

Enhancement of the Ultraviolet Luminescence Intensity from Cd-Doped ZnO Films Caused by Exciton Binding

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ZnO films doped with the cadmium (0.4–0.6%) were grown on crystalline sapphire *c*-Al₂O₃ substrates applying radiofrequency magnetron sputtering at the temperature of 400 °C in Ar–O₂ atmosphere. The as-grown films were investigated in detail using X-ray diffraction, X-ray photoelectron spectroscopy, and cathodoluminescence spectra. The X-ray diffraction analysis revealed that the films possess a hexagonal wurtzite-type structure with the dominant crystallite orientation along the *c* axis. It was found that the small concentration of the cadmium significantly enhances the ultraviolet emission associated with excitonic transitions. We suggest that this enhancement effect mainly results from appearance of the cadmium isoelectronic traps, which may bind an exciton, thereby increasing the probability of radiation recombination. The effect of Cd isoelectronic impurity on structural and luminescent properties of ZnO films is discussed.

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1. Introduction

Recently, zinc oxide (ZnO) films have been extensively investigated due to their potential for the development of novel optoelectronic devices. Their excellent properties such as a wide band gap (3.37 eV at 300 K), a large exciton binding energy (60 meV), and a large melting point of 1975 °C ensure applications in photodetectors, light emitting diodes and laser light source [1–3]. For ZnO, addition of impurities often induces dramatic changes in their electrical and optical properties, which suggests new applications.

For the development of devices based on the zinc oxide, the knowledge about the effect of isoelectronic impurities is of essential interest. It is known that the isoelectronic traps can improve the efficiency of radiative recombination of electrons and holes, as was shown for the light-emitting diodes based on GaP:N [4]. In the paper [5], the authors report the observation of an exciton bound with isoelectronic nitrogen atom in GaP. According to Hopfield et al., the difference of electronegativity between nitrogen and phosphor was suggested to be responsible for the binding of the exciton to the nitrogen center [6]. Alternatively, the binding mechanism can be described as a consequence of the lattice deformation due to the atomic size difference between impurity and host atom.

The goal of this paper is to determine the degree of the cadmium doping influence on the ultraviolet emission of zinc oxide films. In our work it is shown that the isoelectronic impurity of cadmium (0.4 at.%) leads to improvement in structural properties and cathodolu-

minescence of zinc oxide films. By analogy with GaP:N, we explain the enhancement of the UV emission intensity by the appearance of cadmium isoelectronic traps which can bind an exciton. This increases the probability of radiative recombination, i.e. luminescence intensity.

2. Experimental details

ZnO films doped with cadmium were deposited on *c*-Al₂O₃ substrates at the temperature 400 °C by the RF magnetron sputtering. Targets with different ratios of Zn and Cd have a form of zinc metal discs with cadmium inserts (2, 4, 6). Vacuum in deposition chamber was 10⁻³ Pa. The mixture of high-purity (99.99%) argon and oxygen were used as working gases. The specific flux of argon and oxygen was maintained at a ratio of Ar:O₂ ≈ 2 : 1. The distance from the target to the substrate was fixed at 40 mm. The discharge power was 200 W. In order to obtain perfect films, layer-by-layer deposition method was used [7, 8]. The deposition time was 60 min for all samples. The growth process was carried out in four stages. After deposition of every layer the deposition process was interrupted. The interruption time is 5 min. Before deposition of each film, the zinc–cadmium target was presputtered in Ar–O₂ atmosphere for about 10 min in order to remove the surface oxide. The growth rate is about 30 nm/min. The crystal structure of samples was studied by the X-ray diffraction (XRD) where Cu K_α radiation was used as a source (λ = 0.154056 nm). The XRD measurements were carried out with the DRON-4 system. X-ray photoelectron spectroscopy (XPS) valence-band and core-level spectra were measured using the UHV-Analysis-System assembled by SPECS (Germany). The system is equipped with

