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Simulation of Reaction-Induced Phase Separation in Surface Alloy

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Using kinetic Monte Carlo method we simulate the dynamics of biatomic Au_{0.3}Ni_{0.7} surface alloy separation on Ni(111) due to Ni(CO)₄ out-reaction. The experiment of Vestergaard et al. is modeled by counterbalancing dynamical processes and interactions between reactants. The simulations demonstrate step flow rate increase with CO coverage, $c_{\rm CO}$, in qualitative agreement with the experiment only for $c_{\rm CO} \lesssim 0.45$ monolayer. Moreover, we demonstrate both CO influence on reaction process and Au domain formation.

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1. Introduction

Bimetallic catalysts can have radically different properties as compared to those observed for their constituents. Up to now the main focus has been on bimetallic systems that form thermodynamically stable alloys in the bulk. Recent advances in surface growth technologies allowed to fabricate surface alloys, which are formed out of two thermodynamically bulk-immiscible metals only in the surface layer. The Au–Ni surface alloy on Ni(111) is one of such novel catalysts with improved properties in activity and selectivity. However, it was unstable at industrially relevant high CO pressures due to Ni(CO)₄ formation reaction [1].

The kinetic Monte Carlo (MC) modeling of experimentally observed Au-Ni separation in Au/Ni(111) surface alloy was started in [2] by neglecting the role of CO. Here we demonstrate qualitatively the most general trends of the separation dynamics: (i) step flow rate (reaction front) dependence on CO coverage (i.e., pressure) and (ii) Au islands growth dependence on reactant mobility.

2. Simulation model and parameters

Computer simulations are performed at room temperature on a hexagonal lattice of size 128×128 in the units of lattice constant, a_0 , with periodic boundary

conditions. Three types of reactants corresponding to CO, Au, and Ni are considered. Initially lattice is covered with 30% of Au and 70% of Ni atoms randomly as in experiment [1] with two steps created by removing Au and Ni atoms from twelve rows in the middle of the lattice. To meet the observation [2] that Au coverage on step ridges is close to unity, we replace Ni step atoms by Au. The MC simulations are performed exploiting the standard model and pair algorithm [3] with the following processes.

(i) Adsorption of CO from gas phase to both empty lattice sites and atop Ni atoms is assumed as non-activated. In this case the adsorption rate at experimental conditions ($p_{\rm CO} = 10-60$ Torr) could be estimated from the collision theory as $k_{\rm a} = (2-12) \times 10^6$ s⁻¹.

(ii) Desorption rate of CO $k_{\rm d} = 1.1 \times 10^{-7}$ is found from equation $k_{\rm d} = k_0^{\rm d} \exp(-E_{\rm d}/k_{\rm B}T)$ with the pre-factor $k_0^{\rm d} = 10^{13} {\rm s}^{-1}$ and activation energy $E_{\rm d} = 1.18 {\rm eV}$ for Ni(111) surface in the absence of Au [4].

(iii) Only Au atoms can perform activated jumps to the nearest neighbor (NN) empty lattice sites, while CO and Ni are immobile. The estimate of activation barrier for Au diffusion is $E_{\rm a}^{\rm Au} = 0.53$ eV [5]. Thus, at room temperature the experimental jump rate is $\nu^{\rm Au} = 1.1 \times 10^4$ s⁻¹, where we have chosen a pre-factor $\nu_0^{\rm Au} = 10^{13}$ s⁻¹.

(iv) Formation of Ni(CO)₂ occurs with reaction rate k_r , when a Ni atom with atop CO reactant has both another CO and Au atom in NN positions and the total number of its NN Au and Ni atoms is less than five (border, kink or corner site). Since Ni(CO)₂ formation is rate limiting step in Ni(CO)₄ formation [6], we assume that Ni(CO)₂ leaves the surface instantly.

These estimates indicate that MC simulations with experimental rates ν^{Au} , k_a , and k_d are unrealistic for today's computing facilities. Therefore a reduction of rates is required, while keeping the experimental rate inequalities $k_d \ll \nu^{Au} < k_a$ valid. In all simulations we fix both CO adsorption rate $k_a = 4 \times 10^3 \text{ s}^{-1}$ and reaction rate $k_r = 10 \text{ s}^{-1}$. At the same time we consider different Au diffusion (19 and 1900 s⁻¹) and CO desorption rates $(2 \times 10^{-11}-6 \times 10^{-3} \text{ s}^{-1})$ as sensitive parameters, which allow us to manipulate Au cluster growth and CO surface coverage, respectively. Thus we assume that k_d plays a role of pressure in our model.

