Nb}_x$ (where $x = 1, 2, 3, 4$) were

prepared by arc-melting under an argon atmosphere the high purity constituent elements with addition of pre-alloyed Fe–B of known composition. The samples were remelted several times in order to obtain homogeneity. Next, ribbon samples were prepared by melt-spinning technique under the Ar atmosphere. In order to generate amorphous structure the linear speed of the copper roll surface of 35 m/s was used. Subsequently the ribbon samples were sealed/off in a quartz tubes under low pressure of argon to maintain the purity of atmosphere during heat treatment. In order to obtain a nanocrystalline microstructure, the samples were heat treated at temperatures ranging from 923 K to 1023 K for 5 min. The phase analysis of these samples was studied using X-ray diffractometry (XRD) with Cu K_α radiation. Hysteresis loops were measured by LakeShore 7307 vibrating sample magnetometer at external magnetic field up to 2 T at room temperature.

3. Results

The XRD scans measured for $(\text{Nd}_{10}\text{Fe}_{67}\text{B}_{23})_{100-x}\text{Nb}_x$ (where $x = 1, 2, 3, 4$) alloy ribbons in as-cast state are shown in Fig. 1. This studies suggest presence of amorphous structure, evidenced by the lack of peaks corresponding to crystalline phases for all alloy compositions. Annealing at 923 K and 943 K for 5 min resulted in small changes in the crystal structure of the material (Figs. 2, 3). Broadened peaks originating from crystalline phase are observed on the diffraction patterns. The low intensity of these peaks in comparison to the background did not allow to clearly identify crystalline phases present in the sample.

The XRD patterns measured for the compound $(\text{Nd}_{10}\text{Fe}_{67}\text{B}_{23})_{100-x}\text{Nb}_x$ (where $x = 1, 2, 3, 4$) alloy ribbon samples annealed at 963 K for 5 min, are shown in Fig. 4.

Annealing for 5 min at 963 K and higher temperatures, led to nucleation and growth of the crystalline phases.

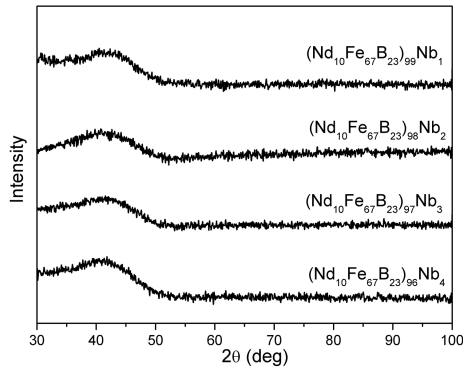


Fig. 1. X-ray diffraction patterns of as-cast $(\text{Nd}_{10}\text{Fe}_{67}\text{B}_{23})_{100-x}\text{Nb}_x$ (where $x = 1, 2, 3, 4$) alloy ribbon samples.

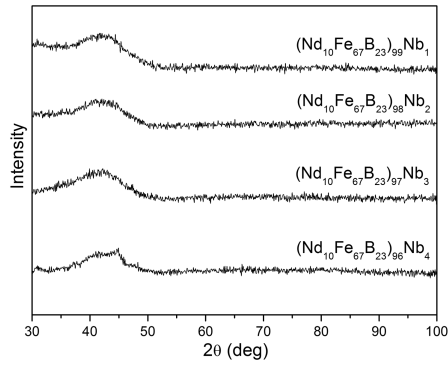


Fig. 2. X-ray diffraction patterns of the compound $(\text{Nd}_{10}\text{Fe}_{67}\text{B}_{23})_{100-x}\text{Nb}_x$ (where $x = 1, 2, 3, 4$) alloy ribbon samples annealed at 923 K for 5 min.

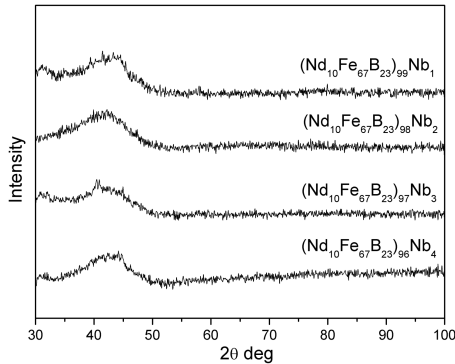


Fig. 3. As in Fig. 2, but for 943 K.

