

Raman Piezospectroscopy of Phonons in Bulk 6H-SiC

K. GRODECKI^{a,b}, A. WYSMOŁEK^a, R. STĘPNIEWSKI^a, J.M. BARANOWSKI^{a,b}, W. HOFMAN^b,
E. TYMICKI^b AND K. GRASZA^{b,c}

^aInstitute of Experimental Physics, Faculty of Physics, University of Warsaw, Hoża 69, 00-681 Warsaw, Poland

^bInstitute of Electronic Materials Technology, Wólczyńska 133, 01-919 Warsaw, Poland

^cInstitute of Physics, Polish Academy of Sciences, al. Lotników 32/46, 02-668 Warsaw, Poland

Raman piezospectroscopy of high quality 6H-SiC crystals is presented. The crystals used in experiments were grown by the seeded physical vapor transport method. Uniaxial stress up to 0.9 MPa, obtained using a spring apparatus, was applied along [11–20] and [10–10] directions. It was found that the application of uniaxial stress led to different energy shifts of the observed phonon excitations in the investigated 6H-SiC crystals. The obtained pressure coefficients vary in the range 0.98–5.5 cm⁻¹ GPa⁻¹ for different transverse optical phonon modes. For longitudinal optic phonon modes pressure coefficients in the range 1.6–3.6 cm⁻¹ GPa⁻¹ were found. The data obtained could be useful in evaluation of local strain fields in SiC based structures and devices including epitaxial graphene.

PACS numbers: 62.50.–p, 42.65.Dr, 63.20.–e, 78.30.–j

1. Introduction

The Raman spectroscopy is one of the classical nondestructive methods to study phonon excitations in semiconductors. This method could be substantially extended when the experiments are performed under uniaxial stress applied along specific crystallographic axes of the investigated crystals. The application of the Raman piezospectroscopy to silicon carbide was very limited, since this technique requires high quality, well oriented bulk crystals, free of micropipes, capable to sustain high uniaxial stresses. That is why the knowledge of stress coefficient of phonon excitations in SiC was restricted to the experiments in which only hydrostatic pressure was used [1, 2]. Recently high quality 6H-SiC crystals, which match superior requirements of piezospectroscopic experiments, were grown at the Institute of Electronic Materials Technology in Warsaw [3]. This allowed us to carry out Raman scattering experiments on 6H-SiC crystals under uniaxial stress presented here. The results provide information about stress coefficients in SiC and can be useful in strain analysis of SiC based devices.

2. Experimental details

The 6H-SiC crystals used in the piezospectroscopic experiments were grown by the seeded physical vapor transport (PVT) method on *c*-face (0001) of 6H-SiC seeds in argon atmosphere. A graphite resistance heater was used to achieve growth temperature in the range 2100–2300 °C. Two X-ray-oriented 6H-SiC samples with the dimensions of 10 × 1.4 × 1.6 mm³ were studied.

Uniaxial stress up to 0.9 MPa, obtained using a spring apparatus, was applied along *X* ([11–20]) and *Y* ([10–10]) directions, which are perpendicular to the *c*-axis of the investigated 6H-SiC crystals. The Raman scattering experiments were performed at room temperature in backscattering geometry using 532 nm line from Nd-YAG laser as a source of continuous wave excitation. The scattered light, propagating perpendicularly to the stress direction, was collected in two configurations: with the *k*-vector of light being parallel (*k* ∥ *c*) or perpendicular (*k* ⊥ *c*) to the *c*-axis of the investigated 6H-SiC crystals.

3. Results and discussion

Representative Raman spectra of 6H-SiC obtained for *X* and *Y* oriented samples in the absence of the external uniaxial stress are presented in Fig. 1.

The Raman spectra obtained for both *X* and *Y* oriented sample are very similar. They are dominated by a group of peaks due to TO-phonon excitations and LO-phonon mode. The observed excitations can be attributed to folded phonon modes of different symmetry [4]. Two modes of *E*₂ symmetry: TO(X1/3) and TO(X1) are observed at 784 cm⁻¹ and 812 cm⁻¹, respectively. The LO(X0) phonon mode of *A*₁ symmetry is present at 978 cm⁻¹. It is observed that the spectra measured for *k* ∥ *c* and *k* ⊥ *c* configurations in the absence of the external uniaxial stress show different intensity patterns. This reflects the properties of selection rules obeyed in the investigated system.

