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Study of Excimer Laser Irradiation Effects on Properties of Crystalline YBaCuO Thin Films

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The influence of UV laser irradiation on the physical properties of epitaxial $\text{YBa}_2\text{Cu}_3\text{O}_{7-\delta}$ thin ($< 1 \mu\text{m}$) films fabricated by laser ablation was studied. The samples were irradiated by pulsed excimer laser beam at different incident energy densities $E_{\text{ir}} \leq 130 \text{ mJ/cm}^2$, i.e. 2–3 times below the ablation threshold of YBaCuO. The analysis of the sample cross-section using transmission electron microscope shows that such irradiation forms a disordered layer on the top of the crystalline film. Sample surface amorphization was observed also by atomic force microscope. Due to strong UV irradiation absorption in YBaCuO this surface layer acts like a protective cover for the sample interior. Only a small reduction of superconducting critical temperature T_c was observed. Both transport measurements in normal state and magneto-optical studies in superconducting state indicate that between the disordered layer and the bottom non-modified part of the film there exists a wide transitional region with reduced carrier concentration.

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1. Introduction

High energy laser irradiation of high-temperature superconductor (HTSC) materials can lead to nonreversible modifications of their physical properties [1].

In the case of thin films (thickness $< 1 \mu\text{m}$), irradiation by IR or visible sources induces structural transformations in the whole sample volume [2, 3]. The influence of UV irradiation is limited to a near-surface region of film due to the short absorption length $d = 44 \text{ nm}$ in this spectral range [4]. This phenomenon can be used to form e.g. a thin dielectric passivation layer on the top of the thin film superconducting structure.

2. Experimental

In this paper we present a brief overview of our investigation of the effects of laser irradiation on the properties of epitaxial $\text{YBa}_2\text{Cu}_3\text{O}_{7-\delta}$ thin films obtained by laser ablation from stoichiometric “123” target. The films were deposited on $5 \times 5 \text{ mm}^2$ single crystalline substrates at heater temperature 785°C and at 280 mTorr oxygen pressure. The excimer KrF (wavelength $\lambda = 248 \text{ nm}$) laser with the pulse duration $\tau = 28 \text{ ns}$, repetition rate 10 Hz and energy density 1.5 J/cm^2 at the target surface was used. After deposition, the samples were cooled down to room temperature with the rate of -20 K/min at 300 Torr oxygen pressure, and patterned, as required, for resistance, the Hall effect, and magneto-optical measurements using conventional UV photolithography and wet-etching process. The samples were irradiated by KrF and XeCl ($\lambda = 308 \text{ nm}$, $\tau = 28 \text{ ns}$) lasers at energy density $E_{\text{ir}} \leq 130 \text{ mJ/cm}^2$ on film surface, which was close to the vaporization (etching) threshold $E_{\text{eth}} = 110 \text{ mJ/cm}^2$ [4] and was 2–3 times below the ablation threshold $E_{\text{th}} = 0.3\text{--}0.4 \text{ J/cm}^2$ of YBaCuO [5]. Irradiation was carried out in ambient conditions.

3. Results and discussion

The atomic force microscope (AFM) images (left panel of Fig. 1) show that laser annealing causes partial melting of defects on the film top (like outgrowths) and remarkably reduces the surface roughness. On the other hand, the cross-section transmission electron microscope (TEM) image (right panel of Fig. 1) of irradiated film gives direct evidence that at high laser fluence the surface layer of thin film becomes amorphous. This “A” region appears to be separated from the crystalline part “C” of the film by a sharp boundary.

A fast increase in the amorphous surface layer thickness at the beginning of the laser treatment produces a marked growth of sample resistance $R(N)$, where N is the number of consecutive laser pulses (Fig. 2a). Due to a strong absorption of UV irradiation in YBaCuO [4], the surface layer considerably reduces the impact of laser light on film interior. As a result, the dependence of R on the irradiation dose becomes weaker (at least at $E_{\text{ir}} < E_{\text{eth}}$) and the crystalline YBaCuO thin film remains superconducting even after several hundreds of laser shots (Fig. 2b).

