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Time Dependent Changes in Ag Doped YBCO Superconductors

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Magnetization measurements at 77 K were conducted on the single-crystalline samples of YBCO superconductor doped with Ag prepared by the top-seeded melt-growth process. The single-grain samples had a nominal composition $Y_{1.5}Ba_2(Cu,Ag)_3O_y$. Magnetization measurements on the samples, oxygenated by two different processes: by standard oxygenation and by high pressure oxygenation, repeated after 24 months, revealed time dependent changes in magnetization behaviour and critical transition temperatures, T_c , which could be associated with redistribution of oxygen vacancies and their interaction with Ag dopant atoms.

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1. Introduction

Melt-processed (RE)–Ba–Cu–O (REBCO, RE = Y, Gd, Nd etc.) bulk superconductors represent a class of materials with great superconducting and magnetic properties which make them very attractive for wide-ranging technical applications. Nevertheless, one of the main interests is still focused to achieve higher critical temperatures, T_c , and further improvement of critical current density, J_c , especially at higher magnetic fields. It is possible to considerably enhance critical current density, J_c , of these materials for example by irradiation [1] or chemical doping [2]. In the case of YBCO melt-solidified bulks it was already shown that Ag addition can enhance pinning properties [3] besides the improvement of mechanical properties [4]. Additionally, Feng et al. consider that Ag-doping of YBCO can effectively prevent samples from degradation [5] which is another important question for practical applications of these materials. We report on time dependent changes in the pinning behaviour and critical transition temperatures, T_c , of Ag-doped YBCO bulk superconductors. The influence of two different thermo-chemical treatments is shown.

2. Experimental details

Ag-doped YBCO bulk single-grain superconductors with a composition: $Y_{1.5}Ba_2(Cu_{1-x}Ag_x)_3O_y$ ($x = 0.0025$ to 0.05) + 25 mol.% Y_2O_3 + 1 wt% CeO_2 were fabricated by the top-seeded melt-growth process (TSMG) in air atmosphere. $SmBa_2Cu_3O_y$ single crystals were used as nucleation seeds. Samples for oxygenation and magnetization measurements had a shape of a slab with the

dimensions $2 \times 2 \times 0.5$ mm³, the smallest dimension was parallel to the c -axis.

The samples were oxygenated by two different processes: standard oxygenation (SO) and high pressure oxygenation (HPO). During SO the samples were heated to 800 °C in a flowing oxygen and kept there for 2 h, then cooled down to 400 °C and held at this temperature for 240 h and finally cooled down to room temperature. During HPO the samples were oxygenated under an oxygen pressure of 160 bars at 750 °C for 24 h with progressively increasing oxygen pressure.

The magnetization measurements were done by a vibrating-sample magnetometer (VSM) with magnetic fields up to 6 T applied parallel to the c -axis of the crystal, at the temperature of 77 K. The critical current densities, J_c , were calculated from the magnetization hysteresis loops using the extended Bean model [6] for rectangular samples. The critical transition temperatures, T_c , and the transition widths, ΔT_c , were determined from the magnetic transition curves taken after zero-field cooling as the $T_c(50\%)$ and, as the $T_c(90\%) - T_c(10\%)$, respectively, of this curve in an applied external magnetic field of 2 mT.

3. Results and discussion

The critical transition temperatures, T_c , and transition widths, ΔT_c , of $YBa_2(Cu_{1-x}Ag_x)_3O_{7-\delta}$ samples measured in short time after oxygenation and repeated after 24 months are presented in Fig. 1a and b, respectively. The diagrams show that even small changes in the composition of YBCO crystal can influence T_c and ΔT_c for both SO and HPO. It can be seen, especially, in the case of the $YBa_2(Cu_{0.95}Ag_{0.05})_3O_{7-\delta}$ sample. The decrease of T_c and increase of ΔT_c for this concentration of Ag

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dopant can suggest that the Cu atoms in the YBCO lattice are partially substituted by the Ag atoms.