Adsorption and desorption rates of CO are modified by CO–CO and CO–Au NN interactions [3]. Since these interactions are unknown, as a starting guess we set them repulsive and equal to $v_{\rm CO-CO} = 0.16$ eV and $v_{\rm Au-CO} = 0.08$ eV in all simulations. For simplicity the attraction between Au–Au and Ni–Au is set equal: $v_{\rm Au-Au} = v_{\rm Au-Ni}$.

3. Results

A fixed desorption rate, k_d , leads to a saturated CO surface coverage, $c_{\rm CO}$, which establishes after Ni removal and Au cluster formation, see inset of Fig. 1a.

Thus, we obtain a time dependence of Ni surface coverage, $c_{\rm Ni}$, on $c_{\rm CO}$ by a subsequent change of $k_{\rm d}$, Fig. 1a. For $c_{\rm CO} \lesssim 0.45$ ($k_{\rm d} < 3.5 \times 10^{-7} {\rm s}^{-1}$) we observe an increase in Ni removal rate with increase in $c_{\rm CO}$, since CO molecules adsorbed from a gas phase readily participate in carbonyl formation reaction. Ni removal rate reaches maximum at about $c_{\rm CO} \approx 0.45$ and further it decreases with increase in $c_{\rm CO}$, since adsorbed CO molecules start to hinder Au diffusion thus blocking the reaction.



Fig. 1. (a) Contour plot of Ni coverage (given in boxes) time dependence on $c_{\rm CO}$. Parameters: $\nu^{\rm Au} = 19 \ {\rm s}^{-1}$, $v_{\rm Au-Au} = -0.05 \ {\rm eV}$. Solid and dotted lines in the inset correspond to Ni and CO coverage, respectively, at $k_{\rm d} = 10^{-9} \ {\rm s}^{-1}$. (b) Step flow rate dependence on CO concentration (i) circles — as in case (a), (ii) triangles — $\nu^{\rm Au} = 1900 \ {\rm s}^{-1}$, and (iii) squares — $\nu^{\rm Au} = 1900 \ {\rm s}^{-1}$, $v_{\rm Au-Au} = -0.075 \ {\rm eV}$.

The gradient of Ni coverage time dependence leads to step-flow rate, v, Fig. 1b, which can be compared with experiment [1]. An increase in v with $c_{\rm CO}$ agrees well with the experiment and is rather insensitive to the change of simulation parameters. However, the following decrease is the artefact of the simplifications made in our model. We assume that CO can occupy fcc lattice sites and, despite mutual repulsion, two CO molecules can be NN, contrary to indications [7] of a a strong CO–CO NN repulsion and occupation of bridge and top sites at high CO coverage. The rise of Au hopping rate by two orders expands the step flow rate growth region up to $c_{\rm CO} \approx 0.5$ and twice increases v, Fig. 1b. Contrary, an increase by 1.5 times of Au–Au (and Au–Ni) interaction in the limit of fast Au diffusion leads to double reduction of the step flow rate, which now accidentally coincides with the slow Au diffusion case.

A time development of phase separation with Au islands formation at the wave front is presented in Fig. 2a, b. Let us note that the increase in Au hopping rate, as expected, increases the size of Au islands, Fig. 2b, c.



Fig. 2. Snapshots of MC simulations with Au (black) and Ni (gray) reactants. Parameters: $v_{Au-Au} = -0.05 \text{ eV}$, $k_d = 1.2 \times 10^{-4} \text{ s}^{-1}$. Times from left to right: (a) 60 s, (b) 280 s at $\nu^{Au} = 19 \text{ s}^{-1}$, and (c) 140 s at $\nu^{Au} = 1900 \text{ s}^{-1}$.

In summary, we have found the set of kinetic parameters which allows, at least qualitatively within the limits of current computer resources, to explain step flow rate increase with CO coverage as well as Au islands formation and ripening process.

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