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Carbon-Encapsulated Magnetic Nanoparticles Based on Fe, Mn, and Cr for Spintronics Applications

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We succeeded in the arc-plasma synthesis of carbon-encapsulated Fe, Cr, and Mn-based nanoparticles. The transmission electron microscopy, Mössbauer spectra (of iron) and SQUID magnetometry results demonstrate that the products of the synthesis contain metals and its carbides. The nanoparticles show ferromagnetic or superparamagnetic behavior at high temperatures, which is demanded for nano-spintronics applications.

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1. Introduction

Carbon-based nanostructured materials are the promising candidates for future applications in novel electronics. Unique flexibility in sp-hybridization of carbon valence orbitals results in different stable carbon allotropic forms, affinity for bonding with other atoms and gives nearly ten million different chemical compounds, with the potential use in miniaturized electronics. Carbon nanotube transistors and logic gates, conducting carbon-based polymers have been already demonstrated [1-7].

Graphite is one of the candidates for future electronics — single graphene layers form two-dimensional materials and show many interesting physical phenomena in the presence of magnetic field, like half integer quantum Hall effect,

electron-hole degeneracy, and vanishing carrier mass near the point of charge neutrality. Intercalation of graphite with various elements yields charge-transfer compounds that may be considered as an analog of semiconductor doping [8, 9].

In this communication we propose to integrate ferromagnetic nanomaterials with several graphene layers in so-called carbon-encapsulated magnetic nanoparticles (CEMNPs) [10–16]. Such hybrid structures possess interesting chemical and physical properties. The carbon shell protects the encapsulated crystallites from the oxidation and further agglomeration. Moreover, the magnetic field from the magnetic core should influence on graphite band electrons [9].

2. Synthesis of nanomaterials

The synthesis was carried out in an arc discharge chamber in helium atmosphere. The discharge current and voltage was between 70 A and 25 V, respectively. The graphite anodes (6 mm in diameter, 10 cm in length) were drilled (3 mm in diameter, 3 cm in length) and filled with a mixture of pure Fe or Cr or Mn (depending on the process) and graphite powder. The as-prepared anodes contained 12-14% at. of metal. The pure graphite rods were used as cathodes. The as-prepared electrodes were arc sublimated. The resulted soot was collected from the walls of the reactor. All products were subjected to the purification procedure (to remove non-encapsulated metals). A sample of soot was put into a flask, boiled with HCl for 24 h, then washed with plenty of water and ethanol, and finally dried in air at 380 K [16].

3. The experimental results and discussions

3.1. Transmission electron microscopy

Morphology of nanosized Mn, Fe, and Cr-based CEMNPs was studied by the transmission electron microscopy (TEM). The core of each nanocapsule consists of magnetic metal or its carbides. Such a core is surrounded by several graphite layers (as in the case of Fe and Mn) or by the amorphous carbon (Cr). The synthesis results in a mixture of the nanoparticles with different sizes, from several nm

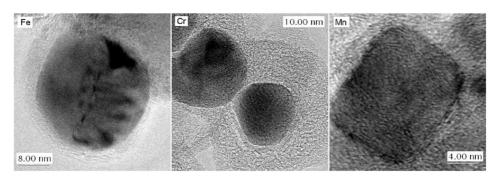


Fig. 1. Morphology of nanosized Mn, Fe, and Cr-based CEMNPs.

to over 100 nm. Typical morphology of obtained particles is presented in Fig. 1. Usually the nanoparticles are attached to each other forming a kind of large agglomerates. This is due to the high surface energy of the nanoparticles. Among the nanoparticles there is also an amorphous carbon and soot.

Selected area electron diffraction provides a detailed piece of information about the phase composition. The peaks corresponding to pure metallic phases and metal-carbides were found in the diffraction patterns. The more quantitative results of the core composition were obtained by the Mössbauer spectroscopy.

3.2. Mössbauer spectroscopy

The Mössbauer spectra were measured in transmission geometry at room temperature. A constant-acceleration spectrometer with a 25 mCi 57 Co-in-Rh source was used. The Mössbauer measurements allowed the identification of iron-containing phases in Fe-based nanoparticles. The relative fractions of the phases were calculated as a ratio of the area of the relevant subspectrum to the total spectral area, assuming similar Debye–Waller factors for each phase. Isomer shifts are given with respect to α -Fe foil.

The fitting of the Mössbauer spectra allowed the evaluation of the relative atomic fractions of iron as the magnetic or paramagnetic phases (Fig. 2). The quantitative analysis evidences the difference in the relative contents of the magnetic phases observed in the Fe-based samples studied. A significant contribution of the magnetic carbide Fe₃C phase is observed (43%). The relative fractions of the magnetic α -Fe and paramagnetic γ -Fe phases are 36% and 21%, respectively. The rest of Fe is in different phases of Fe–C mixture, mostly in an austenite γ -Fe(C).

