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Anharmonic Optical Phonon Effects in ZnO Nanocrystals

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Zinc oxide (ZnO) is a very promising material for optoelectrical devices operating at the short-wavelength end of the visible spectral range and at the near UV. The Raman scattering studies of ZnO heterolayers formed by isothermal annealing show sharp phonon lines. In addition to the $A_1(\text{TO})$, $E_1(\text{TO})$, E_2^{H} , and $E_1(\text{LO})$ one-phonon lines, we observed two-phonon lines identified as: $E_2^{\text{H}} - E_2^{\text{L}}$, $E_2^{\text{H}} + E_2^{\text{L}}$, and 2LO at 332, 541, and 1160 cm^{-1} , respectively (at room temperature). The identification of the $E_2^{\text{H}} - E_2^{\text{L}}$ peak was confirmed by its thermal dependence. Temperature dependent measurements in the range 6–300 K show that the phonon frequencies decrease with temperature. The E_2^{H} peak is at energy 54.44 meV (439.1 cm^{-1}), at 4 K and due to phonon–phonon anharmonic interaction, its energy decreases to 54.33 meV (438.2 cm^{-1}) at room temperature. The Grüneisen parameter found for this oscillation mode was $\gamma_{E_2^{\text{H}}} = 1.1$ at about 300 K. The intensity of the $E_2^{\text{H}} - E_2^{\text{L}}$ peak increases strongly with temperature and this dependence can be described by the Bose–Einstein statistics with activation energy of 13.8 meV (nearly equal to the energy of the E_2^{L} phonon).

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1. Introduction

The II–VI compounds are well known for their bright luminescence and good photoelectric properties. Between them zinc oxide (ZnO) has special place with its direct and wide band gap (about 3.2 eV at 300 K). For example ZnO diodes are very good detectors and emitters in the short-wavelength end of the visible spectral range and at the near UV [1, 2]. The opto-electronic devices are strongly dependent on electron–phonon scattering thus the lattice dynamics of the material is very important.

The Raman scattering spectroscopy is one of the most popular phonon research techniques. The temperature dependences of the phonon frequency follow from anharmonic terms in the vibrational Hamiltonian of the crystal lattice. The anharmonicity gives rise to the thermal expansion of the lattice and to the phonon–phonon interaction.

In the wurtzite (hexagonal) phase, the ZnO primitive cell consists of two Zn–O pairs (twice more than in the case of the cubic phase) so nine optical modes appear. The symmetry representations are: $\Gamma_{\text{opt}} = A_1 + E_1 + 2B_1 + 2E_2$. The B_1 modes are optically inactive. The E_2 modes are nonpolar, but they give strong signal in the Raman spectra (especially the high-frequency E_2^{H} mode). The polar A_1 and E_1 modes can be longitudinal (LO), and transverse (TO) [3].

2. Samples

Substrates measuring $4 \times 4 \times 1$ mm were cut from bulk crystals of cubic ZnSe, grown by the Bridgman method

from stoichiometric melts. The substrate surface was mirror-smooth, and the substrates showed bulk photoluminescence (PL) under excitation by the 337 nm nitrogen laser line. Zinc oxide layers were produced by isovalent substitution method [4]. Isothermally annealing was realized in air at temperatures from 850 to 1050 °C, and time in the range of 5–30 min. The width of the ZnO layers was from 0.1 to 1 μm depending on growth conditions.

X-ray diffraction (XRD) characterization was conducted in X'Pert MPD Philips Analytical Rentgen diffractometer in θ – 2θ configuration. The lattice parameters were determined by optimization to powder-pattern indexing method [5].

The XRD of the ZnO heterolayers (Fig. 1) revealed that the ZnO is polycrystalline. This conclusion is based

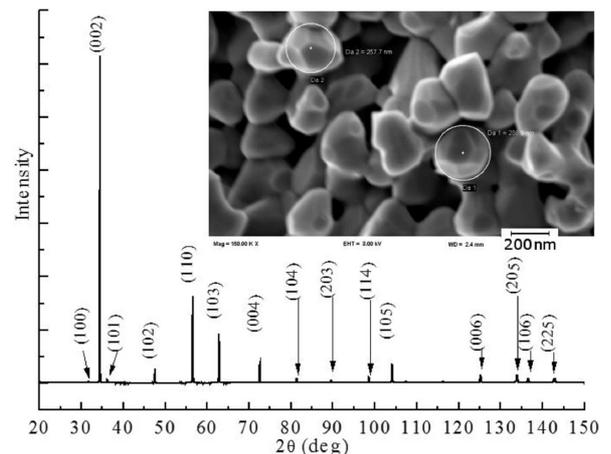


Fig. 1. X-ray diffraction spectrum of the ZnO heterolayers formed on ZnSe. The inset: electron microscope picture of the sample.

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on the presence of many reflections from different crystallographic planes. All diffraction peaks were attributed to the ZnO in wurtzite structure. The strongest peak comes from (002) plane reflection which suggests that the most of the crystallites have grown along the c axis.

