

Mechanical Properties of Tetragonal and Orthorhombic Phases of Quasi-One-Dimensional Antiferromagnet KCuF_3

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The density functional theory was used to calculate the elastic constants for the two tetragonal (*a*-type) and (*d*-type), and the orthorhombic structures of KCuF_3 . Based on the single elastic constants the polycrystalline mechanical properties such as bulk, shear, Young moduli and Poisson ratio using Reuss-Voigt-Hill averaging method were estimated. Furthermore, the sound transverse and longitudinal velocities (along the *a*-axis and *c*-axis) were calculated and the Debye temperatures were determined for all three investigated phases.

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1. Introduction

KCuF_3 is well-known example of the one-dimensional magnetic system in three-dimensional magnetic ion sublattice [1] that exists in three different structures; two of them possessing tetragonal symmetry (*I4/mcm* and *P4/mbm*) and one with orthorhombic symmetry (*P2₁2₁2₁*) [1-2]. For the extended list of physical quantities known about the KCuF_3 as well as the description of the lattice dimensions and Wyckoff positions of the constituents we refer the reader to the recent publication [3]. Here, we report the single crystal elastic constants of the all three polymorphs of KCuF_3 using *ab initio* calculations. Based on those, we investigate the mechanical properties, such as Young, bulk and shear moduli, Poisson ratios, ductility and the Debye temperatures using longitudinal and transverse sound velocities.

2. Methodology

First-principles calculations were performed using the density functional theory [4] in a single electron framework of the VASP code [5]. For details we refer the reader to Ref. [3] where all parameters are specified. For the calculations of elastic constants (C_{ij}) we applied methodology implemented in the VASP code.

3. Results and discussion

The elastic constants of both tetragonal and orthorhombic KCuF_3 structures are summarized in Table I. In general, the intrinsic mechanical stability of a solid is determined by certain conditions concerning the values of C_{ij} related to the crystal symmetry [6]. For all KCuF_3 polymorphs the C_{ij} are positive. The stability conditions

for tetragonal (*I4/mcm* and *P4/mbm*) and orthorhombic (*P2₁2₁2₁*) polymorphs, see e.g. [7], are here fulfilled and hence these three phases of KCuF_3 are mechanically stable. The slight differences between C_{ij} of the two tetragonal polymorphs are observed predominantly for C_{11} , C_{12} , and C_{13} , all being higher (stiffer) by about 4-7 GPa in the *P4/mbm* structure with respect to the *I4/mcm* structure, whereas the C_{33} , C_{44} , and C_{66} are only slightly higher. The increase in C_{ij} , can be attributed to the different location of the fluorine atoms within the tetragonal plane, see e.g. Fig. 1 in [3]. In general, the structure with orthorhombic symmetry is described by 9 independent elastic constants. Results of calculations shown in Table I indicate that the *P2₁2₁2₁* structure of KCuF_3 behaves elastically as being almost tetragonal one, having the values of $C_{11} \approx C_{22}$, $C_{13} \approx C_{23}$, and $C_{44} \approx C_{55}$.

TABLE I
Elastic constants C_{ij} [GPa] of tetragonal (*I4/mcm*, *P4/mbm*) and orthorhombic polymorphs of KCuF_3 .

polymorph	C_{11}	C_{12}	C_{13}	C_{22}	C_{23}	C_{33}	C_{44}	C_{55}	C_{66}
<i>I4/mcm</i>	70	18	33	—	—	129	29	—	25
<i>P4/mbm</i>	77	23	37	—	—	130	31	—	28
<i>P2₁2₁2₁</i>	65	25	35	66	36	134	29.9	30.3	28

For low symmetry phases the isotropic polycrystalline elastic constants, bulk (B_{RVH}) and shear (G_{RVH}) moduli can be calculated by applying the so-called Reuss-Voigt-Hill averaging method and using the single elastic constants C_{ij} [8,9]. The mechanical properties of the polycrystalline solid such are Young modulus (E), Poisson ratio (ν), and B_{RVH}/G_{RVH} (Pugh's modulus ratio [10]) are also determined and listed in Table II. The Pugh's modulus ratio is usually used as an empirical rule to distinguish the ductile ($B_{RVH}/G_{RVH} > 1.75$) from the brittle material ($B_{RVH}/G_{RVH} < 1.75$). One can see in Table II that the orthorhombic phase is on the edge of being brittle, whereas both tetragonal phases are rather ductile. Similarly, the ductility is correlated with the lower values of E and G_{RVH} and higher value of ν , i.e.

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with weaker and less directional bonding [11]. Hence, corresponding values of Table II indicate that the ductility decreases going from $I4$ to $P4$ and to $P2_12_12_1$ structure of KCuF_3 .