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a PHOIBOS 150 hemispherical analyzer. A base pressure in the sublimation ion-pumped chamber of the system was less than 9×10^{-8} Pa during the present experiments. The Mg K_{α} radiation ($E = 1253.6$ eV) was used as a source of XPS spectra excitation. The XPS spectra were measured at the constant pass energy of 25 eV. The binding energy (BE) of 84.00 ± 0.05 eV of the XPS Au $4f_{7/2}$ core-level spectrum was used as a reference. Energy drift due to charging effects was calibrated, taking the XPS C 1s (285.0 eV) core-level spectrum of hydrocarbons. The cathodoluminescence (CL) spectra were taken in a JEOL 35C scanning electron microscope equipped with an Oxford Instruments MonoCL2 CL system and Hamamatsu R943-02 Peltier cooled photomultiplier. All CL spectra discussed here were taken at room temperature.

3. Results and discussions

We have obtained the undoped and Cd-doped ZnO films by the rf magnetron sputtering technique. In order to determine the atomic concentrations of the cadmium in the Cd-doped ZnO films, we investigated the XPS spectra for these samples (Fig. 1a). The Cd content in the ZnO films can be evaluated by comparing the integrated areas of the Cd $3d_{5/2}$ peak and the sum of the Cd $3d_{5/2}$ peak together with the Zn $2p_{3/2}$ peak. Considering their ionization cross-sections, the cadmium content for our films is a small one about 0.4%, 0.5%, and 0.6%, respectively (Fig. 1b).

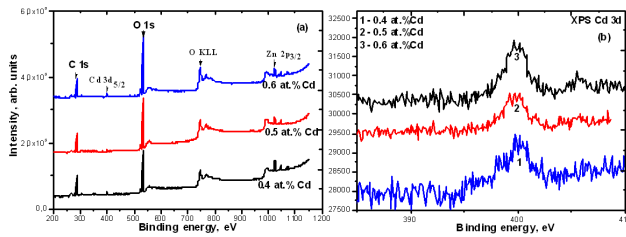


Fig. 1. XPS spectra of Cd-doped (a) and XPS Cd $3d$ core-levels spectra (b) of ZnO films deposited on the $c\text{-Al}_2\text{O}_3$ substrates.

Figure 2a shows the XRD patterns of undoped and Cd-doped ZnO films, in which two prominent peaks are observed. These two peaks are indexed to be (002), (004) crystal planes of hexagonal ZnO according to the standard JCPDS Card (No. 36-1451).

With increasing Cd content, a small decrease of the diffraction angle was observed (Fig. 2b), indicating that the larger ion Cd^{2+} (ionic radius 0.97 Å) replaces the smaller ion Zn^{2+} (ionic radius 0.74 Å) in the cationic sublattice of ZnO. The values of the full width at half maximum (FWHM) for the crystallographic plane (002) are also presented in Fig. 2b. When the cadmium content is about 0.4 at.%, the FWHM reaches the lowest value, indicating the better structural quality of the 0.4 at.% Cd-doped films.

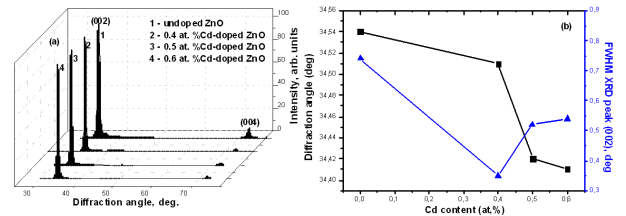


Fig. 2. XRD patterns (a) for undoped and Cd-doped ZnO films deposited on the $c\text{-Al}_2\text{O}_3$ substrates as well as dependences of the diffraction angle (■) and the FWHM peak (▲) on the cadmium content (b).

Figure 3 shows the CL spectra measured at electron beam energy of 15 keV. For 0.4%Cd-doped ZnO film a strong UV near-band emission at approximately 382 nm related to the excitonic transition is observed. It can be seen that CL intensity for the sample with 0.4 at.% cadmium is 1.5 times higher than its value for undoped ZnO (Fig. 4a). The enhancement of luminescence intensity is due to increasing the probability of the processes of radiative recombination. It can be due to the fact that the isoelectronic cadmium atom in ZnO is able to bind an exciton. In order to explain the binding of the exciton to the isoelectronic Cd atom in ZnO, the model of Hopfield et al. [6] might be applied. Hopfield et al. assumed that an isoelectronic trap could be attractive for either electrons or holes. Since there is no net charge involved, the binding will not be due to the Coulomb forces but to forces of very short range. Therefore, it will be impossible to treat this bound state with an effective mass approximation.