The lattice parameters were determined: $a = 3.2506 \pm 0.0002 \text{ \AA}$ and $c = 5.2063 \pm 0.0003 \text{ \AA}$. It means that the a constant of the sample is slightly higher than the standard value (3.2498 \AA) and the c constant is nearly equal (within the experimental error) to the standard value (5.2066 \AA).

Morphology of the ZnO samples was examined by scanning electron microscope Carl Zeiss Neon 40. The SEM measurements confirmed that the ZnO layers are polycrystalline. The average size of the nanocrystals that form the layer is of the order of 200–300 nm (see inset in Fig. 1).

3. Measurements and discussion

The Raman spectroscopy was measured in backscattering geometry with use of a TRIAX 550 spectrometer with a LN₂-cooled charge-coupled device (CCD) detector. For an exact calibration of the wavelength, mercury spectral lines were recorded. A 532 nm light of a frequency doubled Nd:YAG laser was used. Measurements were made in a continuous-flow helium cryostat (Konti Mikro) in 6–300 K temperature range.

Due to the low strain, the phonon modes were sharp and well visible in our samples. Few peaks related to E_1 , E_2 and A_1 phonon modes were observed. The strongest peak at $438.2(5) \text{ cm}^{-1}$ (at 300 K) was the E_2^H mode. It had width of 9 cm^{-1} (FWHM). The $E_1(\text{TO})$, $A_1(\text{TO})$ and $E_1(\text{LO})$ peaks were visible at 381 cm^{-1} , 410 cm^{-1} , and 585 cm^{-1} , respectively. The peak positions were in agreement with previously published data [6–8]. The Raman spectra measured at different temperatures are plotted in Fig. 2.

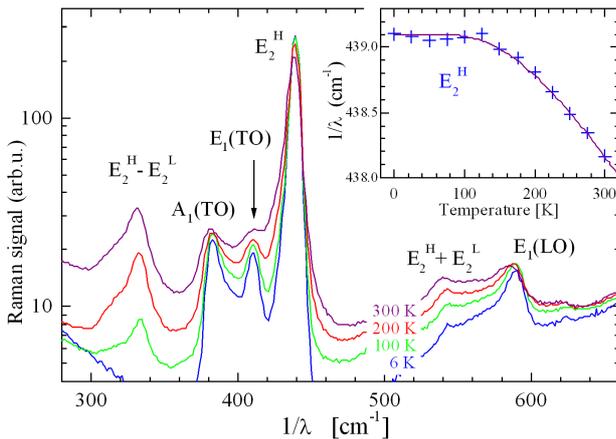


Fig. 2. Raman spectra at different temperatures. The inset: thermal dependence of the E_2^H peak energy: crosses — data points, line — fit of Eq. (1).

During heating, we observed red-shifts of the phonon peaks (in agreement with previous reports [9, 10]), increase of widths and changes of heights of the phonon peaks (see Figs. 2 and 3). At 4 K, the E_2^H peak was at $1/\lambda = 439.1(5) \text{ cm}^{-1}$ (energy = $54.44(5) \text{ meV}$). At the room temperature, the E_2^H energy decreased and the difference was $0.9(1) \text{ cm}^{-1}$. The temperature-related frequency shifts between 6 K and 300 K were: 1.6 cm^{-1} , 0.5 cm^{-1} , and 4 cm^{-1} , for $A_1(\text{TO})$, $E_1(\text{TO})$, and LO, respectively.

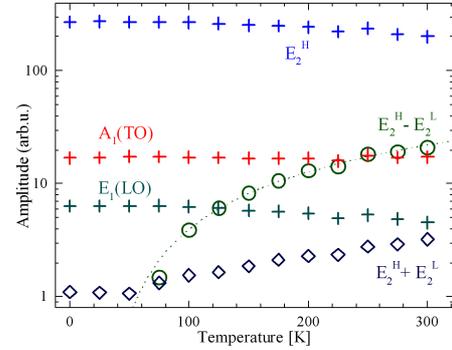


Fig. 3. Dependence of the Raman peak amplitudes versus temperature. Dotted line is fit of Eq. (2).

The thermal frequency change coefficients $\phi = d \ln(\omega)/dT$ near the room temperature were calculated: $2.8(2)$, $0.2(2)$, $1.5(1)$, $4.5(5) [\times 10^{-5} \text{ K}^{-1}]$ for $A_1(\text{TO})$, $E_1(\text{TO})$, E_2^H , and LO modes, respectively. Comparing coefficient ϕ with volume thermal expansion coefficient $\beta = 1.4 \times 10^{-5} \text{ K}^{-1}$ [11], one can obtain the Grüneisen parameter: $\gamma = \phi/\beta$. These parameters were 2, 0.1, 1.1, and 3.2 for $A_1(\text{TO})$, $E_1(\text{TO})$, E_2^H , and LO, respectively. The high γ value for the LO peak was probably an artifact caused by existence of two LO modes with slightly different energies. During increase of temperature, the amplitude of the low-energy $A_1(\text{LO})$ mode grew in respect of high-energy $E_1(\text{LO})$ mode. Such $A_1(\text{LO})$ behavior was already observed in above room temperature measurements [10].

