

14th Czech and Slovak Conference on Magnetism, Košice, Slovakia, July 6–9, 2010

Temperature Dependence of Magnetization Process in Bistable Amorphous and Nanocrystalline FeCoMoB Microwires

P. KLEIN^a, R. VARGA^a AND M. VAZQUEZ^b^aInstitute of Physics, Faculty of Sciences, P.J. Safarik University, Park Angelinum 9, 041 54 Kosice, Slovakia^bInstituto de Ciencia de Materiales de Madrid, CSIC, 28049 Madrid, Spain

We have studied the effect of thermal treatment on the magnetic properties of amorphous and nanocrystalline Fe₄₀Co₃₈Mo₄B₁₈ microwires. The magnetization process was measured within the temperature interval from 80 to 425 K. Microwires shows complex temperature dependence of the switching field in amorphous state due to the presence of complex stress distribution induced during production. After nanocrystallization, the switching field depends linearly on the measuring temperature that makes such microwires ideal for sensing applications.

PACS numbers: 75.50.Kj, 75.60.Ej, 75.60.Jk

1. Introduction

Amorphous glass coated microwires are ideal materials for new, modern microsensors [1]. Having positive magnetostriction, the microwires are characterized by a magnetic bistability due to their magnetization process that runs through the single Barkhausen jump when the external field exceeds the so-called switching field, H_{sw} [2]. Such bistability can be used in many applications like magnetic coding, sensors of magnetic field, mechanical stress, etc. However, the crucial parameter for their application is the time and temperature stability. Therefore, new classes of nanocrystalline microwires have been developed recently, having bistable behaviour, high Curie temperature and being structurally very stable [3, 4].

The temperature dependence of the magnetization process in glass-coated microwires has been studied by different authors [5, 6]. The aim of this paper is to present a study of temperature dependence of magnetization process on different thermally treated amorphous and nanocrystalline FeCoMoB microwires.

2. Experimental

Amorphous glass-coated Fe₄₀Co₃₈Mo₄B₁₈ microwires were produced by the Taylor–Ulitoski method. The diameter of metal core is 16 μm and thickness of glass coating is 9 μm . The length of all samples used in the magnetization measurements was 1.5 cm and in the switching field measurement it was 10 cm.

The microwires were annealed for 1 h at different temperatures in the range 300–825 K in order to obtain various stage of crystallization. The saturation magnetization was measured by the SQUID Magnetometer (Quantum Design) within the temperature range of 10–380 K in the magnetic field 1 T. Temperature dependences of the switching field in the temperature range 80–425 K were measured by an induction method at a frequency of 50 Hz.

3. Results and discussions

Due to the production process (rapid quenching, drawing), there is a complex stress distribution of the stresses induced in as-cast microwires. Therefore, the temperature dependence of switching field is also complex (Fig. 1). Firstly, it decreases at low temperature, then it rises, close to the 300 K, showing discontinuous behaviour at high temperatures. Such temperature dependence can be understood taking into account the complex stress distribution and the additional stresses applied on microwire due to a different thermal expansion coefficient of metallic nucleus, α_m , and glass-coating, α_g . Varying the temperature of microwires, the complex stress distribution in as-cast microwires varies, too.

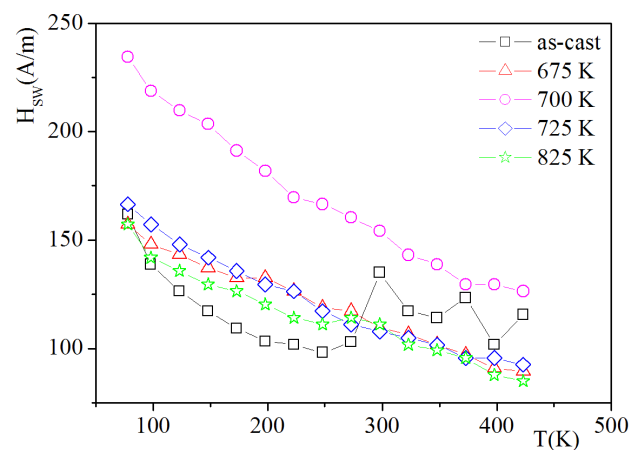


Fig. 1. Temperature dependence of switching field for heat-treated FeCoMoB microwire. Annealing temperature as a parameter.