After an isoelectronic trap has captured an electron or a hole, this trap is then negatively or positively charged. By the Coulomb interaction, it will capture a carrier of opposite charge. This leads to an exciton bound to the isoelectronic trap. The Pauling electronegativities of Zn and Cd are 1.65 and 1.70, respectively. Thus, within this model, Cd tends to bind an electron more strongly than Zn. Then via the Coulomb interaction an electron binds a hole and, thereby, forms an exciton. In addition, covalent radii of zinc and cadmium are 0.131 nm and 0.148 nm, respectively, i.e. the covalent radius of Cd is 13% greater than the covalent radius of zinc. Due to the induced lattice deformation about the Cd atoms a strain field should develop. Both mechanisms can explain the formation of an isoelectronic trap at Cd in ZnO in a qualitative way.

It was also found from CL spectra that the peak of near band (NBE) emission of 0.4%Cd-doped ZnO film, in comparison to undoped ZnO, blue shifted to the higher-energy side by 17 meV; and the red shift for 0.5%Cd-doped and 0.6%Cd-doped ZnO films was about 21 meV and 28 meV, respectively (Fig. 4b).

Simple diagram of the blue shifting of the NBE emission of 0.4%Cd-doped ZnO film is presented in Fig. 5. The blue shift of NBE emission in the 0.4%Cd-doped

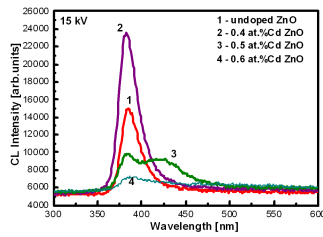


Fig. 3. CL spectra for undoped and Cd-doped ZnO films deposited on the $c\text{-Al}_2\text{O}_3$ substrates at electron energy of 15 keV.

ZnO film is believed to be caused by the Burstein–Moss effect [9]. It is well known that group (II) element dopants, such as Cd, act as donors in ZnO. The donor electrons occupy states at the bottom of the conduction band. Since the Pauli principle prevents double occupancy of states and optical transitions are vertical, the low-energy transitions are blocked. Blockage of the low-energy transitions is known as the Burstein–Moss effect, and it enhances optical energy.

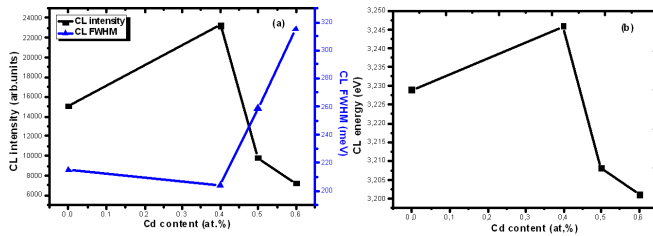


Fig. 4. Dependences of the CL intensity, CL FWHM (a) and CL energy (b) on the cadmium content.

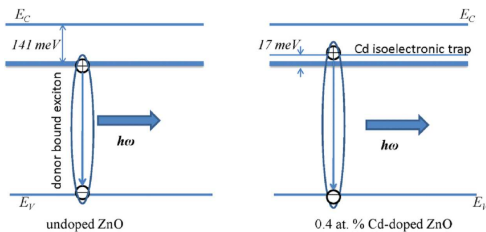


Fig. 5. A simple diagram showing the excitonic transitions in the undoped and 0.4 at.% Cd-doped ZnO films.