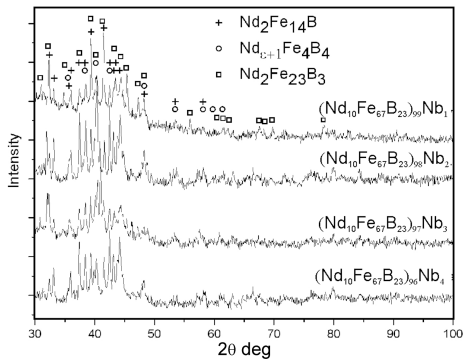


Fig. 4. As in Fig. 2, but for 963 K.

The main crystalline phase precipitating during annealing was the hard magnetic $\text{Nd}_2\text{Fe}_{14}\text{B}$. The analysis of XRD patterns also indicated a presence of the paramagnetic $\text{Nd}_{\varepsilon+1}\text{Fe}_4\text{B}_4$ and metastable $\text{Nd}_2\text{Fe}_{23}\text{B}_3$ metastable phases [11]. However, due to the proximity of peaks coming from the $\text{Nd}_2\text{Fe}_{14}\text{B}$, the presence of this phase was confirmed by the Mössbauer spectroscopy. For samples annealed at 963 K and higher, no change in phase composition of the alloy was observed. Also the intensities of peaks corresponding to the crystalline phases are not significantly changed. However, in case of the $(\text{Nd}_{10}\text{Fe}_{67}\text{B}_{23})_{96}\text{Nb}_4$ ribbon annealed at 1023 K for 5 min an evolution in the phase composition occurred. Except the dominant $\text{Nd}_2\text{Fe}_{14}\text{B}$ hard magnetic phase, diffraction peaks from soft magnetic $\alpha\text{-Fe}$ were also observed.

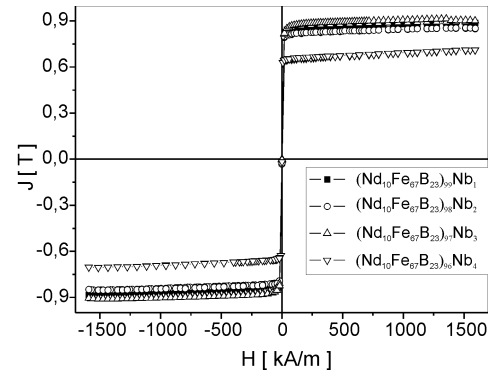


Fig. 5. The hysteresis loops measured for the as-cast $(\text{Nd}_{10}\text{Fe}_{67}\text{B}_{23})_{100-x}\text{Nb}_x$ (where $x = 1, 2, 3, 4$) alloy ribbon samples.

The hysteresis loops measured for as-cast ribbons are shown in Fig. 5. As-cast ribbons were soft magnetic confirming their fully amorphous structure. Increase of Nb in the alloy composition resulted in significant decrease of saturation polarization. Annealing at 923 K and 943 K resulted in a small change in the shape of hysteresis loops, due to the low fraction of crystalline phases in the constitution of the samples.

The hysteresis loops for ribbons subjected to annealing at 963 K and 1023 K for 5 min, are shown respectively in

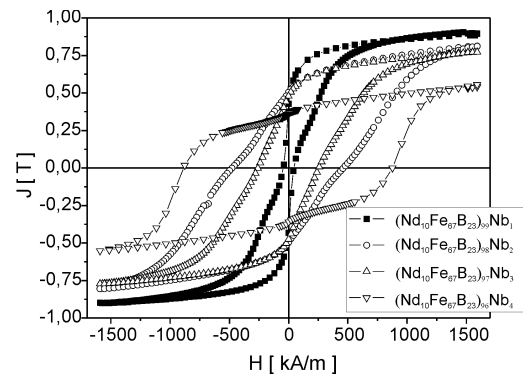


Fig. 6. The hysteresis loops measured for $(\text{Nd}_{10}\text{Fe}_{67}\text{B}_{23})_{100-x}\text{Nb}_x$ (where $x = 1, 2, 3, 4$) alloy ribbon samples annealed at 963 K for 5 min.