Similar behaviour can be found from the measurement of saturation magnetization M_s (Fig. 2). Small disconti-

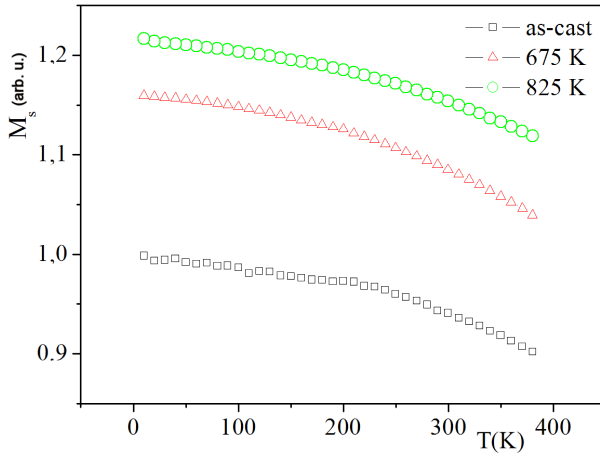


Fig. 2. Temperature dependence of saturation magnetization for heat-treated FeCoMoB microwire. Annealing temperature as a parameter.

nities in the saturation magnetization are visible at low temperatures. The temperature dependence of magnetization is not smooth reflecting the changes in the complex stress distribution due to stress coming from glass coating. However, the applied field 1 T is strong enough to saturate the sample and such laborious behaviour is observed only below 200 K. At higher temperature, the temperature dependence of M_s is smooth.

Annealing of microwires at 675 K leads to the stress relaxation and homogenization of the stress distribution in microwires. However, this temperature is still low enough for microwires to be amorphous [4]. The main anisotropy that determines magnetic properties of amorphous microwires is magnetoelastic one: $K_\sigma \sim \lambda_s \sigma$ (K_σ is the anisotropy constant, λ_s being the magnetostriction, σ being the applied stress). Hence the homogenization of the stress distribution leads to the smooth temperature dependence of the switching field. It decreases with a temperature following the decrease of the stress σ applied on microwire by glass-coating due to the change of temperature ΔT : $\sigma = E(\alpha_g - \alpha_m)\Delta T$ (E is the Young modulus). Moreover, the above-mentioned annealing (together with homogenization of the stress distribution) leads to the increase of M_s , which has also very smooth temperature dependence.

Annealing at 700 K leads to an increase of the switching field. It was found that crystallization process starts close to this temperature [4]. However, crystalline precipitates are small, having a long distance between them. Hence, they act the role of the pinning centers for the domain wall and it results in the increase of H_{sw} (Fig. 1).

Annealing at the temperatures higher than 700 K leads to the appearance of nanocrystalline structure [4] that is characterized by the small crystalline grains (smaller than exchange length) embedded in amorphous matrix.

Exchange interaction of crystalline grains leads to the averaging out of magnetocrystalline anisotropy that results in a magnetic softness (switching field decreases). FeCoMoB composition was selected due to its positive magnetostriction in order to obtain bistable hysteresis loop. Such bistability was confirmed by measuring of hysteresis loops in [4]. Therefore, the temperature dependence of H_{sw} is driven mainly by magnetoelastic interaction of magnetic moments with the stress applied due to the glass-coating. The nanocrystalline FeCoMoB microwire is as soft as the amorphous one after the stress relaxation (by annealing at 675 K) [4]. The advantage of FeCoMoB composition is that magnetic softness obtained by annealing is almost independent of the temperature of annealing within wide range of temperatures 725–875 K. The temperature dependence of H_{sw} is also similar for different temperature of annealing within this range. However, its structural stability of nanocrystalline microwires is much higher even at high temperatures [4]. Moreover, the saturation magnetization of nanocrystalline microwires is higher than that of amorphous one (Fig. 2).

4. Conclusions

The temperature dependence of magnetization process in FeCoMoB amorphous and nanocrystalline glass-coated microwires has been studied. It was shown that nanocrystallization leads to the magnetic softness of such microwires. The temperature dependence of the switching field is smooth and almost linear and saturation magnetization increases by almost 25%. This shows that the nanocrystalline FeCoMoB microwires are very promising materials for sensor applications.

Acknowledgments

This work was supported by the project NanoCEX-mat No. ITMS 26220120019, Slovak VEGA grant No. 1/0076/09, MVTS 6RP/Manunet/UPJS/08 and project VVGS UPJS No. 30/09-10.

References

- [1] M. Vázquez, *Handbook of Magnetism and Advanced Magnetic Materials: Advanced Magnetic Microwires*, Eds. H. Kronmüller, S. Parkin, Wiley, Chichester, UK 2007, p. 2193.
- [2] H. Chiriac, T.A. Ovari, *Prog. Mater. Sci.* **40**, 333 (1996).
- [3] E. Komova, M. Varga, R. Varga, P. Vojtanik, J. Bednarcik, J. Kovac, M. Provencio, M. Vazquez, *Appl. Phys. Lett.* **93**, 062502 (2008).
- [4] P. Klein, R. Varga, P. Vojtanik, J. Kovac, J. Ziman, G.A. Badini-Confalonieri, M. Vazquez, *J. Phys. D, Appl. Phys.* **43**, 045002 (2009).
- [5] M. Vázquez, A.P. Zhukov, K.L. Garcia, K.R. Pirota, A. Ruiz, J.L. Martinez, M. Knobel, *Mater. Sci. Eng.*, **375-377**, 1145 (2004).
- [6] A. Zhukov, *Appl. Phys. Lett.* **78**, 3106 (2001).