The red shift of NBE emission in 0.5%Cd-doped and 0.6%Cd-doped ZnO films cannot be attributed to electrons filling in the lower part of the conduction band, as in the case of the Burstein–Moss effect. This indicates that the band structure, including the band gap itself, must have been substantially modulated. It was found that the CL intensity of 0.5%Cd-doped and 0.6%Cd-doped ZnO films significantly decreases compared to the 0.4%Cd-doped ZnO film, while the CL linewidth (FWHM) increases (Fig. 4a). This increase of broadening of 0.5%Cd-doped and 0.6%Cd-doped ZnO films is

mainly due to the increase of disorder in semiconductor, which leads to the appearance of localized electron and/or hole states [10]. Two localization mechanisms have been suggested [11]. One is that exciton is localized by the Coulomb interaction in very deep potential wells. The other is a case of wide potential well with dimensions larger than the excitonic Bohr radius. In this case, the exciton is localized as a whole in a wide potential well. Martin et al. [10] suggested the additional mechanism. An additional localizing mechanism may be found in the nature of the segregated quantum dots (confinement effect) or alternatively from the large piezoelectric fields present in wurtzite semiconductors (the Coulomb effect). Emission spectra are broadened due to these localized states. Nevertheless, the proposed interpretations of the red- and blue shift of the NBE emission are tentative and need further verification.

4. Conclusions

The Cd-doped ZnO films with different cadmium concentrations have been prepared by radiofrequency magnetron sputtering method on sapphire substrates at 400 °C. The films were characterized by different methods to understand their structural, optical and chemical properties. The X-ray diffraction analysis revealed that the films are polycrystalline in nature having a hexagonal wurtzite type crystal structure with a preferred grain orientation in the (002) direction. It was found that the small concentration of the cadmium (0.4 at.%) significantly enhances the ultraviolet emission associated with excitonic transitions. We suggest that this enhancement effect mainly results from appearance of the cadmium isoelectronic traps, which may bind an exciton, thereby increasing the probability of radiation recombination. CL measurements showed that the band gap decreases from 3.229 (undoped ZnO) to 3.201 eV (0.6 at.% Cd) with increasing Cd concentration. Increasing the Cd concentration (> 0.4 at.%) also leads to the broadening of the emission peak and degrading the crystalline quality. The characterization studies clearly indicate the incorporation of Cd into ZnO; hence the observed decrease in the CL energy which can be directly attributed to the effect of Cd ion incorporation into the ZnO lattice. The possible mechanisms of the Cd influence on the luminescence features of the ZnO films are proposed.

References

- [1] O. Lupan, L. Chow, G. Chai, L. Chernyak, O. Lopatiuk-Tirpak, H. Heinrich, *Phys. Status Solidi A* **205**, 2673 (2008).
- [2] S.D. Lee, Y.S. Kim, M.S. Yi, J.Y. Choi, S.W. Kim, *J. Phys. Chem. C* **113**, 8954 (2009).
- [3] Y. Ohno, T. Shirahama, S. Takeda, A. Ishizumi, Y. Kanemitsu, *Appl. Phys. Lett.* **87**, 043106 (2005).
- [4] P.J. Dean, *J. Lumin.* **7**, 51 (1973).
- [5] D.G. Thomas, J.J. Hopfield, C.J. Frosch, *Phys. Rev. Lett.* **15**, 857 (1965).

- [6] J.J. Hopfield, D.G. Thomas, R.T. Lynch, *Phys. Rev. Lett.* **17**, 312 (1966).
- [7] A. Ievtushenko, V. Karpyna, G. Lashkarev, V. Lazorenko, V. Baturin, A. Karpenko, M. Lunika, A. Dan'ko, *Acta Phys. Pol. A* **114**, 1131 (2008).
- [8] A.I. Ievtushenko, V.A. Karpyna, V.I. Lazorenko, G.V. Lashkarev, V.D. Khranovskyy, V.A. Baturin, O.Y. Karpenko, M.M. Lunika, K.A. Avramenko, V.V. Strelchuk, O.M. Kutsay, *Thin Solid Films* **518**, 4529 (2010).
- [9] J.I. Pankove, *Optical Processes in Semiconductors*, Prentice-Hall, NJ 1971.
- [10] R.W. Martin, P.G. Middleton, K.P. O'Donnell, W. Van der Stricht, *Appl. Phys. Lett* **74**, 263 (1999).
- [11] C.F. Klingshirn, *Semiconductor Optics*, Springer, Berlin 1995.