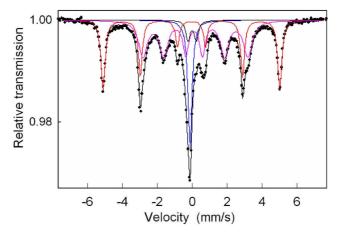


Fig. 2. Mössbauer measurements of iron-containing phases in Fe-based nanoparticles. Experimental points are marked by circles. The fitting curve consists of four phases α -Fe, γ -Fe(C) (austenite), and Fe₃C.

3.3. Magnetic measurements

Magnetic properties of encapsulated Mn, Fe, and Cr (and their carbides) were studied by means of SQUID magnetometry in the temperature range of 10.0-300.0 K and magnetic fields 0.0-6.0 T. Diamagnetic contributions to the signal (resulting from amorphous carbon and graphite powder) were subtracted in order to obtain saturation at high magnetic fields. The CEMNPs showed ferromagnetic and superparamagnetic behavior at high temperatures, similar for Mn, Fe, and Cr, while at low temperatures different behavior was observed, which depended on composition of magnetic cores (Fig. 3). If the nanocapsules contain pure metallic manganese or chromium one should expect antiferromagnetic behavior. Indeed, the antiferromagnetism manifests at low temperature, where the magnetization attains saturation at higher magnetic fields. However even at 10.0 K the dominant signal comes from ferromagnetic interaction, which means that Mn, Fe, and Cr-based CEMNPs contain ferromagnetic particles.

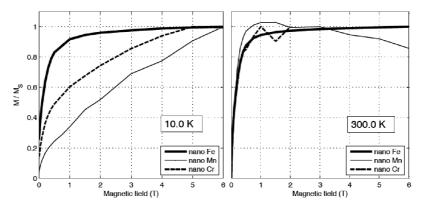


Fig. 3. Magnetization of encapsulated Mn, Fe, and Cr (and their carbides) as a function of magnetic field at 10.0 K and 300.0 K. Diamagnetic contributions were subtracted. Magnetization M is normalized to the value of saturation magnetization $M_{\rm S}$ which is (in emu/g units) at 10 K for Fe $M_{\rm S}=60$, Cr $M_{\rm S}=1.0$, Mn $M_{\rm S}=1.2$ and at 300 K for Fe $M_{\rm S}=53$, Cr $M_{\rm S}=0.4$, Mn $M_{\rm S}=0.15$.

This ferromagnetism is probably caused by some metal-carbide compounds, similarly to the Fe-carbides, whose presence was detected by means of the Mössbauer spectroscopy (Fig. 2). Magnetic properties of Mn-carbide were not investigated yet, however it was mentioned that it is ferromagnetic [17]. Chromium carbides like Cr₃C₂, Cr₂₃C₆, Cr₇C₃ are known (from the literature of structure investigations), but there is no information about their magnetic properties. If samples contain also metal oxides one can expect antiferromagnetism of MnO and ferromagnetism of CrO₂. However, since the synthesis of Fe, Mn, and Cr nanocapsules is similar, and since there was no evidence of any Fe-oxides from

the Mössbauer spectroscopy (Fig. 2), one should rather exclude the presence of metal-oxides in our samples. Therefore, the ferromagnetic properties of investigated CEMNPs are due to the ferromagnetic properties of metal-carbides.

At higher temperatures, the magnetization saturates at 0.5 T (Fig. 3). The value of observed coercivity for each sample is small and comparable to the coercivity of the superconducting coil of the SQUID magnetometer. Therefore, it is difficult to judge if CEMNPs are ferro- or superparamagnetic and some further investigations are needed.

For spintronics devices working at room temperatures the origin of magnetism of nanoparticles is irrelevant [18]. However the value of saturation magnetization of iron-based nanoparticles is almost 60 times larger than the one of manganese or chromium (Fig. 3 caption). Therefore, the iron-based nanoparticles are more preferable for future spintronics devices.

4. Conclusions

We succeeded in the arc-plasma synthesis of Fe, Cr, and Mn-based nanoparticles. The plasma synthesis in carbon arc provides also carbides and therefore the obtained material is non-homogeneous. The nanoparticles show ferromagnetic or superparamagnetic behavior at high temperatures, which is demanded for nanospintronics applications. The next step toward the practical applications will focus on the removing the amorphous carbon and the separation of the magnetic phases via sedimentation route.

Acknowledgments

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