The temperature dependences of the phonon frequency and linewidth come from anharmonic terms in the vibrational Hamiltonian of the crystal lattice [12, 13] leading to phonon–phonon interactions. These processes could explain both energy and width dependences of the Raman lines. The thermal dependences should be proportional to the numbers of phonons, as given by

$$E(T) = E_0 - \frac{b}{\exp(\theta/T) - 1}, \quad (1)$$

where the oscillator energy, E_0 , thermal change, b , and phonon activation temperature, θ , were the fitting parameters. The obtained activation energy, $k_B\theta = 64(5) \text{ meV}$, is slightly higher than E_2^H energy (54 meV) but close to $E_2^H + E_2^L$ energy (67 meV).

In addition to one-phonon peaks, it was possible to observe few two-phonon peaks: at 332 cm^{-1} , 541 cm^{-1} , and

1160 cm^{-1} . The phonon–phonon interactions were due to anharmonic terms in the potential. The peaks were previously reported [14] and attributed to the second-order features. The 332 cm^{-1} peak was attributed to two-phonon scattering from Σ – M branch having maximum in phonon density of states at 160 cm^{-1} [14]. The 541 cm^{-1} peak was attributed to two-phonon scattering on 270 cm^{-1} phonons near the M point [14]. The 1160 cm^{-1} peak was attributed to two-phonon scattering on LO phonons from Γ point. We agree with the last identification but we suggest that 332 cm^{-1} and 541 cm^{-1} were anti-Stokes and Stokes lines of photon scattering on two (E_2^{H} and E_2^{L}) phonons. It means that the 332 cm^{-1} line was due to absorption of E_2^{L} phonon with simultaneous emission of E_2^{H} ($E_2^{\text{H}} - E_2^{\text{L}}$ process). The difference of E_2^{H} energy 438 cm^{-1} and E_2^{L} 102 cm^{-1} gives value close to the 332 cm^{-1} . The 541 cm^{-1} line is due to simultaneous emission of E_2^{L} and E_2^{H} phonons ($E_2^{\text{H}} + E_2^{\text{L}}$ process). The 1160 cm^{-1} line is due to simultaneous emission of the two LO phonons (about 580 cm^{-1} each).

Moreover our thesis about the $E_2^{\text{H}} - E_2^{\text{L}}$ line origin is strongly supported by temperature dependence of scattering probability (see Fig. 3).

The data points in Fig. 3 were obtained by fitting procedure. Background and a set of spectral curves were fitted simultaneously to the spectra measured at different temperatures.

It is visible that the amplitude of the $E_2^{\text{H}} - E_2^{\text{L}}$ peak strongly depended on temperature. It increased an order of magnitude between 80 K and 300 K, as expected for anti-Stokes processes. This observation is supported by other reports [9, 10]. For example, at 873 K the 330 cm^{-1} peak was observed to be of the same height as the E_2^{H} peak [10].

The height of the $E_2^{\text{H}} - E_2^{\text{L}}$ peak should depend on the number of the E_2^{L} phonons. This number, n_s , is given by the Bose–Einstein statistic

$$n_s(\Omega, T) = \frac{1}{\exp(\hbar\Omega/k_{\text{B}}T) - 1}. \quad (2)$$

The curve (2) was fitted to the experimental dependence (see dotted line at Fig. 3.). The energy obtained from the fit, $\hbar\Omega = 13.8 \pm 2$ meV ($1/\lambda = 111 \pm 15$ cm^{-1}) was nearly equal to the expected energy of the E_2^{L} phonons (98 cm^{-1} [7] to 102 cm^{-1} [8]).

4. Conclusion

Several one- and two-phonon lines in nanocrystalline ZnO were observed by the Raman spectroscopy and identified. Temperature dependence analyzed for the Raman peaks showed that the energy of the peaks decreased significantly with increase of temperature. The

obtained Grüneisen parameters were 2, 0.1, 1.1, and 3.2 for $A_1(\text{TO})$, $E_1(\text{TO})$, E_2^{H} , and LO modes, respectively.

Two-phonon processes were observed: absorption of the E_2^{L} phonon with simultaneous emission of the E_2^{H} phonon, at 332 cm^{-1} , simultaneous emission of the E_2^{L} and E_2^{H} phonons, at 541 cm^{-1} , and emission of two LO phonons, at 1160 cm^{-1} . The amplitudes of the Stokes peaks were weakly sensitive to temperature, whereas amplitude of the Stokes+anti-Stokes peak $E_2^{\text{H}} - E_2^{\text{L}}$ increase strongly during heating.

Both the thermal frequency change and two-phonon processes arise from the phonon–phonon interactions which are due to anharmonic terms in the potential.

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