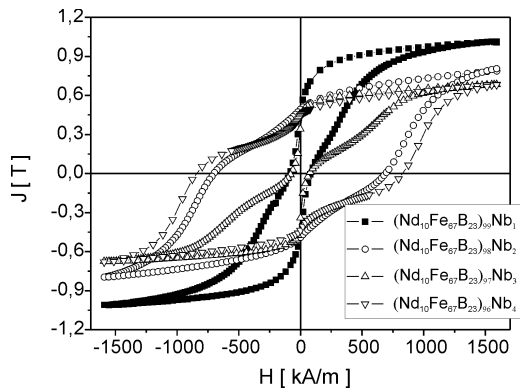


Fig. 7. As in Fig. 6, but for 1023 K.

Figs. 6 and 7. The magnitude of saturation magnetization decreased slightly in comparison to the corresponding as-cast $(\text{Nd}_{10}\text{Fe}_{67}\text{B}_{23})_{100-x}\text{Nb}_x$ (where $x = 1, 2, 3, 4$) ribbons. The addition of Nb led to an enhancement of the coercivity.

Further annealing at 1023 K of ribbons of all compositions resulted in decrease of their magnetic parameters. Especially in case of ribbon containing 4 at.% of Nb, even though the coercivity reaches 833 kA/m, $(BH)_{\text{max}}$ decreases to 96 kJ/m³ due to low value of remanence $J_r = 0.44$ T. Similar results were obtained by Zhang et al. [9] and Tamura et al. [7].

4. Conclusions

The magnetic properties and phase constitution of the $(\text{Nd}_{10}\text{Fe}_{67}\text{B}_{23})_{100-x}\text{Nb}_x$ (where $x = 1, 2, 3, 4$) alloy ribbons annealed at various temperatures from 923 K to 1023 K for 5 min were investigated. It was found that $(\text{Nd}_{10}\text{Fe}_{67}\text{B}_{23})_{100-x}\text{Nb}_x$ (where $x = 1, 2, 3, 4$) alloy ribbons, produced by free jet melt-spinning technique had fully amorphous structure and soft magnetic properties. Heat treatment led to nucleation and growth of the $\text{Nd}_2\text{Fe}_{14}\text{B}$ hard magnetic phase and $\text{Nd}_{\varepsilon+1}\text{Fe}_4\text{B}_4$ paramagnetic and $\text{Nd}_2\text{Fe}_{23}\text{B}_3$ metastable phases which was confirmed by the Mössbauer spectroscopy. Furthermore, in XRD carried out on the $\text{Nd}_{9.6}\text{Fe}_{64.32}\text{Nb}_4\text{B}_{22.08}$ alloy ribbon annealed at 1023 K, confirmed presence of the soft magnetic α -Fe phase. Magnetic studies have revealed that the heat treatment led to nanocrystalline structure of ribbon samples and good magnetic properties. Furthermore, it was found that the relevant contents of niobium has a significant influence on the magnetic properties. The ribbon samples containing 4 at.% of Nb exhibit the best magnetic properties.

References

- [1] M.T. Thompson, *Proc. IEEE* **97**, 1 (2009).
- [2] J.I. Betancourt, *Rev. Mexic. Fis.* **48**, 283 (2002).
- [3] J.J. Croat, J.F. Herbst, R.W. Lee, F.E. Pinkerton, *J. Appl. Phys.* **55**, 2078 (1984).
- [4] M. Sagawa, S. Fujimura, N. Togawa, H. Yamamoto, J. Matsuura, *J. Appl. Phys.* **55**, 2083 (1984).
- [5] A. Manaf, R.A. Buckley, H.A. Davies, *J. Magn. Magn. Mater.* **128**, 302 (1993).
- [6] K. Pawlik, P. Pawlik, J.J. Wysocki, *Acta Phys. Pol. A* **118**, 900 (2010).
- [7] R. Tamura, S. Kobayashi, T. Fukuzaki, M. Isobe, Y. Ueda, *J. Phys. Conf. Series* **144**, 012068 (2009).
- [8] M. Szwaja, P. Pawlik, J.J. Wysocki, P. Gębara, *Archiv. Metall. Mater.* **57**, 233 (2012).
- [9] J. Zhang, K.Y. Lim, Y.P. Feng, Y. Li, *Scr. Mater.* **56**, 943 (2007).
- [10] P. Pawlik, K. Pawlik, H.A. Davies, J.J. Wysocki, W. Kaszuwara, *J. Phys. Conf. Ser.* **144**, 012060 (2009).
- [11] H. Mayot, O. Isnard, J.-L. Soubeyroux, *J. Magn. Magn. Mater.* **316**, e477 (2008).