TABLE II

The Reuss-Voigt-Hill averaged bulk (B_{RVH}), shear (G_{RVH}) and Young (E) moduli in GPa, Pugh's modulus ratio (B_{RVH}/G_{RVH}), Poisson ratio (ν) of tetragonal ($I4/mcm$, $P4/mbm$) and orthorhombic ($P2_12_12_1$) polymorphs of KCuF_3 .

polymorph	B_{RVH}	G_{RVH}	B_{RVH}/G_{RVH}	E	ν
$I4/mcm$	45	23	1.99	59	0.285
$P4/mbm$	51	24	2.13	62	0.294
$P2_12_12_1$	48	28	1.71	70	0.254

Next, we determine the Debye temperature using longitudinal (v_l) and transverse (v_t) sound velocities obtained from B_{RVH} and G_{RVH} following the Ref. [12-14]. The mean sound velocity (v_m), used for the estimation of Debye temperature, was obtained as $\nu_m = \nu_l \frac{2s^2+1}{2s^3+1}$, where $s = \frac{v_l}{v_t}$. All three phases have very similar volume per formula unit [3] leading to the density of 3.64 g/cm^3 , whereas the experimental value is 3.94 g/cm^3 [13]. The former value is used to evaluate the sound velocities. Results are summarized in Table III. For a comparison, we also derived the Debye temperature from the phonon density of states (θ_{Dph}), published in Ref. [3], using the procedure described in Ref. [15]. For all three phases the θ_D (based on C_{ij}) are in very good agreement with θ_{Dph} obtained from the atomic vibrations (phonons). The effect of the experimental ρ on θ_D is negligible, as it lowers θ_D by less than 5 K.

TABLE III

The longitudinal, transverse and mean sound velocities and the Debye temperature calculated using C_{ij} (θ_D) and from the phonon density of states θ_{Dph} of Ref. [3].

phase	v_l (cm/s)	v_t (cm/s)	v_m (cm/s)	θ_D (K)	θ_{Dph} (K)
$I4/mcm$	4.578+E5	2.512+E5	2.800+E5	261	276
$P4/mbm$	4.775+E5	2.577+E5	2.877+E5	268	292
$P2_12_12_1$	4.827+E5	2.772+E5	3.079+E5	289	294

It seems that the increase of the Debye temperature (θ_D) and decrease of the ductility correlates with the phase stability of the KCuF_3 polymorphs. The lowest θ_D and higher ductility have tetragonal $I4/mcm$ and $P4/mcm$ structures whereas the brittle behavior possesses the least stable orthorhombic $P2_12_12_1$ structure.

4. Conclusions

Using the density functional theory calculations we determined the mechanical properties of two tetragonal and one orthorhombic phase of the quasi-one-dimensional antiferromagnet KCuF_3 . Using Hill averaged calculated single elastic constants (C_{ij}) we showed that the brittle

phase is the orthorhombic one, whereas both tetragonal phases are rather ductile. We estimated Debye temperatures for all three phases using C_{ij} and phonon density of states of Ref. [3], all values being in very good accord.

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References

- [1] A. Okazaki, Y. Suemune, *J. Phys. Soc. Japan* **16**, 671 (1961).
- [2] M. Hidaka, T. Eguchi, I. Yamada, *J. Phys. Soc. Japan* **67**, 2488 (1998).
- [3] D. Legut, U.D. Wdowik, *J. Phys.: Condens. Matter* **25**, 115404 (2013).
- [4] P. Hohenberg, W. Kohn, *Phys. Rev.* **136**, B864 (1964).
- [5] G. Kresse, J. Furthmüller, *Phys. Rev. B* **54**, 11169 (1996).
- [6] D.C. Wallace, *Thermodynamics of Crystals*, Wiley, New York 1972.
- [7] O. Beckstein, J.E. Klepeis, G.L.W. Hart, O. Pankratov, *Phys. Rev. B* **63**, 134112 (2001).
- [8] R. Hill, *Proc. Phys. Soc. A* **65**, 349 (1952).
- [9] G. Grimvall, *Thermophysical Properties of Materials* Elsevier, Amsterdam 1999. ISBN-10: 0444827943.
- [10] S.F. Pugh, *Phil. Mag.* **45**, 823 (1954).
- [11] F. Chu, T.E. Mitchell, B. Majumdar, D. Miracle, T.K. Nandy, D. Banerjee, *Intermetallics* **5**, 147 (1997).
- [12] O.L. Anderson, *J. Phys. Chem. Solids* **24**, 909 (1963).
- [13] R. Berman, *Thermal Conductivity in Solids* Clarendon Press, Oxford 1976. ISBN: 0198514298.
- [14] V. Tkáč, A. Orendáčová, M. Orendáč, D. Legut, K. Tibenská, A. Feher, M. Poirier, M.W. Meisel, *Acta Phys. Pol. A* **121**, 1098 (2012).
- [15] D. Legut, U.D. Wdowik, *J. Phys. Cond. Matter.* **22**, 435402 (2010).