UV irradiation increases both film normal state resistance $R = \rho_{ab}lw/t$ and the Hall voltage $U_{\text{H}} = R_{\text{H}}IH_z/t$ (see Fig. 2c), where l , w , and t are the length,

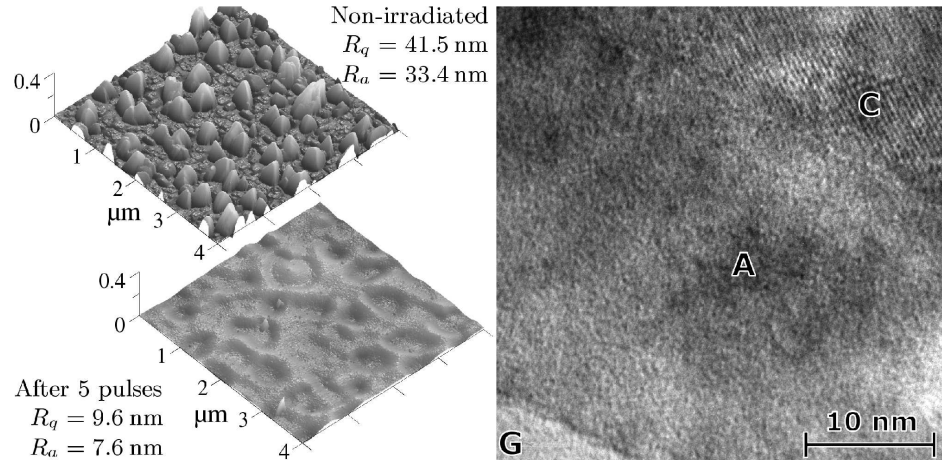


Fig. 1. Left panel: AFM images of surface of 100 nm thick YBaCuO film deposited on SrTiO₃ substrate before and after irradiation by XeCl laser at energy density $E_{ir} = 100 \text{ mJ/cm}^2$. R_q and R_a are roughness average and RMS value of the thickness fluctuation, respectively. Right panel: TEM photograph of cross-section of the 200 nm thick YBaCuO film grown on SAT(70%)–CAT(10%)–LA(20%) substrate and irradiated by three pulses of KrF laser at $E_{ir} = 100 \text{ mJ/cm}^2$ per pulse. Amorphous and crystalline parts of the film are denoted as (A) and (C), respectively, the layer of glue on the sample surface — as (G).

width, and thickness of the sample, respectively, ρ_{ab} is the longitudinal resistance, R_H is the Hall coefficient, I is the measurement current, and H_z is the external magnetic field (in the considered case $H_z = 0.5 \text{ T}$). Assuming that changes of R and U_H are caused only by the reduction of t , we can expect that the following expression is true:

$$\frac{R^{ir}(T)}{R(T)} = \gamma \frac{U_H^{ir}(T)}{U_H(T)}, \quad \gamma = 1, \quad (1)$$

where R^{ir} and U_H^{ir} relate to the irradiated sample. However, for sample 1 $\gamma = 1.3$ and for sample 2 $\gamma = 1.5$. It means that a wide transitional region with reduced carrier concentration (possibly oxygen-depleted) was formed under the amorphous top layer. Temperature dependence of the Hall angle θ_H ($\tan(\theta_H) = R_H H_z / \rho_{ab}$) shown in Fig. 2d for both virgin and irradiated samples has a typical form for HTSC materials proposed by Anderson [6]: $\cot(\theta_H) = AT^2 + B$, where A and B are constants. Growth of both A and B corresponds to decreasing oxygen concentration in the sample [7] and to increasing concentration of point defects in Cu–O planes of YBaCuO [8].