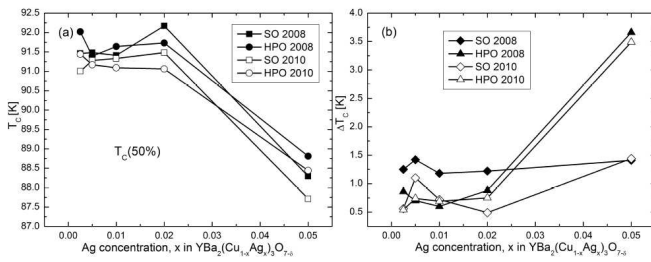


Fig. 1. Critical transition temperatures, T_c (a), and transitions widths, ΔT_c (b), as a function of Ag concentration, x .

After 24 months, the decrease of T_c was observed for all samples after SO as well as for samples after HPO (Fig. 1a). No significant changes were found in the dependences $\Delta T_c(x)$ for samples after HPO (Fig. 1b) whereas the decrease of ΔT_c was found for almost all samples after SO. This can be related to higher non-homogeneous distribution of oxygen in entire sample after SO in comparison to samples after HPO. We suppose that samples after SO become more homogeneous after aging.

Figure 2 shows the dependences of the critical current density, J_c , on applied magnetic field, B , at 77 K for five different Ag concentrations, x in $\text{YBa}_2(\text{Cu}_{1-x}\text{Ag}_x)_3\text{O}_{7-\delta}$ oxygenated by SO (Fig. 2a) and HPO (Fig. 2b) measured in 2008. Magnetization measurements on the same samples, repeated after 24 months, are shown in Fig. 2c and d, respectively.

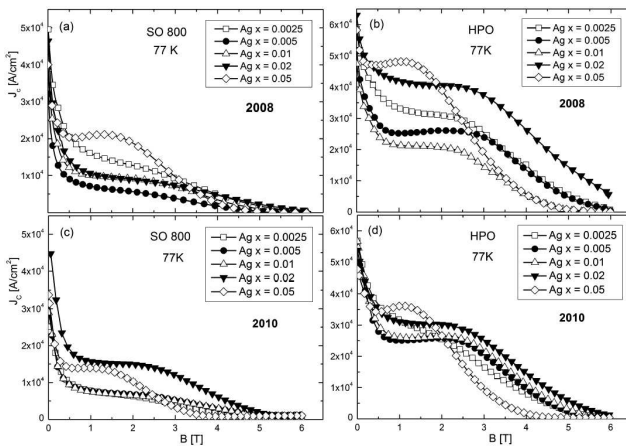


Fig. 2. Field dependences of critical current density, J_c , at 77 K for all Ag concentrations, x in $\text{YBa}_2(\text{Cu}_{1-x}\text{Ag}_x)_3\text{O}_{7-\delta}$ oxygenated by SO (a) and HPO (b) repeated after 24 months (c) for SO and (d) for HPO.

In the case of SO treatment (Fig. 2a), the peak effect was observed only at the highest nominal dopant concentration $x = 0.05$. The explanation of this behaviour was given by detailed observation [7] which revealed in-

homogeneous distribution of Ag in the single-grain bulk samples caused by the low Ag partition coefficient at solidification. Significant improvement of pinning properties of Ag-doped YBCO bulk superconductors can be reached by HPO [8] (Fig. 2b) which eliminates the formation of oxygenation cracks and increases the effective cross-section of the samples [9].

Magnetization measurements repeated after 24 months show changes in the field dependence of the critical current densities, J_c (Fig. 2c,d). The decrease of J_c for all concentrations of the Ag dopant was observed, except of the sample after HPO with concentration of Ag $x = 0.01$ and $x = 0.02$ for the sample after SO, where the increase of J_c was found. Possible origin of such changes is a redistribution of pinning centres in the samples caused by diffusion of oxygen vacancies and their interaction with Ag dopant atoms.

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