In Fig. 2 the Raman spectrum obtained for *k* ⊥ *c* configuration for TO(X1) mode under applied pressure of

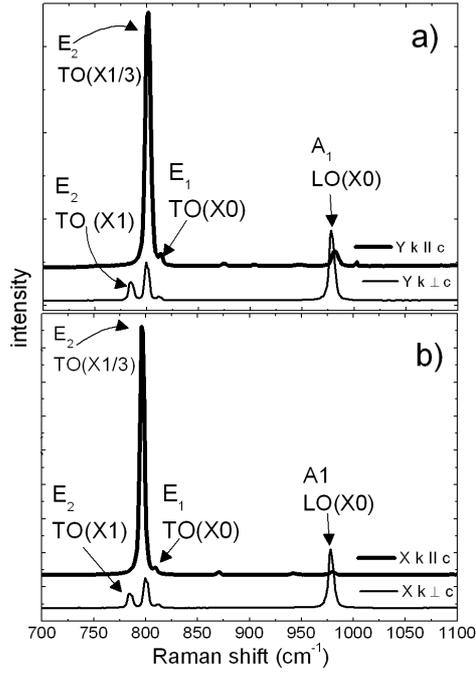


Fig. 1. Raman spectra of 6H-SiC samples oriented along (a) Y $[11\bar{2}0]$ and (b) X $[10\bar{1}0]$ directions for $k \parallel c$ and $k \perp c$ light propagation.

0.9 MPa along Y direction is presented, together with zero-stress spectrum. In spite of the fact that the observed shift is rather small, the central position of the peak can be evaluated for each applied stress value. In Fig. 3 stress dependences obtained for TO(X1/3) as well as for LO(X0) phonon modes are shown.

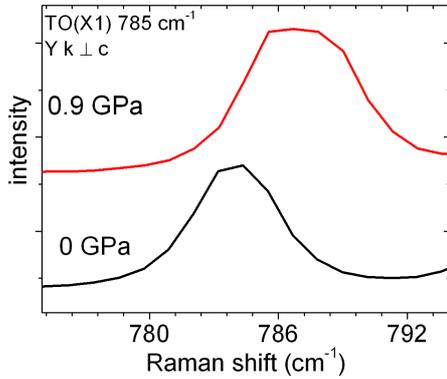


Fig. 2. The Raman spectra due to the TO(X1) phonon mode obtained for $k \perp c$ configuration at 0.9 GPa (upper curve) and in the absence of external stress (lower curve).

It is observed that under external stress the modes show linear energy shifts with pressure coefficients dependent both on the stress direction and the light propagation direction (Fig. 3). This is a manifestation of the symmetry of particular phonon excitations in 6H-SiC.

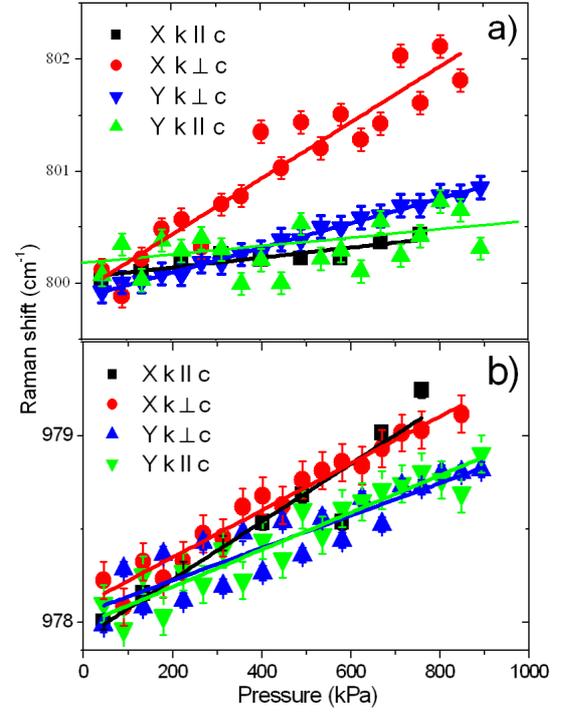


Fig. 3. Pressure dependence of Raman excitations obtained for TO(X1/3) (a) and LO(X0) (b) phonon modes for different stress and light propagation directions.