Measurements of magnetic moment distribution in the superconducting films at various temperatures, from 10 to 70 K, and in the external magnetic fields up to 350 Gs were realized by means of a ferromagnetic iron-garnet indicator,

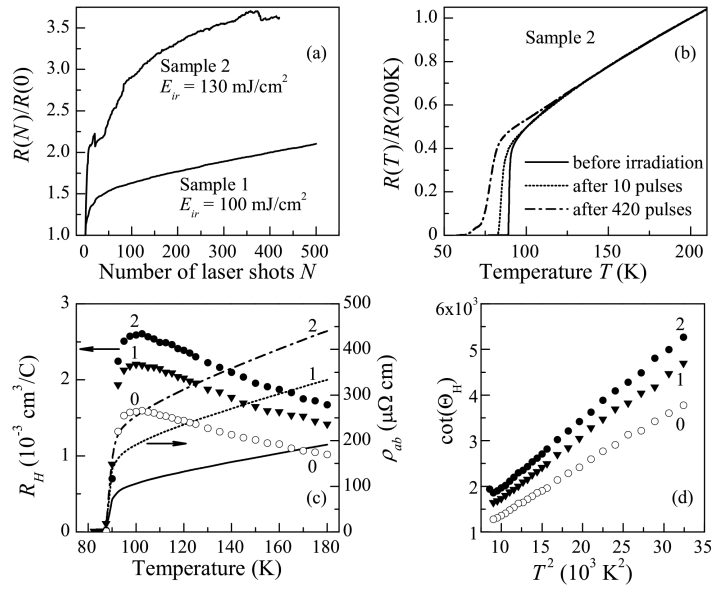


Fig. 2. (a) Normalized sample resistance R as a function of number of KrF laser pulses N ; (b) resistance vs. temperature characteristics of YBaCuO thin film deposited on SAT-LA for different irradiation doses; temperature dependences of the Hall coefficient, resistivity (c) and the Hall angle (d) for virgin film grown on SrLaGaO₄ and for samples irradiated by 1 and 2 pulses of KrF laser at $E_{ir} = 100 \text{ mJ/cm}^2$.

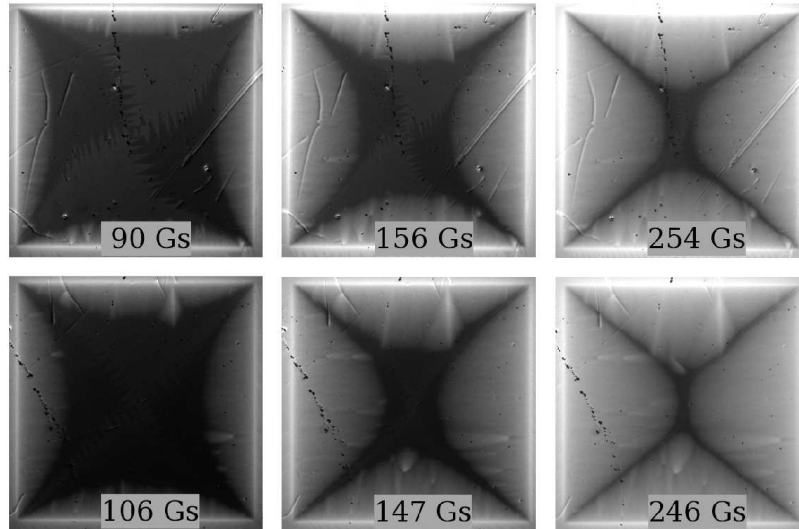


Fig. 3. Visualization of magnetic field penetration in virgin (top 3 images) and laser treated YBaCuO film at 20 K (bottom row). The 200 nm thick film was deposited on SAT-CAT-LA substrate and irradiated by 3 pulses of KrF laser at 100 mJ/cm^2 .

which rotates the plane of polarization of linearly polarized light proportionally to the local magnetic field (Faraday effect). Images of polarized light intensity corresponding to the magnetic field distribution were obtained in such manner (Fig. 3). Magnetic field penetrates the laser treated film and escapes from it more easily than in the case of a non-irradiated sample. It means that laser treatment of the film leads to a critical current I_c decrease in the irradiated areas. The decrease in superconducting layer thickness should lead to an equivalent reduction of the absolute value of I_c . In the future, we plan to use the magneto-optical images in order to calculate I_c , and to use these data to estimate the thickness of the non-modified layer.

4. Conclusions

To conclude, our results indicate that high energy density UV laser irradiation forms an amorphous surface layer followed by an oxygen-depleted region in crystalline YBaCuO thin films. The film region close to the substrate remains non-modified.

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