The pressure coefficients obtained for different stress and light propagation directions are collected in Table.

TABLE

Pressure coefficients obtained for different phonon modes under uniaxial stress applied along X and Y direction for different orientation of the light k -vector with respect to the c -axis of the 6H-crystals.

Stress direction	k -vector direction	E_2	E_2	E_1	A_1
		TO(X1)	TO(X1/3)	TO(X0)	LO(X0)
[cm ⁻¹ GPa ⁻¹]					
X	$k \parallel c$		0.44		1.55
X	$k \perp c$	2.58	2.16	2.35	1.26
Y	$k \perp c$	1.13	1.10	1.31	0.87
Y	$k \parallel c$		0.38		0.99

The obtained uniaxial stress coefficients of different phonon modes are smaller than the values reported for the hydrostatic pressure, which are of about 3.8 cm⁻¹ GPa⁻¹ for LO phonon mode and 3.1 cm⁻¹ GPa⁻¹ for TO phonon mode [1, 2]. The obtained data would provide information about stress coefficients in SiC in the way proposed in Ref. [5]. The obtained information could also be useful for the analysis of strain effects in the SiC-based structures and devices, especially in the case of graphene structures grown by annealing of SiC at high temperatures [6]. Since graphene

layers are usually grown on [0001]-oriented substrates, backscattering Raman experiments would easily provide information about strain components perpendicular to the c -axis, thus corresponding to the $k \parallel c$ configuration examined in this paper. Performing Raman experiments on epitaxial graphene layers, simultaneous information on phonon excitations in SiC could be obtained. This would allow for an evaluation of the influence of strain fields in the SiC substrate on the optical and electrical properties of graphene layers studied [7–9].

4. Conclusions

The Raman experiments in high quality 6H-SiC crystals were performed. Preliminary data show that application of uniaxial stress (up to 0.9 MPa) leads to different linear energy shifts of the phonon excitations in the 6H-SiC crystals. The results obtained allow to determine stress coefficients in SiC and can be useful in evaluation of strain effects in the SiC-based structures.

Acknowledgments

The work was partially supported by Polish Ministry of Science and Higher Education, Grants PBZ-

-MEiN-6/2/2006 and DWM/N179/PICS-FR/2008 and EU project No. MTKD-CT-2005-029671.

References

- [1] A. Debernardi, C. Ulrich, K. Syassen, M. Cardona, *Phys. Rev. B* **59**, 10 (1999).
- [2] C. Ulrich, A. Debernardi, E. Anastassakis, K. Syassen, M. Cardona, *Phys. Status Solidi B* **211**, 293 (1999).
- [3] A. Wyszmołek, K. Wardak, R. Stępniewski, J.M. Baranowski, M. Potemski, E. Tymicki, K. Graszka, *Acta Phys. Pol. A* **110**, 437 (2006).
- [4] S. Nakashima, H. Harima, T. Tomita, T. Suemoto, *Phys. Rev. B* **62**, 24 (2000).
- [5] Z.C. Feng, W.J. Choyke, J.A. Poweil, *J. Appl. Phys.* **64**, 12 (1988).
- [6] J. Borysiuk, R. Bożek, W. Strupiński, A. Wyszmołek, K. Grodecki, R. Stępniewski, J.M. Baranowski, *J. Appl. Phys.* **105**, 023503 (2009).
- [7] Z.H. Ni, T. Yu, Y.H. Lu, Y.Y. Wang, Y.P. Feng, Z.X. Shen, *Am. Chem. Soc. Nano* **2**, 2301 (2008).
- [8] N. Ferralis, R. Maboudian, C. Carraro, *Phys. Rev. Lett.* **101**, 156801 (2008).
- [9] J.A. Robinson, C.P. Puls, N.E. Staley, J.P. Stitt, M.A. Fanton, K.V. Emtsev, T. Seyller, Y. Liu, *Nano Lett.* **9**, 3 (2009).